# Elasticity generates indissoluble biomolecular condensates

Lingyu Meng<sup>1</sup> and Jie Lin<sup>1,2</sup>

<sup>1</sup>Peking-Tsinghua Center for Life Sciences, Peking University, Beijing, China <sup>2</sup>Center for Quantitative Biology, Peking University, Beijing, China (Dated: February 10, 2022)

While biomolecular condensates are often liquid-like, many experiments found that condensates also exhibit solid-like behaviors, which may make them irreversible and indissoluble. Despite the critical biological significance of indissoluble condensates to cellular fitness and diseases, the physical mechanisms underlying the stabilities of solid-like condensates are still unclear. In this work, we study the effects of elasticity on the dissolution of biomolecular condensates. We demonstrate that the bulk stress inside condensates may prevent the condensates from dissolution and obtain the new equilibrium conditions of elastic condensate: the osmotic pressure minus the bulk stress is uniform inside and outside condensates. To verify our theories, we simulate the two-fluid model in which the slow component corresponding to the polymer network generates elastic stress. Our theoretical predictions are nicely confirmed and independent of microscopic details. Moreover, we obtain a phase diagram on the stability of elastic condensates and identify a minimum bulk modulus for the condensates to be indissoluble, both numerically and theoretically. Our results may have implications in developing drugs targeting irreversible condensates.

19 They often have crucial biological functions 21 [10, 11], such as adaptive responses to stresses, accelerating biochemical reactions, and sequestering molecules from reactions. Therefore, the accurate regulation of biomolecular condensates' formation and dissolution is critical. Meanwhile, experiments have also found that biomolecular condensates are viscoelastic rather than completely viscous [12]: they are solid-like on a short time scale and liquid-like on a long time scale. More interestingly, they exhibit aging behaviors, and the viscoelastic relaxation time, which separates solid and liquid behaviors, increases over time [13]. Indeed, aged condensates may become indissoluble or infusible in conditions where newly formed condensates can easily dissolve or fuse [3, 4, 7, 9, 14–18]. Moreover, indissoluble condensates may affect cellular fitness, e.g., failure to dissolve condensates during mitosis leads to aberrant condensates that cause the cell-cycle arrest and ultimately cell death [19]. In general, indissoluble condensates are believed to lead to aging at the cellular or organismic level and are related to multiple diseases [15, 20–22].

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Theoretical studies on the formation and dissolution of biomolecular condensates have so far been limited to fluid models, in which the elastic nature of condensates are usually neglected [23]. Meanwhile, numerous experiments have demonstrated the importance of solid-like 47 nature on the dissolution of condensates [13, 22], and our theoretical understanding of this problem is still in its infancy. In this work, we seek to fill this gap and investigate the effects of elasticity on the dissolution and 53 following, we first introduce our theoretical frameworks 86 show in the following, the elastic force may prevent the 54 focusing on elastic condensates that are subject to an 87 dissolution.

Biomolecular condensates are widely observed in vari- 55 abrupt parameter change. Without elasticity, they are ous organisms, usually composed of proteins and RNAs 56 supposed to dissolve. We then derive the equilibrium 57 conditions for elastic condensates and find that the bulk 58 stress plays an essential role in preventing the dissolu-59 tion. A finite bulk modulus may render the condensate  $_{60}$  irreversible and indissoluble. To test our theoretical pre-61 dictions, we simulate the two-fluid model [24, 25] beyond 62 the traditional fluid model of phase separation dynamics 63 (Model H) [23, 26], by including elastic stress generated 64 by the polymer network. Our theories are nicely con-65 firmed and valid independent of the microscopic details, 66 such as the free energy form. Furthermore, we provide a 67 phase diagram of condensate stability and demonstrate 68 a minimum bulk modulus for the condensate to be indis-69 soluble. Finally, we discuss the biological implications of 70 our work and propose future directions to explore.

## Equilibrium conditions of elastic condensates

Biomolecular condensates usually have well-defined 73 viscoelastic relaxation times, below which the conden-74 sates behave as elastic materials [12, 13]. In this work, we 75 simplify the problem by considering an aged condensate 76 with its viscoelastic relaxation time much longer than 77 the time scales of biological interests, e.g., the duration 78 of cell-cycle phases. Therefore, its viscoelastic relaxation 79 time can be taken as infinite, which is the main focus 80 of this work (Figure 1). We introduce an abrupt change 81 to the attractive interaction between the monomers of 82 biomolecules, either due to environmental change, e.g., <sup>83</sup> an increase in temperature, or active cellular regulation, stability of biomolecular condensates, combining both 84 e.g., post-translational phosphorylation. In the absence analytical theories and computer simulations. In the 85 of elasticity, the condensate will dissolve. However, as we

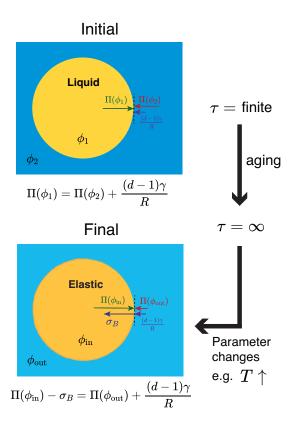


FIG. 1. Equilibrium conditions of elastic condensates. Liquidlike condensates are initially formed from liquid-liquid phase separation, which then become aged and solid-like. abrupt change in the effective attractive interaction between biomolecular monomers is introduced, e.g., through an increase in temperature or post-translational modification. The bulk stress  $\sigma_B$  in the condensate is involved in the new equilibrium condition to prevent dissolution.

In liquid-liquid phase separation, a stable condensate 89 requires the Gibbs-Thomson relation of osmotic pressure: 90  $\Pi_{\rm in}=\Pi_{\rm out}+(d-1)\gamma/R.$  Here,  $\Pi_{\rm in~(out)}$  is the osmotic 131 95 derivations in Methods),

$$\Pi_{\rm in} - \sigma_B = \Pi_{\rm out} + \frac{(d-1)\gamma}{R},$$
(1)

 $_{96}$  Here  $\sigma_B$  is the bulk stress inside the condensate (Figure 97 1). As we show later, the inclusion of bulk stress com-98 pensates the imbalance of osmotic pressures. To find the 99 expression of  $\sigma_B$ , we use the constitutive equation of the bulk stress and the continuity equation of density  $(\phi)$ 

$$\frac{\partial \sigma_B}{\partial t} = G_B \nabla \cdot \mathbf{v_p}, \qquad (2)$$

$$\frac{\partial \phi}{\partial t} = -\phi \nabla \cdot \mathbf{v_p}. \qquad (3)$$

$$\frac{\partial \phi}{\partial t} = -\phi \nabla \cdot \mathbf{v_p}.\tag{3}$$

which is responsible for the bulk stress.  $G_B$  is the bulk 103 modulus. In writing the above two equations, we assume that  $\phi$  and  $\sigma_B$  are uniform inside the condensate, which we confirm numerically later. Combining Eqs. (2, 3), we 106 obtain

$$\sigma_B = G_B \ln \left( \frac{\phi_1}{\phi_{\rm in}} \right) \tag{4}$$

where  $\phi_1$  and  $\phi_{\rm in}$  are respectively the densities of the condensate before and after the condition changes.

We note that to uniquely determine the density inside and outside the condensate,  $\phi_{\rm in}$  and  $\phi_{\rm out}$ , we still need one more equation. For liquid condensates, it is a uniform chemical potential. However, in our case, the conden-113 sate is solid; therefore, the exchange of molecules is suppressed [3, 8, 22]. Instead, we propose that condensate 115 size does not change upon the weakening of attractive interaction between biomolecules' monomers, namely, R = $R_0$ , where R and  $R_0$  are respectively the radii of elastic 118 condensate before and after the condition changes. We  $_{119}$  confirm this assumption numerically later. Given R and 120  $\phi_{\rm in}, \phi_{\rm out}$  can be calculated using the conservation of total molecular number:  $V\phi_0 = \sum_i V_i \phi_{\text{in},i} + (V - \sum_i V_i) \phi_{\text{out}}$ . 122 Here V is the total volume,  $V_i$  is the volume of condensate i, and the summation is over all condensates.  $\phi_0$  is 124 the average density over the total volume. Finally, we 125 remark that in our case, the bulk stress inside the con-126 densate stabilizes the condensate, in contrast to the bulk 127 stress outside a condensate, e.g., due to the surrounding 128 polymer network that suppresses the formation of con-<sub>129</sub> densates [27–29].

### Numerical simulations

In the following, we numerically simulate the two-fluid <sub>91</sub> pressure inside (outside) the condensate,  $\gamma$  is the surface <sub>132</sub> model in two dimensions [24, 25] to test our theories with  $g_2$  tension constant, d is the spatial dimension and R is the  $g_3$  two components: the slow component corresponding to 93 condensate radius. In this work, we propose the following 134 the polymer and the fast component corresponding to the 94 equilibrium condition for elastic condensates (see detailed 135 solvent. It is the polymer component that generates the 136 elastic stress. The average velocity field  $\mathbf{v} = \phi \mathbf{v_p} + (1 - \phi \mathbf{v_p})$  $_{137} \phi) \mathbf{v_s}$  where  $\mathbf{v_p}$  are  $\mathbf{v_s}$  are respectively the polymer and (1) 138 solvent velocity field. The dynamics of polymer density and velocity field follows (see details in Methods)

$$\frac{\partial \phi}{\partial t} = -\nabla \cdot (\phi \mathbf{v}) + \nabla \left( \frac{\phi (1 - \phi)^2}{\zeta} (\nabla \cdot \mathbf{\Pi} - \nabla \cdot \sigma) \right), \quad (5)$$

$$-\nabla \cdot \mathbf{\Pi} + \nabla \cdot \sigma - \nabla p + \eta \nabla^2 \mathbf{v} = 0. \tag{6}$$

140 Here,  $\zeta$  is the friction constant between polymer and sol-(2) 141 vent, and  $\eta$  is the viscosity. The pressure p is determined by the incompressible condition:  $\nabla \cdot \mathbf{v} = 0$ . The stress (3) 143 tensor  $\sigma = \sigma_S + \sigma_B \mathbf{I}$  where  $\sigma_B$  is the bulk stress and  $\sigma_{S}$  is the shear stress tensor. They follow the Maxwell 101 Here  $\mathbf{v_p}$  is the velocity field of the polymer, e.g., proteins, 145 fluid dynamics with the bulk and shear modulus  $G_B$  and

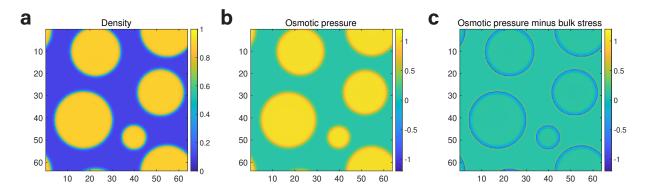


FIG. 2. Simulations of multiple coexisting condensates. (a) The density field  $\phi$  after decreasing  $\chi$  from 3.0 to 1.5. (b) The osmotic pressure  $\Pi$  from the same simulation of (a). (c)  $\Pi - \sigma_B$  from the same simulation of (a). In this figure, we take  $\phi_0 = 0.45$ ,  $G_B = 20$ ,  $G_S = 20$ , and  $\phi_c = 0.5$ .

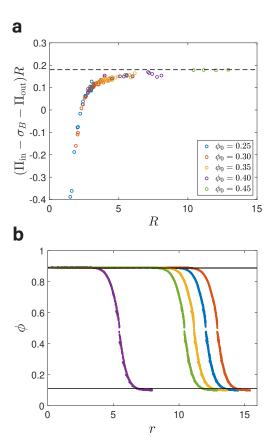


FIG. 3. Computations of the surface tension constant  $\gamma$  and predictions of the density field. (a) The inferred surface tension constant  $\gamma = (\Pi_{\rm in} - \sigma_B - \Pi_{\rm out})R$  approaches an asymptotic value in the large radius limit. (b) A comparison of the theoretical predictions of  $\phi_{\rm in}$  and  $\phi_{\rm out}$  (black lines) and the simulations in Figure 2(a). Each colored curve represents one condensate and r is the distance from the condensate center. In both (a) and (b),  $G_B = 20$ ,  $G_S = 20$ ,  $\phi_c = 0.5$ ,  $\chi_i = 3.0$ ,  $\chi_f = 1.5$ . In (b),  $\phi_0 = 0.45$ .

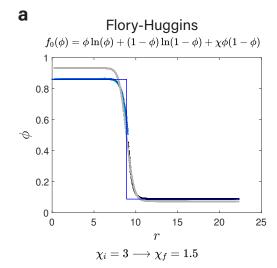
147 sates, we take the relaxation time of bulk stress to be 148 diverging at a critical density  $\phi_c$  such that

$$\tau_R^{-1}(\phi) = (\phi_c - \phi)\Theta(\phi_c - \phi),\tag{7}$$

where  $\Theta(x)$  is the Heaviside function. We remark that if we take the stress tensor to be zero in Eqs. (5, 6), they are reduced to the classical Model H [26]. The osmotic stress tensor is determined by the polymer free energy  $f(\phi), \nabla \cdot \mathbf{\Pi} = \phi \nabla f'(\phi)$  where  $f(\phi) = f_0(\phi) + \frac{C}{2} (\nabla \phi)^2$  and  $_{154}$  C is a constant. If not mentioned explicitly, we use the Flory-Huggins free energy:  $f_0 = \phi \ln(\phi) + (1 - \phi) \ln(1 - \phi)$  $\phi$ ) +  $\chi \phi (1 - \phi)$ . The condition of stable liquid condensate is that the control parameter  $\chi > 2$ . In this work, we use the osmotic pressure  $\Pi$  to represent the scalar osmotic stress computed from  $f_0$ ,  $\Pi = \phi f_0' - f_0$ .

We simulate multiple coexisting elastic condensates by changing  $\chi$  from  $\chi_i = 3$  to  $\chi_f = 1.5$ . According to our theories, they can be simultaneously stable if each has its osmotic pressure difference balanced by the bulk stress. The density field is indeed uniform inside condensates as assumed (Figure 2a). We also confirm our assumptions of uniform bulk stress and constant radii (Figure S1). Example of simulations are shown in Movie S1. We find that the osmotic pressure is significantly different across the boundaries of condensates (Figure 2b). For liquid 170 condensates, they will quickly dissolve due to the large pressure difference. In contrast, the bulk stress balances the osmotic pressure difference for elastic condensates. Indeed, we find that the  $\Pi - \sigma_B$  field is uniform across the boundaries (Figure 2c).

We note that the uniform  $\Pi - \sigma_B$  is not valid near 176 the condensate boundaries due to the surface tension. Using the variable sizes of condensates, we compute the radius dependence of surface tension constant  $\gamma$ . We find that  $\gamma$  converges to a constant value in the large radius 180 limit (Figure 3a), suggesting that it is well defined in the 181 thermodynamic limit. We find that the surface tension 182 constant is relatively small in our simulations; therefore,  $_{146}$   $G_S$  (Methods). In particular, to simulate elastic conden-  $_{183}$  our predictions of the density  $\phi_{\rm in}$  and  $\phi_{\rm out}$  assuming  $\gamma=$ 



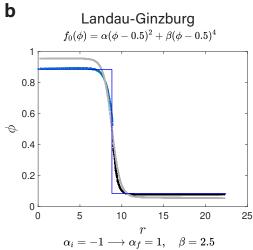


FIG. 4. Simulations using different types of free energy. (a) For the Flory-Huggins free energy, when the control parameter  $\chi$  decreases from 3.0 to 1.5, the initial density field (gray dots) cannot be maintained and the final density field is established (blue dots). The blue line is the theoretical prediction based on the equilibrium conditions of elastic condensates. (b) For the Landau-Ginzburg free energy, the control parameter  $\alpha$  increases from -1 to 1, and the equilibrium density field can also be predicted by our theories. In both (a) and (b), a single condensate is simulated, and  $G_S = 20$ ,  $\phi_c = 0.5$ ,  $R_0 = 9$ .  $G_B = 10$  in (a) and  $G_B = 20$  in (b).

184 0 (Figure 3b) are very close to the predictions with a finite 238 arated into three parts in which both the surface ten-185 \gamma (Figure S2). To test the generality of our theories, 239 sion force and the osmotic force change their signs (see 187 that our theories are equally applicable to both forms of 241 in Figure S9a). For an elastic condensate, the osmotic 188 free energy (Figure 4). Finally, we also test the effects 242 force always points outwards from the condensates since 189 of shear modulus and critical density, and find that our 243 the chemical potential is now a monotonically increasing <sub>190</sub> results are insensitive to the values of  $G_S$  and  $\phi_c$  (Figure <sup>244</sup> function of  $\phi$ . Therefore, in this case, an inward elastic 191 S3-S6), further corroborating our theories.

## Minimum bulk modulus for stable condensates

In the following, we systematically investigate the in-194 dissoluble conditions of elastic condensates. We simulate a condensate with two control parameters  $\chi_f$  and  $G_B$ and monitor its dissolution dynamics after  $\chi$  is reduced from 3 to  $\chi_f$ . We label the condensate as dissoluble or indissoluble depending on if the system becomes uniform or not after a long waiting time  $t = 10^4$  (Figure 5b, c and Movie S2). As expected, when  $G_B = 0$ , the condensate is stable only if  $\chi > 2$ . For  $\chi < 2$ , the condensate becomes indissoluble if the bulk modulus is larger than a critical value  $(G_{B,c})$ . The results are summarized in the phase diagram (Figure 5a).

We also succeed in finding the theoretically predicted phase boundary separating the dissoluble and indissoluble phase. We compute the theoretically predicted  $\phi_{\rm in}$  as a function of  $G_B$  using Eq. (1) with the conservation of total molecular number, and find that  $\phi_{\rm in}$  decreases as  $_{210}$   $G_B$  decreases (Figure S7). Therefore, a minimum bulk modulus should exist to ensure  $\phi_{\rm in} > \theta \phi_c$  so that the elas-212 ticity of the condensate can be maintained. Here,  $\theta$  is a constant presumably close to 1. We plug  $\phi_{\rm in} = \theta \phi_c$  into Eq. (1) (with  $\phi_{\text{out}}$  determined by the conservation of total molecular number) and note that both  $\chi$  and  $G_B$  appear linearly in Eq. (1). Therefore, the phase boundary separating dissoluble and indissoluble phase must be linear in the  $\chi_f$ - $G_B$  parameter space. Intriguingly, we find that the theoretically computed phase boundary nicely matches the simulated phase diagram (Figure 5a) with  $\theta = 1.1$ , slightly larger than 1. Our results are not sen-222 sitive to the values of  $\phi_c$  as we get similar results using 223 different  $\phi_c$  (Figure S8).

To understand why  $\theta \gtrsim 1$ , we remark that for the elas-225 tic condensate to be stable,  $\phi_{\rm in}$  must be larger than  $\phi_c$ . 226 This is the consequence of force balance across the con-227 densate boundary. The polymer network is subject to 228 two types of force: the force from the gradient of the 229 osmotic tensor  $(\nabla \cdot \Pi)$  and the force from the gradient 230 of the elastic stress  $(\nabla \cdot \sigma)$ . We can further decompose 231 the former force into two parts, one is the from the free 232 energy  $f_0(\phi)$ , which we call as the osmotic force in the 233 following, and the other is from the  $\frac{C}{2}(\nabla\phi)^2$  term in the 234 free energy, which we call as the surface tension force in the following. For a liquid condensate, the osmotic force always balances the surface tension force across the 237 condensate boundary. The crossover regime can be sepwe also use the Landau-Ginzburg free energy and find 240 the schematic in Figure 5d and numerical simulations 245 force must exist to balance the sum of surface tension

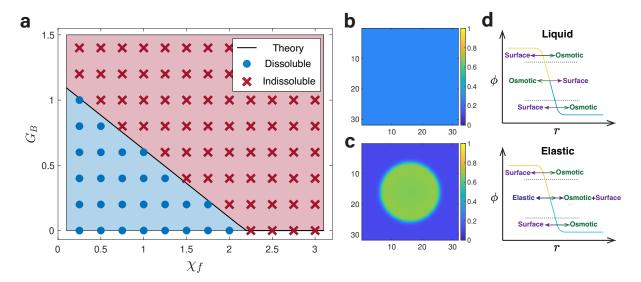


FIG. 5. A critical bulk modulus  $G_{B,c}$  above which condensates are indissoluble. (a) Phase diagram of condensate stability with control parameters  $\chi_f$  and  $G_B$ . The theoretical predicted  $G_{B,c}$  is the black line and the simulation results are the blue dots and red crosses. (b) and (c) The density fields  $\phi$  for dissolvable and indissoluble cases respectively. In (b),  $G_B = 0.4$  and  $\chi_f = 0.5$  and the final density is uniform. In (c),  $G_B = 1.2$  and  $\chi_f = 1.5$  and the condensate is indissoluble due to elasticity. (d) Schematics for the force balance in the crossover regime of condensates.

246 force and osmotic force in the crossover regime (Figure 275 phase diagram of the dissolution conditions of elastic con-<sub>248</sub> than  $\phi_c$  to ensure a finite elastic force in the crossover <sub>277</sub> densates to be stable upon the condition changes, such 249 regime; therefore,  $\theta \gtrsim 1$ .

## Discussion

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251 are largely neglected in theoretical modeling, the elasticity of condensates are crucial to cellular fitness in both positive and negative ways. The conversion of condensates into a solid-like state could help to preserve biological structures and suppress detrimental biochemical <sup>286</sup> reactions [8, 18]. Furthermore, they help to sequester misfolded proteins [22, 30, 31]. On the other hand, the 287 dissolution of solid-like condensates may need assistance by energy-consuming enzymes [16, 32–34], therefore, recondensates are supposed to dissolve such as the dissolution of stress granules after stress, failure to dissolve may lead to aberrant condensates [8, 18], which further trigger cellular aging and even diseases.

Our work provides the first mechanistic understanding on the irreversible nature of aged biomolecular condenside the condensates, and therefore prevents the disso- 295 surface term. 272 lution. Our theoretical predictions are nicely confirmed 296 273 by the numerical simulations using the two-fluid model. 297 action between biomolecules become weaker. In the pres-274 Moreover, we both numerically and theoretically obtain a 298 ence of elasticity, the osmotic pressure difference inside

5d and Figure S9b). In conclusion,  $\phi_{\rm in}$  should be larger 276 densates, and obtain a minimum bulk modulus for con-278 as an increase in temperature. There remain some open 279 questions including the effects of shear modulus and finite 280 relaxation times, and we expect future works to explore 281 these questions. Finally, our results may have implica-282 tions to develop condensate-targeting drugs to specifi-While the solid-like nature of biomolecular condensates <sup>283</sup> cally change condensate properties inside cells, e.g., low-284 ering the bulk modulus to dissolve irreversible conden-

## **METHODS**

#### Derivations of equilibrium conditions

To find the equilibrium condition of elastic condensate, ducing cellular fitness. Moreover, in conditions where 289 we need to find the change in the elastic energy of a 290 condensate due to a small change in displacement field  $\mathbf{u_p} \to \mathbf{u_p} + \delta \mathbf{u_p}$ , which can be generally written as

$$\delta F_{el} = \int -\mathbf{f} \cdot \delta \mathbf{u_p} dV = \int -(\nabla \cdot \sigma) \cdot \delta \mathbf{u_p} dV$$
$$= \int \sigma \nabla \delta \mathbf{u_p} dV - \int \sigma \delta \mathbf{u_p} d\mathbf{S}. \tag{8}$$

sates. We derive the equilibrium conditions of elastic 292 Here f is the force field and  $\sigma$  is the stress. The above condensates and demonstrate that the bulk stress can 293 equation shows that the elastic energy change can be balance the osmotic pressure difference inside and out- 294 generally expressed as the sum of one bulk term and one

Upon active biological regulation, the attractive inter-

 $_{300}$  tic force, which is generated by the deformation field.  $_{341}$   $\dot{F}_{
m mix}$  is calculated through We consider a condensate with spherical symmetry and 302 the initial displacement field due to the osmotic pressure 303 difference is radial with constant  $\nabla \cdot \mathbf{u_p}$ . This gener-304 ates a dilution of density in bulk of condensate such that  $\nabla \cdot \mathbf{u_p} = \log(\phi_1/\phi_{\rm in})$  where  $\phi_1$  ( $\phi_{\rm in}$ ) is the density before 306 (after) the biological regulation. Note that the stress  $_{307}$  field inside the condensate is diagonal and also constant

Now we consider a small change in the volume  $_{310}$  of the condensate without change in the number of 311 biomolecules. This leads to a change in the density:  $\delta \phi / \phi = -\delta V / V$ . The resulting change in the displacement field therefore satisfies  $\nabla \cdot \delta \mathbf{u}_{\mathbf{p}} = -\delta \phi/\phi = \delta V/V$ . 314 Using the spherical symmetry, we obtain  $\delta \mathbf{u}_{\mathbf{p}}(r) =$ <sub>315</sub>  $r\delta V/(V\cdot d)\mathbf{e_r}$  where d is the spatial dimension and  $\mathbf{e_r}$ 316 is the unit radial vector. Therefore, near the surface, 317 the change in the bulk displacement field  $\delta \mathbf{u}_{\mathbf{p}}(r \to R) =$ 318  $R\delta V/(V\cdot d)\mathbf{e_r}$ . Interestingly,  $\delta \mathbf{u_p}(r\to R)$  is just the dis- $_{319}$  placement change needed to increase the volume by  $\delta V$ .  $_{348}$  and all components mentioned above, the Rayleighian of  $_{320}$  This means that the displacement change on the surface  $_{349}$  the solution is therefore: 321 is zero, and only the bulk term contributes to the elas-322 tic energy change. Therefore, the change in the elastic 323 energy becomes

$$\delta F_{el} = \sigma_B \delta V. \tag{9}$$

 $_{324}$  Since the free energy changes due to the osmotic pressure 325 and surface tension are the same as the usual liquid con-326 densate, the equilibrium condition due to volume change 327 becomes

$$\Pi_{\rm in} - \sigma_B = \Pi_{\rm out} + \frac{(d-1)\gamma}{R}.$$
 (10)

## Details of the two-fluid model

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Numerical simulations of viscoelastic phase separation 330 are based on the two-fluid model [24, 25], which consid- $_{331}$  ers the dynamics of polymer velocity  $\mathbf{v}_{\mathbf{p}}$ , solvent velocity 332  $\mathbf{v_s}$  and the average velocity  $\mathbf{v} = \phi \mathbf{v_p} + (1 - \phi) \mathbf{v_s}$ . This  $^{333}$  model is derived by minimizing the Rayleighian R of the 334 solution [24, 25, 35], the sum of the energy dissipation  $\Phi$  and the temporal changing rate of the free 336 energy  $\dot{F}$ . The energy dissipation function  $\Phi$  consists 337 of two parts, which are respectively the friction between 338 polymer and solvent  $\Phi_1$ , and the overall viscous dissipa-339 tion of the solution  $\Phi_2$ 

$$\Phi_1 = \int d\mathbf{r} \, \frac{\zeta}{2} (\mathbf{v_p} - \mathbf{v_s})^2 = \int d\mathbf{r} \, \frac{1}{2} \zeta \frac{(\mathbf{v_p} - \mathbf{v})^2}{(1 - \phi)^2}, \quad (11)$$

$$\Phi_2 = \int d\mathbf{r} \, \frac{\eta}{4} (\nabla \mathbf{v} + (\nabla \mathbf{v})^T) : (\nabla \mathbf{v} + (\nabla \mathbf{v})^T).$$
 (12)

299 and outside the condensate can be balanced by the elas- 340 The temporal changing rate of the mixing free energy

$$\dot{F}_{\text{mix}} = \int d\mathbf{r} \, \dot{\phi} f'(\phi) = \int d\mathbf{r} \left[ -\nabla \cdot (\phi \mathbf{v}_{\mathbf{p}}) \right] f'(\phi)$$
$$= \int d\mathbf{r} \left[ \phi \nabla f'(\phi) \right] \cdot \mathbf{v}_{\mathbf{p}} = \int d\mathbf{r} \left( \nabla \cdot \mathbf{\Pi} \right) \cdot \mathbf{v}_{\mathbf{p}}, \quad (13)$$

342 involving the continuous equation of the density:  $\dot{\phi} =$  $_{343}$   $-\nabla \cdot (\phi \mathbf{v_p})$ . Here  $f(\phi) = f_0(\phi) + \frac{C}{2}(\nabla \phi)^2$  and C is a 344 constant. The elastic energy comes from the polymer, so 345 its temporal changing rate is

$$\dot{F}_{el} = \int d\mathbf{r} \,\sigma_{ij}\partial_j v_{p_i} = \int d\mathbf{r} \,(-\nabla \cdot \sigma) \cdot \mathbf{v_p}. \tag{14}$$

346 Combined with the constrain from the incompressible

$$\nabla \cdot \mathbf{v} = 0, \tag{15}$$

$$R = \int d\mathbf{r} \left[ -p(\nabla \cdot \mathbf{v}) + \frac{\zeta}{2} \frac{(\mathbf{v}_{\mathbf{p}} - \mathbf{v})^{2}}{(1 - \phi)^{2}} + \frac{\eta}{4} (\nabla \mathbf{v} + (\nabla \mathbf{v})^{T}) : (\nabla \mathbf{v} + (\nabla \mathbf{v})^{T}) + (\nabla \cdot \mathbf{\Pi}) \cdot \mathbf{v}_{\mathbf{p}} - (\nabla \cdot \sigma) \cdot \mathbf{v}_{\mathbf{p}} \right].$$
(16)

 $_{350}$  By setting the functional derivative of R with  $\mathbf{v_p}$  and  $\mathbf{v}$ 351 to be 0, we obtain the following equations:

$$\frac{\zeta}{(1-\phi)^2}(\mathbf{v_p} - \mathbf{v}) + \nabla \cdot \mathbf{\Pi} - \nabla \cdot \sigma = 0, \quad (17)$$

$$\nabla p - \frac{\zeta}{(1-\phi)^2} (\mathbf{v_p} - \mathbf{v}) - \eta \nabla^2 \mathbf{v} = 0.$$
 (18)

352 Finally, we rewrite the above equations and obtain

$$\frac{\partial \phi}{\partial t} = -\nabla \cdot (\phi \mathbf{v_p}),\tag{19}$$

$$\mathbf{v_p} - \mathbf{v} = -\frac{(1 - \phi)^2}{\zeta} (\nabla \cdot \mathbf{\Pi} - \nabla \cdot \sigma), \tag{20}$$

$$-\nabla \cdot \mathbf{\Pi} + \nabla \cdot \sigma - \nabla p + \eta \nabla^2 \mathbf{v} = 0, \tag{21}$$

353 Clearly, Eq. (5) is obtained from Eq. (19) and Eq. (20). 354 Combined with the incompressible condition  $\nabla \cdot \mathbf{v} = 0$ 355 and Eq. (21), the average velocity  $\mathbf{v}$  is calculated as

$$\mathbf{v}(\mathbf{r}) = \int d\mathbf{r}' \, \mathbf{T}(\mathbf{r} - \mathbf{r}') \cdot (-\nabla \cdot \mathbf{\Pi}(\mathbf{r}') + \nabla \cdot \sigma(\mathbf{r}')), \quad (22)$$

while  $\mathbf{T}(\mathbf{k}) = \frac{1}{\eta |\mathbf{k}|^2} (\mathbf{I} - \frac{\mathbf{k}\mathbf{k}}{|\mathbf{k}|^2})$  is the Oseen tensor in the (11) 357 Fourier space. We can then obtain the polymer veloc-358 ity  $\mathbf{v_p}$  with equation (20). Therefore, the density  $\phi$  in  $_{359}$  simulation can be updated by calculating  $\mathbf{v_p}$  when the osmotic pressure  $\Pi$  and the stress  $\sigma$  are known.

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The two components in stress tensor  $\sigma$ , which are shear <sup>395</sup> stress tensor  $\sigma_S$  and bulk stress scalar  $\sigma_B$ , respectively <sup>396</sup> obey the following Maxwell-type equations [25]: <sup>397</sup>

$$\frac{\partial \sigma_S}{\partial t} = -(\mathbf{v_p} \cdot \nabla)\sigma_S + \sigma_S \cdot \nabla \mathbf{v_p} + (\nabla \mathbf{v_p})^T \cdot \sigma_S 
- \frac{1}{\tau_S(\phi)}\sigma_S + G_S(\phi)(\nabla \mathbf{v_p} + (\nabla \mathbf{v_p})^T),$$
(23)

$$\frac{\partial \sigma_B}{\partial t} = -\left(\mathbf{v_p} \cdot \nabla\right)\sigma_B - \frac{1}{\tau_B(\phi)}\sigma_B + G_B(\phi)\nabla \cdot \mathbf{v_p}. \quad (24)$$

 $_{364}$  In our simulations, we first do not include elastic stress  $_{365}$  to form condensates by taking  $G_S=G_B=0$ . We then  $_{366}$  introduce the elastic stress by taking

$$G_S(\phi) = G_S \phi^2, \tag{25}$$

$$\tau_S(\phi) = \frac{1}{\phi_c - \phi} \Theta(\phi_c - \phi), \tag{26}$$

$$G_B(\phi) = G_B\Theta(\phi - \phi_c), \tag{27}$$

$$\tau_B(\phi) = \frac{1}{\phi_c - \phi} \Theta(\phi_c - \phi). \tag{28}$$

 $_{367}$  Here  $G_B$  and  $G_S$  are constants. We assume a critical  $_{420}$   $_{368}$  density  $\phi_c$  above which the polymer network is percolated  $_{421}$   $_{369}$  and becomes fully elastic with a finite bulk modulus and  $_{422}$   $_{370}$  a diverging relaxation time.

### Details of numerical simulations

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We perform numerical simulations in a 2D grid by solv-373 ing the two-fluid model using the explicit Euler method 430 with the periodic boundary condition on MATLAB. Sim- 431 ulations for a single condensate are in a  $127 \times 127$ grid, and simulations for multiple condensates are in a  $255 \times 255$  grid. The grid size is  $\Delta L = 0.25$  by setting the unit length equal to the lattice size when de-379 riving the mixing free energy. The time interval for the 437 380 simulation is  $\Delta t = 0.001$ . In our simulations, we take 438  $_{381}$   $k_BT=C=\xi=\eta=1$  for simplicity. The elastic  $_{439}$ 382 stress is introduced at  $t=10^3$  for a single condensate 440 and  $t = 4 \times 10^3$  for multiple condensates. The control parameter  $\chi$  is changed 50 time units after adding the 443 385 elasticity. Eq. (22) is solved with fast Fourier transfor-386 mation and other equations are calculated in real space. 445 387 For the simulation of multiple condensates, we initially 446  $_{388}$  add a Gaussian noise with variance 0.001 to the uniform  $_{447}$ 389 density field. In Figure 5a, simulations are initiated with 390 a single condensate with  $R_0 = 9$ . The condensate is con-391 sidered dissoluble if the variance of the  $\phi$  field at  $t=10^4$  $_{392}$  is less than 0.01.

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