In Situ Unzipping of Long, High Molecular Weight DNA in Solid State Nanopores

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Abstract. Nanopores are an established paradigm in genome sequencing technology, with remarkable advances still being made today. All efforts continually address the challenges associated with rapid, accurate, high-throughput, and low cost detection, particularly with long-read length DNA. We report on the *in situ* melting and unzipping of *long, high molecular weight* DNA. At varying salt concentration, we directly compare the translocation conductance and speeds between SiN and graphene nanopores at sub-10 nm pore diameters. We observe the force-induced unzipping of dsDNA at higher salt concentrations than previously reported in literature. We observe free running translocation without secondary structures of ssDNA that is an order of magnitude longer than reported before. We hypothesize that the frayed single strands at the molecule's end get captured with a higher likelihood than both ends together. In understanding this phenomena for long-read lengths, we continue to address the challenges revolving around future generations of sequencing technology.

I. INTRODUCTION

Nanopore biosensors are a rapidly advancing technology, investigating the unknowns of human genetics and genome biology. With potential applications in clinical diagnostics or personalized medicine, nanopores utilize electrophoretically driven DNA molecules through a trans-membrane channel; providing immediate and direct measurements of nucleic acids by their electrical current [1]. Biological nanopores have shown success in single-molecule detection, in both academic and commercial settings, using trans-membrane proteins like alpha-hemolysin [2], with automated ratcheting for precision [3], and more recently, *de novo* sequencing for longer read lengths ($100 - 882$ kb) [4]. Additional studies propose platforms integrating force spectroscopy over an array of nanopores accompanied with avidin-biotin anchors, removing the need for polymerase chain reaction (PCR) assays [5–7]. However, due to the structural instability and natural lifetimes of proteins, commercial scalability and long-term storage are limited.

Solid state nanopores are a more robust alternative and offer mechanical, thermal, and chemical stability with pore size customization (for a review, see Dekker [8]). Additionally, novel fabrication techniques like focused ion-beam sculpting [9], TEM ablation lithography [10–12], and dielectric breakdown [13] have been created and successfully replicated. Previous studies with ultra thin materials like silicon nitride (SiN) and graphene have demonstrated these advantages in conjunction to dsDNA translocations. Similar to protein pores, solid-state nanopores also act as a transmembrane channel between two chambers of electrolyte solution. When electrophoretically driving DNA, an ionic current flows in response to an applied transmembrane electric field. Every translocated nucleotide perturbing the electric field generates a distinct blockage current and immediately recorded.

In order to progress towards single-molecule *sequencing*, accurate detection of individual nucleotides is paramount. Achieved through DNA denaturation, the separation of hydrogen bonds between dsDNA backbones, single-stranded DNA (ssDNA) is required to detect and read nucleotides. Previous studies of DNA denaturation have shown dependence on melting temperature when treated with varying temperature, chemical solvents, and salt concentrations [14, 15]. With initial theory on the mechanical stability, entropy, and heat capacity changes, additional studies report a force-induced unzipping of dsDNA during translocation [16, 17]. This was experimentally demonstrated and confirmed in protein pores with PCR analysis [18] and subsequently demonstrated in synthetic pores. Notably with SiN membranes at sub-2 nm pore diameters, sequencing is indicated along with unzipping kinetics, reporting an Arrhenius barrier dependence on duplex length [19] and distinctive three-level current blockades [20]. At slightly larger pore diameters ($4-10$ nm), single-stranded translocations was also observed with unfolded and unraveled configurations [21, 22]. However, these reports where conducted with short duplex DNA comprised with unstructured homo-polymers, hairpin molecules, or hybrid structures with single-stranded overhangs. In order to address the challenges of rapid, high accuracy, high throughput, and low cost detection of *long* read-length DNA, further understanding of long DNA interactions is required.

In this paper, we present *in situ* melting and unzipping of long, high molecular weight lambda-DNA in both SiN and graphene nanopores. Directly comparing between SiN and graphene membranes, we report the changes in conductance values, translocation duration and speeds at varying salt concentrations with sub-10 nm pore diameters. We observe low salt translocation of ssDNA with minimal secondary structures. More importantly, we also observe mechanical unzipping of dsDNA, at higher

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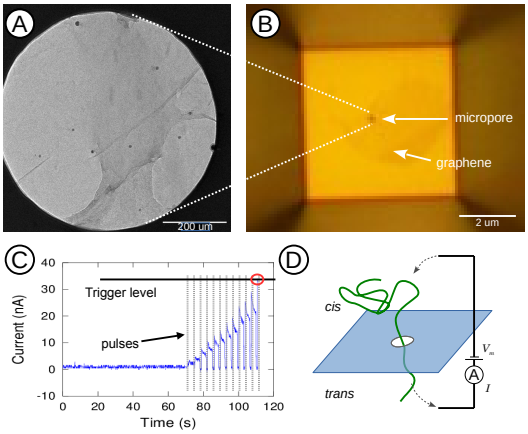

FIG. 1. Fabrication of nanopores and DNA translocation procedure. A) Transmission Electron Micrograph of suspended graphene membrane over micropore. B) Optical image of graphene membrane over micropore. C) Nanopore fabrication procedure. For graphene nanopore, a pulsing procedure was used, which was terminated at reaching a predetermined trigger level. For SiN membranes, a constant voltage procedure was used (not shown). D) A transmembrane voltage V_m is applied and the ion current I (dashed) is measured as DNA (green) translocates through a nanopore from the *cis* to *trans* side.

salt concentrations than expected from melting temperatures reported in literature [14]. We hypothesize that at low n_{KCl}, the dsDNA begins to fray [23, 24]. Due to the reduced persistence length L_P, these frayed ends have a higher likelihood of capture in the nanopore over the common mode double strand. Furthermore, under a strong electric field, *in situ* unzipping would occur as DNA molecules translocate. In understanding this phenomena, we can continue to address the challenges in solid-state devices and progress towards new generations of sequencing technology.

II. MATERIALS & METHODS

A. Materials

Solid SiN membranes were acquired from Norcada Inc (NX5004Z-60O). They consist of a $5 \times 5\,\mathrm{mm}^2$ Si frame with thickness of 200 µm, a $(20-30\,\mu\mathrm{m})^2$ membrane with thickness 12 nm, and a 60 nm thick underlayer of SiO_2. SiN membranes with a micropore for graphene studies were acquired from Norcada Inc (custom order). They either had a $5 \times 5\,\mathrm{mm}^2$ or $3.5 \times 5\,\mathrm{mm}^2$ Si frame with 200 µm thickness, a 50 µm membrane with thickness of 200 nm or 50 nm, respectively, and a 60 nm thick underlayer of SiO and a 5 nm overlayer of SiO for improved adhesion to graphene. Micropore diameters ranged from $0.5-2\,\mu\mathrm{m}$.

B. Device Fabrication

Graphene samples (Fig. 1) were made from mechanically exfoliated $5-10\,\mathrm{mm}$ size natural graphite (NGS Naturgraphit GmbH) flakes with Blue Nitto tape (Nitto Denko, SPV 224LB-PE). Graphite flakes were deposited on 300 nm SiO_2/500 µm Si wafers that were previously cleaned in separate ultrasonic baths of Acetone (PHARMCO-AAPER), IPA (99%, PHARMCO-AAPER), de-ionized Water, and dipped in a 6:1 Buffered Oxide Etch (HF, JT.Baker 1178-03). Graphene flakes were identified with an optical microscope (Meiji MT7530 BF/DF), with thicknesses determined by optical contrast using a calibration curve obtained with Atomic Force Microscopy (Pacific Nanotech Dual Scan).

We used the wedge transfer technique developed by Schneider *et al.* [25] to transfer, and deposit individual graphene flakes onto micropores. We covered individual graphene flakes in a droplet of Cellulose Acetate Butyrate (CAB, Sigma-Aldrich 419036-250G) in Ethyl Acetate (EtAc, Sigma Aldrich 319902-1L) mixture, with its location manually marked. The graphene flake and CAB transfer polymer were then wedged off from its Si/SiO_2 substrate in de-ionized water with a freshly aspirated surface, before being transferred onto a hydrophilic micropore treated with RF oxygen plasma for three minutes (Harrick Plasma PDC-32G, 18 W). Under an optical microscope with a modified stage, the graphene flake and CAB was manually positioned over the micropore before residual water completely dried. After ensuring the graphene was placed correctly (Fig. 1b), the micropore with graphene under CAB was placed on a hotplate at 75 °C for 30 min to promote secure adhesion. The CAB was then dissolved in two separate EtAc baths for 30 s. The remaining graphene flake was deposited onto the micropore by annealing at 400 °C for 30 min. Suspended graphene samples were imaged in a Transmission Electron Microscope (FEI Titan S/TEM) (Fig. 1a).

SiN samples were acquired commercially and cleaned with Piranha Etch and treated with RF oxygen plasma.

Completed devices were immersed in ethanol (EtOH, PHARMCO-AAPER, 111000200) to remove trapped air inside the micropore pyramid. A custom microfluidic flow cell was immersed in DI water with its channels preflushed to remove residual air. The pre-wetted device was then transferred and mounted in the flow cell under water. Once removed from water, the mounted device was secured and the outside of the cell was dried thoroughly before both channels were flushed with 2 M KCl buffer solution for initial $I(V)$ measurements.

C. Nanopore Fabrication

Once membranes were confirmed to be insulating, nanopores were fabricated in both SiN and graphene membranes using dielectric breakdown[13, 26]. **Graphene pores** were formed by rapid pulses of in-

creasing duration Δt and height ΔV. Depending on the desired pore size, we simultaneously observed whether R exceeded $10-20\,\mathrm{M\Omega}$ at $0.2\,\mathrm{V}$. Initial pulses were $\Delta t = 1\,\mu\mathrm{s}$ and $\Delta V = 3\,\mathrm{V}$ high. Then ΔV was slowly increased up to $10\,\mathrm{V}$. If no initial pore was formed, ΔV was reduced and Δt was increased. Nanopores formed at $6-10\,\mathrm{V}$ at $1\,\mu\mathrm{s}-1\,\mathrm{ms}$. Upon initial pore formation, pulses at $3-4\,\mathrm{V}$ of similar duration as initial formation were used to adjust the nanopore diameter d (Fig. 1d).

SiN nanopores were made by slowly ramping a *constant bias* V until a desired trigger current level I was reached. The pore was then enlarged by resetting V and ramping up again, after which the trigger level is reached at lower and lower V levels. Initial trigger currents were $I = 30-80\,\mathrm{nA}$ and initial formation occurred at $V = 4-6\,\mathrm{V}$ and post breakdown enlarging caused triggering to occur below $\sim 1\,\mathrm{V}$.

An initial value of the nanopore diameter d was determined from the open pore conductance G using the bare pore conductance G_0 and access resistance R_a [27, 28]

$$\begin{aligned} G &= (G_0^{-1} + R_a)^{-1} \\ G_0 &= \frac{\pi d^2}{4L_{\mathrm{pore}}} \underbrace{(\mu_K + \mu_{Cl})\, e n_{KCl}}_{\equiv\sigma} + \frac{\pi \mu_K \sigma_q d}{L_{\mathrm{pore}}} \\ R_a &= \frac{1}{\sigma d} \end{aligned} \tag{1}$$

Note that σ_q is surface charge density while n_{KCl} conductivity is σ. The pore diameter d and depth L_{pore} were determined self consistently using $G(n_{\mathrm{KCl}} = 2\,\mathrm{M})$ and $\Delta G(n_{\mathrm{KCl}} = 2\,\mathrm{M})$ [29].

D. Buffers and DNA

$2\,\mathrm{M}$ KCl buffer solution was made with $2\,\mathrm{M}$ KCl, $10\,\mathrm{mM}$ Tris HCl, and $1\,\mathrm{mM}$ EDTA. Additionally, Lambda DNA (N3011L, New England Biolabs) and KCl buffers were titrated with $6\,\mathrm{M}$ KOH to pH = 8. When changing n_{KCl} from $2\,\mathrm{M}$ to a low value, we track G *vs.* t and wait until it saturates at a value consistent with Eq. 1. We do not see a contribution from σ_q in these experiments, which we comment on below. Experiments were performed at room temperature.

E. Data Acquisition & Analysis

The *cis*-channel of the flow cell was flushed with lambda DNA solutions at selected KCl concentrations. The *trans*-channel flushed with selected KCl concentration only. A trans-membrane voltage V_m was applied to electrophoretically drive the DNA through the nanopore. For SiN at all n_{KCl} values $V_m \leq 100\,\mathrm{mV}$, while it was $\leq 200\,\mathrm{mV}$ for graphene nanopores. The corresponding ion current change was recorded (Fig. 2) with a data acquisition board (National Instruments PCI-6251) and custom acquisition software at $0.1-1.25\,\mathrm{MS/s}$, with a Fourier filter to remove stray interference.

To filter out low-current signal events in the presence of a high $1/f$-noise background, common in nanopore systems [30, 31], we detect events as follows. The signal was leveled by subtracting a 100-point convolution average, recording the standard deviation for every 100-point section of the leveled signal, *i. e.* the background I_{bg}, and recording the median standard deviation σ_m of that set. Possible events were recorded as exceeded parts of the signal by a multiple of σ_m, without yielding false positives: 3 to 5 σ_m. The signal was then convolved with a 2-point average, before the procedure repeated. Subsequently, events were fitted with the expected rectangular shape of height ΔI and duration τ of translocation waveforms, deduplicated, and reduced based on the χ^2 values and signal to noise level $\Delta I/\sigma_m$.

This manner of event detection is particularly useful for detecting low-signal events against a high $1/f$ background. The convolution filter preserves the squareness of the event waveform so it is not distorted when large parts of the spectrum are rejected. Furthermore, the ratio of the shortest timebase τ_s (reduced timebase due to convolution) to longest timebase τ_l (100 point baseline subtraction) gives a constant bandwidth ratio that gives constant noise RMS values $\sigma_m^2 \approx C_{LF} I_{DC}^2 \log(\tau_l/\tau_s)$ when $1/f$ noise dominates. This facilitates a numerical threshold criterion that does not depend on the part of the spectrum under consideration.

III. OBSERVATIONS

We observe a series of short changes in the ion conductance ΔG across the nanopore (Fig. 2) every time a DNA molecule translocates through the pore [8]. To ensure these fluctuations are well separated from the baseline current I_{bg} and display a distinctive peak in the ion current histogram, we limit the passage of folded molecules by fabricating relatively small nanopores, $d = 4\ldots7\,\mathrm{nm}$.

At high n_{KCl}, we normally find $\Delta G < 0$ because it is dominated by the exclusion of ions from the nanopores. Conversely at low n_{KCl}, $\Delta G > 0$, since counterion current from the DNA backbone dominates [27]. Both effects are described by

$$\Delta G_{\mathrm{pore}} = \left(\underbrace{-\frac{\pi}{4} d_{DNA}^2 \sigma' \mu_R}_{\text{exclusion}} + \underbrace{\mu_K^\star q_{L,DNA}^\star}_{\text{backbone}} \right) / L_{\mathrm{pore}} \quad . \tag{2}$$

Due to the charge $q^\star$ and cross-section $\pi d^2/4$ of dsDNA being halved during denaturation or unzipping, its conductance is also halved, $\Delta G_{\mathrm{pore,ssDNA}} = \frac{1}{2}\Delta G_{\mathrm{pore,dsDNA}}$. Over a range of n_{KCl} concentrations, the expected conductance values of dsDNA (black) and ssDNA (blue) are shown in Fig. 2 and are directly compared to our observations.

We observe two major deviations from Eq. 2 for SiN and graphene nanopores. With pore diameters between

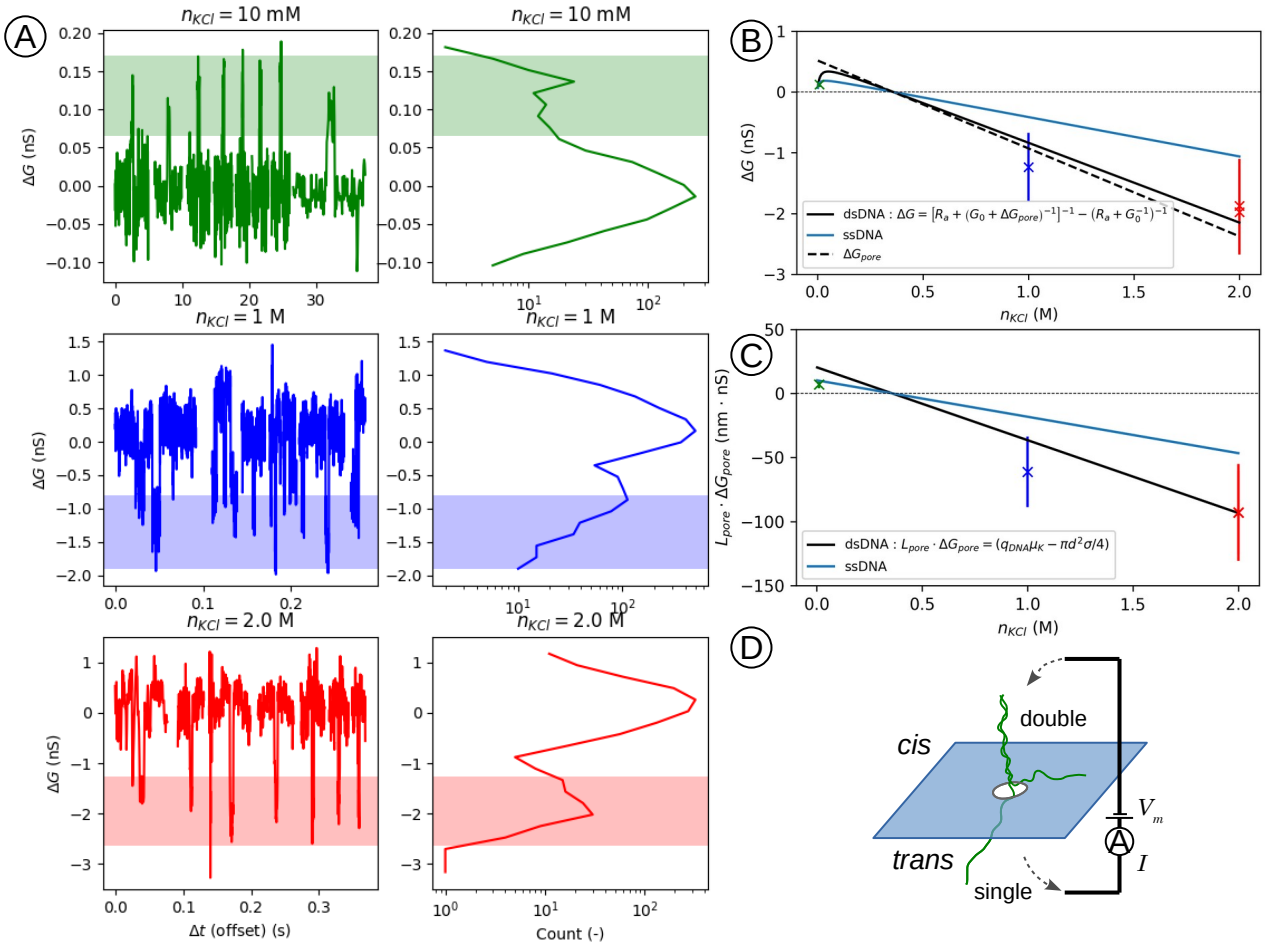

FIG. 2. Unzipping of high-molecular-weight DNA through SiN nanopores. A) Left: Subset of ΔG for events at $n_{KCl} = 0.010$ (green), 1 (blue), and 2 M (red). Right: Histogram of all event current traces for corresponding conditions in left panel with secondary peak width (shaded). B) Expected mean of ΔG values of ssDNA (blue), dsDNA (black), and Eq. 2 (dashed) *vs.* n_{KCl}. C) Expected mean of $L_{\mathrm{pore}} \cdot \Delta G_{\mathrm{pore}}$ for ssDNA (blue) and dsDNA (black) *vs.* n_{KCl}. The 2 M (red) and 1 M (blue) align with dsDNA and the 10 mM (green) aligns with ssDNA. D) Diagram of dsDNA (green) unzipping, as it moves through a nanopore in a membrane (blue) from the *cis* to *trans* side driven by a transmembrane potential V_m while monitoring the current I.

$d = 5 - 7$ nm, SiN nanopores observed at lower n_{KCl} had ΔG values significantly smaller than Eq. 2 predicts. For graphene nanopores with diameters between $d = 4 - 7$ nm, we observe an overall trend of $\Delta G > 0$ at high n_{KCl} and $\Delta G < 0$ at low n_{KCl}. However, we specifically observe at $n_{\mathrm{KCl}} \sim 220$ mM, $\Delta G < 0$ in contrast to the prediction of $\Delta G > 0$ from Eq. 2 (Fig. 3).

As an alternative indicator of DNA-graphene interactions, we also designed our graphene nanopores to deliberately be treated *without* a passivation layer, which would inhibit ssDNA binding due to pi-stacking interactions [32]. At $n_{\mathrm{KCl}} = 10$ mM, we see translocation events with $\Delta G > 0$ and much reduced μ (Fig. 3, 4). Simultaneously, we see a slow increase of background resistance R in a linear manner at a rate of $\mathrm{d}R/\mathrm{d}t \approx 10\,\%/\,\mathrm{min}$. We do not see an exponential dependence within ~ 5 min. This increase is only visible at forward bias, $V_m > 0$, not reverse bias. When applying reverse bias pulses with duration and height typical for pore enlargement, the conductance does not recover. In contrast, at higher n_{KCl}, the background resistance is stable and there is an upper limit to the R drift, *i.e.* $|\,\mathrm{d}R/\mathrm{d}t\,| \lesssim 0.5\,\%$. This increase ultimately limits the duration of the experiment and lifetime of graphene devices, and we do not observe this with SiN nanopores.

For both SiN and graphene nanopores, we studied the mobility μ of DNA driven through the nanopore. It was determined through $\mu = v/E$, where v is the translocation speed and E the effective field in the nanopore (Fig. 4). We used $E = V_{\mathrm{eff}}/L_{\mathrm{pore}}$, where V_{eff} is the voltage drop across $G_0 + \Delta G_{\mathrm{pore}}$ in a series $1/(G_0 + \Delta G_{\mathrm{pore}}) + R_a$ network that divides down the applied voltage V_m. Typical field strengths were $2\,\mathrm{MV\,m^{-1}}$ for SiN and $20\,\mathrm{MV\,m^{-1}}$ for graphene. At high ionic strengths, we find typical μ for DNA in both graphene and SiN nanopores [33]. However, at 10 mM, μ is reduced by a

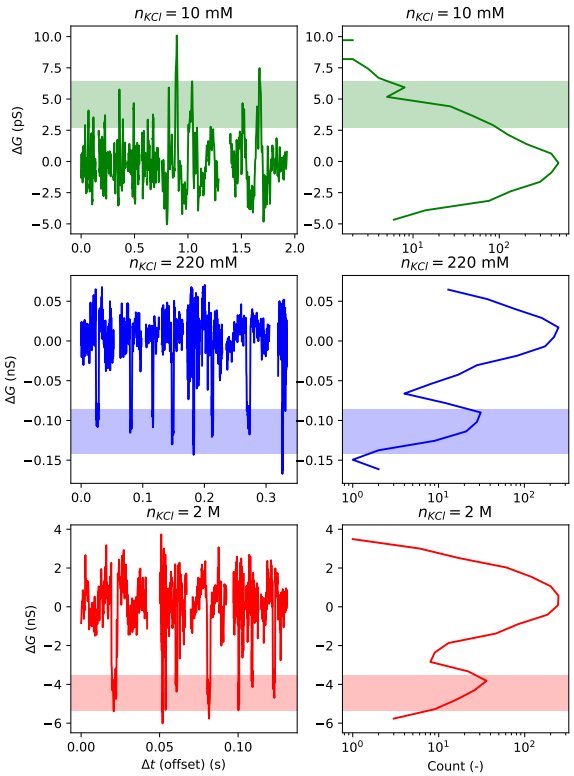

FIG. 3. Unzipping of high-molecular-weight DNA through graphene nanopores. Left: Subset of ΔG for events at n_{KCl} = 0.010 (green), 1 (blue), and 2 M (red). Right: Histogram of all event current traces for corresponding conditions in left panel with secondary peak width (shaded).

factor ~ 100, for both SiN and graphene nanopores.

IV. DISCUSSION

A. SiN Nanopores

The role of $\mathbf{R_a}$. The values for ΔG at low n_{KCl} are much lower than Eq. 2 describes (Fig. 2). We argue that in a low n_{KCl} regime, the effect of access resistance R_a must be taken into account. When the pore is open, R_a and $1/G_0$ constitute a voltage divider, whereas when DNA is inside the pore, R_a and $1/(G_0 + \Delta G_{\mathrm{pore}})$ constitutes one. If $d \ll L_{\mathrm{pore}}$ at high n_{KCl}, most of the voltage drops across the pore and ΔG follows Eq. 2. However, at low n_{KCl}, the geometric dependence of ΔG_{pore} changes and the condition $d \ll L_{\mathrm{pore}}$ no longer guarantees $R_a \ll (G_0 + \Delta G_{\mathrm{pore}})^{-1}$. Consequently, the quantity

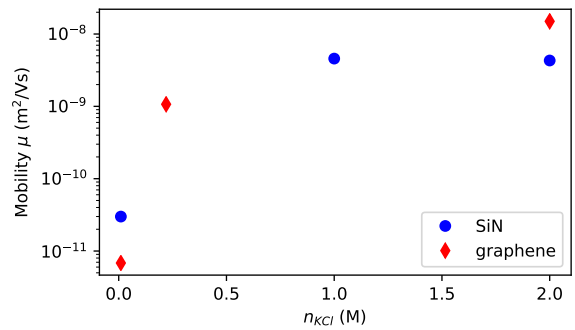

FIG. 4. Mobility of DNA driven through graphene (red diamonds) and SiN nanopores (blue circles) as a function of n_{KCl}.

actually measured is

$$\Delta G = \left[R_a + (G_0 + \Delta G_{\mathrm{pore}})^{-1}\right]^{-1} - \left(R_a + G_0^{-1}\right)^{-1} \quad (3)$$

(Fig. 2b). We extract ΔG_{pore} using the inverse of Eq. 3, causing the ssDNA and dsDNA curves at low n_{KCl} to separate (Fig. 2c) and observe a distinct grouping of ΔG_{pore} on the ssDNA curves.

The role of $\sigma_{\mathbf{q}}$. We do not see a contribution of surface charge σ_q to the open pore conductance G of our nanopores. While this is expected for graphene, for SiN, the charge at 10 mM can be expected to increase G by a factor ~ 5 [27]. We use the settling of G to a value expected according to n_{KCl} as an indicator that we have reached the right concentration. One may argue that due to residual n_{KCl}, a higher n_{KCl} is actually reached. However, that also *increases* G. One could incorporate σ_q by assuming n_{KCl} is much lower than introduced. However, in our experience, exchanging high n_{KCl} for lower n_{KCl} only appears to lead to slightly *higher* n_{KCl} than desired, if not enough fresh solvent is used and/or we do not wait long enough. Pore shrinkage combined with a higher effective n_{KCl} together could allow for a finite σ_q that would explain our open pore conductance values. However, we confirm that SiN pores usually do not shrink by going back to the starting n_{KCl} and finding the same open pore conductance. [We note that if n_{KCl} were higher, it would also significantly reduce the unzipping capability]. One possible resolution is that we have a source of contamination on the SiN membranes, despite our exhaustive cleaning the membranes with organic solvents, oxygen plasma, and Piranha. We calibrate the pore geometry as described and find $L_{\mathrm{pore}} = 20 - 40$ nm larger than the bare membrane. If this is due to an accumulation of material that does not contribute to σ_q that would be consistent with our observations.

Halving of $\mathbf{\Delta G_{pore}}$. As ΔG_{pore} is on the ssDNA curve (Fig. 2C), we suspect that we are translocating *single* strands despite starting with high-MW *double* strands. Under physiological conditions, the two strands of dsDNA bind together because the hydrogen bonds and

stacking energies are stronger than the effective electrostatic repulsion of the two negatively charged backbones. This repulsion is overcome because the cations screen the backbone with a Debye length $\lambda_D = \sqrt{\epsilon k_B T/2e^2 n_{\mathrm{KCl}}}$, where ϵ is the permittivity, k_B is Boltzmann's constant, T is the absolute temperature, and e is the electron charge. At room temperature $\lambda_D(n_{\mathrm{KCl}} = 1\,\mathrm{M}) = 0.3\,\mathrm{nm}$ while $\lambda_D(n_{\mathrm{KCl}} = 10\,\mathrm{mM}) = 3.0\,\mathrm{nm}$, which exceeds the distance between the backbones [34]. This effect favors the separation of the dual strands and lowers the melting temperature T_m. However, T_m for high-MW dsDNA at n_{KCl}= 10 mM is usually around 70 °C [14] and depends on sequence [35].

We hypothesize that we are unzipping the dsDNA due to the large E field at the pore. While the unzipping or force-induced melting of dsDNA is a well studied subject, the nanopores used previously are usually smaller than the dsDNA size ($< 2\,\mathrm{nm}$) and very short DNA ($\sim 50\,\mathrm{b}$) strands are typically employed [5, 6, 18–20, 36, 37]. When $\sim 5\,\mathrm{kb}$ strands are used, secondary structures cause large blockage signatures and a slowdown of the translocation process [22]. Here, we appear to drive *high-molecular-weight* ssDNA through a nanopore *without secondary structures*. The energetics of the unzipping process are usually analyzed using a Kramers-type 1D escape problem [5, 38]. An applied force f acts over a range Δx in overcoming the barrier, such that

$$\tau = \tau_0 \exp((E_b - f\Delta x)/k_B T) \qquad . \tag{4}$$

Using a hydrogen bond range of $\Delta x \approx 0.6\,\mathrm{nm}$ [36], a field strength of $E \approx 8.3\,\mathrm{MV\,m^{-1}}$, an effective charge of $q = \frac{1}{4}e$ per base [39], we find $f\Delta x = 0.05\ k_B T$. Once unzipping has started under physiological conditions, it takes $\sim 0.1\ k_B T$ [40] to break each additional pair, well within range of the available energy. Under this scenario, the capture rate could be limited by the initial energy barrier to get the unzipping going, afterwards the process is downhill.

Unzipping selection mechanism. We are using nanopores that are in principle large enough to accommodate both strands simultaneously. In fact, we use the dsDNA signal to calibrate d and L_{pore} *in situ*. Without a selection mechanism, dsDNA is as likely to translocate as unzipped ssDNA. How is it then possible that we appear to translocate unzipped single stranded molecules? First, since we see few if any folded and/or double-strand signals, we assume the dominant capture mode is end-capture. In addition to small d, folded capture is also limited by the much larger persistence length L_p at low ionic strength [41–44]. Second, since we are close to T_m, denaturing bubbles [45] are expected to cause a rapid time dependent fraying of the ends of the molecule [23, 24]. We put forward two hypotheses. First, the individual frayed strands at the molecule's end are expected to fluctuate more than the common mode double strand fluctuations due to the much reduced L_p of ss over dsDNA, even at low ionic strength. If this constitutes an increase in attempt frequency for capture, this would favor unzipping. Second, if the pore is charged and that charge is poorly screened, it could renormalize d thus favoring passage of single over double strands. However, we do not see a role of σ_q for SiN pores. Therefore, we do not think that surface charge of the pore plays a role in unzipping. We therefore posit that the preferential trapping of the frayed single strands is the selection mechanism responsible for unzipping over double-strand translocation. A compounding mechanism could be envisioned that works similarly to geometric selection, *i. e.* denaturation bubbles are larger than fully hybridized dsDNA and due to that size being close to d could be prevented from entering the pore. While the bubble size is most likely limited to the hydrogen bond length of 0.6 nm for few-base bubbles, it may be larger at the frayed end of dsDNA. Experiments at larger d could elucidate the role of this effect.

Grouping of translocation events. One may expect that once a single strand has translocated, its complement would be captured quickly afterwards, which could cause a grouping in inter-translocation times. The two strands would translocate quickly after one another, followed by translocation of another molecule at a longer time. We do not see such a grouping which may be due to the large scatter in inter-event times.

B. Graphene Nanopores

ΔG for graphene nanopores does not appear to follow Eq. 2, especially at $n_{\mathrm{KCl}} = 220\,\mathrm{mM}$ where the sign of ΔG is opposite from that expected (Fig. 3). We argue that the particular geometry of our graphene nanopores have an effect on ΔG. The graphene nanopores we study have a significantly lower aspect ratio $\alpha = L_{\mathrm{pore}}/d$ than SiN nanopores. This causes R_a to dominate the nanopore resistance resulting in $G \propto d$, instead of $G \propto d^2$ [12, 28, 46]. Continuum modeling shows that the current density j_i is carried mostly at the perimeter of the nanopore. Moreover, non-uniform j can also be expected due to the local dipole moments at the graphene nanopore edge. The crossover $n_{\mathrm{KCl}}{}^{\star}$ deduced from Eq. 2 assumes uniform j_i inside the nanopore. Since j_i is not uniform, but carried at the nanopore perimeter, we must compare the counter-ion current density j_c with j_i and integrate over the DNA cross-section. For small nanopores such as the ones studied in this paper, a significant part of the DNA's cross section is blocking the higher j_i at the pore edge. Therefore, this would push $n^{\star}_{\mathrm{KCl}}$ lower, and could explain why we observe $\Delta G < 0$. In contrast, for larger nanopores, if the point of translocation is approximately in the center of the nanopore, the j_i that is blocked is *lower* than the uniform j_i for nanopores with larger α. This would then push $n^{\star}_{\mathrm{KCl}}$ higher. However, a thorough experimental and theoretical evaluation of this effect has not been reported before, and is beyond the scope of this report.

Selection mechanism for graphene. Specifically

for unzipping at graphene pores, the pi bonding could cause one of the strands to be tied to the graphene, freeing up the other strand for translocation. However, we see similar μ at low n_{KCl} with graphene, which does not carry a major surface charge save for some dipoles at the nanopore edge, *e. g.* carboxyl groups.

$\mathrm{I_{bg}}$ **reduction at low** $\mathrm{n_{KCl}}$ **in graphene**. As the reduction does not appear in SiN pores at low n_{KCl}, and only at forward bias with graphene, the decrease is consistent with a very slow binding of DNA material close to the pore. As the binding does not appear reversible by voltage pulsing, pi-bonding of free bases in denaturation bubbles [45] to the hexagonal graphene lattice is a logical candidate [47]. Close to the nanopore, we expect E to be stronger than in SiN nanopores owing to the much smaller L_{pore}, even when factoring in the larger role of R_a in graphene pores. If unzipping occurs at SiN due to E, it is reasonable to expect it does so for graphene nanopores as well.

C. Both SiN and Graphene Nanopores

Single nanopores. The breakdown method of making nanopores requires nucleation of a defect, followed by a slow opening of the pore through the same defect [13]. It is possible that we have more than a single defect, and rather than opening a single defect, we are opening two or more defects. However, if we allow for a parallel current path, and use that to solve for the pore geometry, we do not get self-consistent results. We therefore conclude we only measure single nanopores.

Mobility reduction. This may be expected due to the unzipping introducing a rate limit. One can quantify the rate with Eq. 4 but the attempt frequency is difficult to determine independently for the complicated nano-electro-mechanical landscape in the vicinity of the pore and is subject of further studies. The mobility reduction can also be due to DNA adhering to the substrate and that could explain the lower μ for graphene than SiN. However, since the largest share of reduction in μ is similar for both types of pores, we hypothesize the unzipping itself is primarily responsible for lowering μ, as has been observed in space-constricted unzipping in smaller nanopores [48].

V. CONCLUSIONS

We present evidence that we are unzipping high-MW dna *in situ* due to the strong field at a nanopore entrance and translocating single strands without secondary structure which results in a $\sim$ 100-fold reduction in μ. Despite the pores being large enough to accommodate both strands, we preferentially capture single strands at the frayed ends of the DNA. We hypothesize that single end capture mechanism is due to an increased attempt frequency for capture for single over double strands owing to the much shorter persistence length.

For follow-up studies, we plan to explore the phase diagram of the unzipping process. First, raising the cell temperature to T_m should yield unambiguous melted DNA, and therefore a clear ΔG *at the same* n_{KCl} that we can compare to. Second, the trade-off between E and proximity to T_m represents a 2D phase diagram in which more points can be chosen to determine the phase boundary. Third, adjusting the pH value could alter the surface charge of the SiN surface and allow an independent inspection of the role of σ_q. Fourth, the unzipping speed should scale with energy barrier once the bias is factored in, which is a function of proximity to T_m, that can be explored further. Finally, unzipping with much larger nanopores may shed light on what role, if any, the nanopore diameter plays in favoring single-strand translocation.

VI. ACKNOWLEDGEMENTS

The research reported here was supported by the National Human Genome Research Institute of the National Institutes of Health under Award Number R21HG10056. We are grateful for the experimental assistance in TEM imaging by William A. Hubbard and Jared J. Lodico from the Regan group at UCLA and discussions with Aleksei Aksimentiev, Vincent Tabard-Cossa, and Michael Zwolak.

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[29] d is the real root of the cubic equation $l_3 d^3 + l_1 d + l_0 = 0$ where $G_2 \equiv G(n_{\mathrm{KCl}} = 2\,\mathrm{M})$, $\Delta G_2 \equiv \Delta G(n_{\mathrm{KCl}} = 2\,\mathrm{M})$, $l_3 \equiv \pi\sigma/4(1/(\Delta G_2 + G_2) - 1/G_2)$, $l_1 \equiv [-d_{DNA}^2 \pi\sigma/4 + \mu^\star q_{DNA}]/(\Delta G_2 + G_2)\lambda$, $l_0 \equiv [d_{DNA}^2 \pi\sigma/4 - \mu^\star q_{DNA}]/\sigma$. It follows $d = (0.38157x/l_3 - 0.87358 l_1/x)\lambda$, where $x = (1.7321\sqrt{27 l_3^4 l_1^2 + 4 l_3^3 l_1^3} - 9 l_3^2 l_0)^{1/3}$. The scale parameter $\lambda = 10^{-9}$ is used to keep l_3, l_1, l_0 of similar magnitude. The depth is then $L_{\mathrm{pore}} = \pi\sigma d^2/4(1/G_2 - 1/(\sigma d))$. This calibration works well for SiN nanopores, but we expect an overestimate of d and L_{pore} for graphene nanopores due to the non-uniformity of j as described.

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