

1 **Plasma FIB milling for the determination of structures *in situ***

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17 **Abstract**

18 Structural biology inside cells and tissues requires methods able to thin vitrified specimens to  
19 electron transparent thicknesses. Until now, focused ions beams based on gallium have been  
20 used. However, ion implantation, changes to surface chemistry and an inability to access high  
21 currents limit Gallium as an ion beam source. Here, we show that plasma-coupled ion sources  
22 can produce cryogenic lamella of vitrified human cells in a robust and automated manner,  
23 with quality sufficient for pseudo-atomic structure determination. In addition, these lamellae  
24 were produced in a prototype microscope equipped for long cryogenic run times (>1 week)  
25 and with multi-specimen support fully compatible with modern-day transmission electron  
26 microscopes. We demonstrate for the first time that plasma ion sources can be used for  
27 structural biology within cells, determining a structure in-situ to 4.9 Å and describing a  
28 workflow upon which different plasmas can be examined to streamline lamella fabrication  
29 further.

## 30 Introduction

31 Cryo-electron tomography (cryoET) enables the structural study of macromolecular  
32 complexes within their native cellular environment. Understanding the structural landscape  
33 of macromolecules in relation to their subcellular environment is key in linking structure to  
34 function. Since most biological samples are too thick to directly image with a transmission  
35 electron microscope (TEM), methods to thin down biological material to the < 300 nm  
36 necessary for electron transparency have been instrumental for enabling cryoET of the  
37 internal regions of cells. Focused ion beam (FIB) milling was originally developed for material  
38 science applications, such as microchip manufacture, at room temperature, where its ability  
39 to precisely shape objects on the nanoscale revolutionised electronics<sup>1,2</sup>. More recently, the  
40 approach has been adapted for use with biological samples through the adaptation of existing  
41 dual beam FIB/scanning electron microscopes (FIB/SEM) with cryogenic stages enabling  
42 frozen-hydrated samples to be shaped and thinned to thicknesses enabling subsequent  
43 cryoET analyses<sup>3,4</sup>. This capability has enabled intracellular structural studies on a wide variety  
44 of biological samples and to pseudo-atomic resolution<sup>5-9</sup>.

45  
46 Current FIB/SEM instruments used to prepare FIB lamella are not optimised for cryogenic  
47 applications<sup>10-12</sup>, reducing their throughput. The large sample chamber (~40 L), originally  
48 designed for 8" Si wafers at room temperature, has a vacuum typically of the order 10<sup>-6</sup> mbar,  
49 dropping by a ~10-fold when cold. This results in ice (re)deposition, with rates of up to 85  
50 nm/h reported<sup>11,13</sup>. This limits number of lamellae that can be prepared on one grid in an  
51 experiment and impacts the proportion of successful lamella<sup>11</sup>. Studies have recently  
52 demonstrated success with automated FIB-lamella fabrication<sup>11,12,14,15</sup>, but use of gas cooled  
53 systems limits length of overnight runs and ice growth effectively limits the number of lamella  
54 that can be prepared in one automated session.

55  
56 Lamella fabrication to-date for structural biology has typically been performed with a gallium  
57 FIB. Gallium works well as a metal ion source due its low melting point, low volatility, low  
58 vapour pressure, good emission characteristics and good vacuum properties<sup>16</sup>. Gallium also  
59 sputters a range of different materials efficiently. The point source nature of a gallium ion  
60 source enables the beam to be tightly focussed<sup>17</sup> enabling control at the tens to hundreds of  
61 nanometer scale. At higher beam currents the beam becomes increasingly divergent, limiting  
62 the bulk milling these beams are capable of. Implantation is likely from any focussed ion  
63 beam<sup>18</sup>, but gallium has been shown to react with various samples, migrate and significantly  
64 alter the original structure<sup>18-20</sup>. Alternative approaches are therefore of interest to reduce  
65 these effects.

66  
67 An alternative is inductively coupled plasmas generated from a gas, e.g., nitrogen, oxygen,  
68 xenon, or argon. These plasma sources are comparable in size to gallium at low current but  
69 have smaller probe sizes than gallium at larger beam currents<sup>21</sup>. This means that in principle  
70 plasma beams are highly suited to milling large volumes but remain capable of the fine milling  
71 required for thinner, more fragile, lamella used in life sciences<sup>22</sup>.

72  
73 Here, we describe a protocol for automated plasma FIB milling lamella of cryogenic cellular  
74 samples. We were able to hold multiple specimens at cryogenic temperatures at low  
75 contamination rates (<2 nm / hr) for weeks. We characterised the milling rates of plasma ion  
76 sources (O, N, Ar, Xe) at a range of currents. We demonstrate that an argon plasma source

77 can produce lamellae with a high success rate (~80%). The lamellae were used to generate a  
78 cryoET dataset of hundreds of tomograms, which show thicknesses, features, and resolution  
79 on a sub-nanometre scale, enabling sub volume averaging of the human 80S ribosome to a  
80 resolution of ~4.9 Å, with the well-ordered regions of the structure at resolutions close to the  
81 Nyquist limit of 3.8 Å. Our study demonstrates for the first time the use of a plasma ion source  
82 to enable structural biology *in situ* and paves a way for higher throughput.

## 83 Results

84

### 85 **A plasma FIB/SEM for screening and imaging multiple samples with minimal contamination** 86 **over long time scales**

87 We used a custom designed plasma FIB/SEM microscope (Supplementary Fig. 1) equipped  
88 with a coincident FIB and SEM inside a chamber with redeposition rates < 2 nm/hr  
89 (Supplementary Fig. 2), and a stage with rotational freedom of +14° to -190°. The sample  
90 chamber vacuum is maintained at a pressure of  $\sim 1 \times 10^{-7}$  mbar. The redesigned sample  
91 chamber is approximately 6 litres in volume. Up to 12 clipped grids can be mounted into a  
92 multi-specimen cassette which can be robotically loaded into the chamber, one at a time  
93 (Supplementary Fig. 1). The PFIB can be configured to be used with different ion sources,  
94 including xenon, oxygen, argon. The cryo box (anti-contaminator) and stage are braided cooled  
95 from liquid nitrogen dewars that are filled automatically when needed, allowing continuous  
96 cold runtimes of up to a week or more. The characteristics of this microscope, including  
97 minimal ice growth (Supplementary Fig. 2), robotic sample transfers and long cryo runtimes  
98 enabled preparation of FIB-lamellae on biological specimens for up to one week. In effect the  
99 number of lamellae that can be produced is sample dependent. The high-vacuum exchange  
100 of samples and robotic sample entry meant that samples could be handled with minimal  
101 damage or disruption.

102

### 103 **Milling rates of plasma ions on vitreous cellular samples**

104 Determining the sputter rate of each plasma beam can inform the design of milling protocols.  
105 Others have already characterised the sputter rate for different materials<sup>23</sup>, but not on  
106 vitrified biological samples. We used a second microscope (Thermo Scientific Helios Hydra  
107 with gas-cooled cryogenic stage) with the same ion column as the first to determine sputter  
108 rates more accurately for xenon, nitrogen, oxygen, and argon, at normal incident angles.  
109 Based on 3 experimental repeats on different regions of the same plunge frozen yeast sample,  
110 our measurements show that for vitrified biological samples xenon has the highest sputter  
111 rate of  $16.8 \pm 0.2 \mu\text{m}^3/\text{nC}$ . Nitrogen and oxygen have values of  $10.6 \pm 0.2$  and  $10.0 \pm 0.4$   
112  $\mu\text{m}^3/\text{nC}$  respectively, while argon is  $4.3 \pm 0.21 \mu\text{m}^3/\text{nC}$  (Supplementary Fig. 3; Table 1). This  
113 compares with a sputter rate of  $7.7 \mu\text{m}^3/\text{nC}$  for gallium in ice<sup>24</sup>. These sputter rates are  $\sim 20$ -  
114  $100$  times higher than those reported at 30kV in Silicon for these plasma beams<sup>23</sup>. This is of  
115 the same order as the factor of ten that is suggested within the community as a rule of thumb  
116 for the difference between milling in vitrified ice and silicon<sup>25</sup>. During lamella fabrication for  
117 cryoET, typical FIB angles for milling can range between 8 – 13°, with angles up to  $\sim 25^\circ$  being  
118 used for some applications. We measured the sputter rate of vitrified samples using xenon  
119 and argon at currents between 20 pA up to 2 nA (currents typically used for lamella  
120 fabrication<sup>13,26</sup> at angles from 10° to 40°, and calculated sputter rates. (Supplementary Fig. 3;  
121 Table 2)). Our results show that at high angles, there was a greater difference in sputter rates  
122 between the two gases (at 90° xenon is 3.7x greater) compared to shallow angles (10° xenon  
123 is 2.7x greater), suggesting that the incidence angle influences sputter efficiency in vitrified  
124 water, and that this dependence varies with milling gas.

125

### 126 **Plasma is a suitable ion source for automated, high throughput lamella fabrication**

127 While xenon has the greater sputter rate, we opted to use argon for our first experiment of  
128 fully automated lamella fabrication. This is because we had accumulated more experience in  
129 the control of argon plasma to produce smooth flat surfaces. We reasoned that a smoother

130 surface would correlate with lower damage and therefore a better initial test of the approach.  
131 Since the instrument is not limited by ice redeposition in the chamber for these initial  
132 experiments the additional time taken by argon was not critical. Over two days 34 lamella  
133 were milled automatically from two grids (Fig. 1). The average time taken for milling was  
134 approximately 45 mins per lamella; 32 minutes of coarse milling and 13 minutes of fine milling  
135 (details in Table 3).

136  
137 Once completed the grids were transferred to a compatible TEM in the same cassette,  
138 preserving the milling angle close to perpendicular to the tilt axis (robotic automation still  
139 leads to some slight rotational variability upon loading), eliminating need for manual  
140 handling. No lamellae were lost during sample transfer between PFIB and TEM.

141  
142 From the 34 positions initially prepared, 26 lamellae were imaged in the TEM (i.e., were thin  
143 enough to enable transmission electron imaging), giving a successful fabrication rate of 76%  
144 (Fig. 1e). Of those, 17 were deemed suitable for tilt series acquisition (excluding lamella with  
145 incomplete vitrification, cracks, or unsuitable acquisition areas (e.g nucleus)). Therefore, of  
146 the lamella produced 65% (50% of initial set-up positions) were able to produce data from  
147 electron cryo-tomography; a total of 180 tilt-series.

148

#### 149 **Electron cryo-tomography of plasma FIB milled lamella shows vitreous ice and allows an** 150 **assessment of cellular morphology and ultrastructure**

151 Tomograms acquired from the lamellae generally exhibited characteristics associated with  
152 vitrified cellular tomograms (no Bragg reflections within images, no aggregation-segregation  
153 induced contrast, uniform rounded membranes). We could observe a wide variety of  
154 macromolecular complexes (Fig. 1f-h, Supplementary Video 1) in the reconstructed  
155 tomograms including nuclear pore complexes<sup>27,28</sup>, microtubules<sup>29</sup>, vault complexes<sup>30</sup>, and  
156 vesicle budding via retromer coat proteins<sup>31</sup>. The reconstructed tomograms had an overall  
157 mean thickness of 250 nm  $\pm$  70 nm for the recorded tilt-series (Fig. 2). Although manually  
158 prepared FIB-lamellae are generally thinner<sup>32</sup>, the lamellae generated here were suitable for  
159 successful TEM tilt-series acquisition, observing cellular features in tomograms > 350 nm thick  
160 (Fig. 2).

161

162 We measured the thickness of all lamellae and the distance from the leading edge at which  
163 the tomogram was acquired (Fig. 2; Supplementary Fig. 4). Interestingly, a number of lamella  
164 showed a gradient of thickness<sup>13</sup>, with some lamella being twice as thick at the back of the  
165 lamella as toward the front (<200 nm vs >350 nm), while others were relatively flat. The  
166 lamella with the greatest thickness gradient were the longest (Supplementary Fig. 4). There  
167 is a need to optimise the protocol to best reflect the cellular size/morphology to achieve flat  
168 lamella. In the current test case, no over tilt was used in coarse or fine milling steps to attempt  
169 to optimise flatness<sup>13</sup>.

170

#### 171 **Sub-volume averaging at sub-nm and the effect of particles at the lamella edge**

172 We determined structures of the human ribosome within the HeLa cells. We could obtain a  
173 structure of the human 80S ribosome using 15628 particles from 17 lamellae with a global  
174 resolution of 4.9 Å, and local resolution of 3.8 Å for large areas of the 60S subunit (Fig. 3,  
175 Supplementary Fig. 5, Supplementary Movie 2 and 3). During refinement of the structure to  
176 sub-nm resolutions, the 60S and 40S diverged in resolution, with the 60S approaching a

177 resolution close to the Nyquist information limit (3.8 Å), while the 40S was a resolution above  
178 8 Å. As the particle population is taken from non-arrested HeLa cells, our data captures the  
179 conformational states of the ribosome during translation. These data are available on EMPIAR  
180 (EMPIAR-XXXXX) should a study wish to probe human ribosome dynamics further.

181

182 We interrogated the effect of plasma on damage propensity close to the milling boundary of  
183 the lamella. Currently it is difficult to ascertain the damage induced by an ion beam during  
184 lamella fabrication for life science samples; diffraction methods are not possible on cellular  
185 samples and sub-volume averaging has, until recently<sup>8,33</sup>, been limited to low resolution. The  
186 ability to produce structures at sub-nm resolutions offers the opportunity to determine  
187 damage due to ion beam milling (based on ability to align particles to a given resolution).  
188 Boundary models for all 180 tomograms were produced and then interpolated to form a  
189 geometric model of the tomograms. This enabled robust modelling of the lamella geometry  
190 local to each tomogram (Supplementary Fig. 6). When we compared the number of particles  
191 3D classification with the number after picking, there was an appreciable loss of particles  
192 within 20 nm from the lamella edge (Supplementary Fig. 7). The particle positions taken from  
193 sub-volume averaging of 3D classified 18,119 particles were then used to calculate the  
194 distance at which each ribosome was located relative to the boundary of the  
195 lamella/tomogram. Across the 180 tomograms, 2099 ribosomes were calculated to be within  
196 30 nm of the lamella edge (mean tomogram thickness  $190 \pm 39$  nm). 2099 particles outside  
197 of this 30 nm layer were then randomly taken as a comparison (mean thickness  $216 \pm 49$  nm).  
198 Each set of particles were then independently subjected to sub-volume averaging using a new  
199 global 3D refinement. Interestingly, particles taken from the 30 nm layer refined to a  
200 resolution of 9.2 Å compared to a resolution of 8.0 Å for the ribosomes outside of that  
201 distance (Fig. 4; Supplementary Fig. 8). The B-factors<sup>34</sup> for the two particle populations also  
202 suggest particles within the 30 nm are of lower reconstruction quality, with a 24 % difference  
203 between the two particle populations ( $628 \text{ \AA}^2$  vs  $480 \text{ \AA}^2$ ) (Fig.4).

## 204 Discussion

205 The determined sputter rates for each plasma (Table 2; Supplementary Fig. 3) are non-linear  
206 with molecular weight (MW) of the used gas. While xenon (MW 131.29) has the greatest  
207 sputter rate and largest MW, argon (MW 39.95) has the lowest sputter rate while weighing  
208 more than nitrogen (MW 14) and oxygen (MW 15.99). The sputter rates for oxygen and  
209 nitrogen are the same within error. While there is an expectation that a greater MW should  
210 correlate with sputter rate from first logic, the sputter rate for different plasmas has been  
211 demonstrated previously to be material dependent<sup>23</sup>. In our case, the values are calculated  
212 for (vitreous) biological samples. Interestingly, the angle dependence we were able to  
213 calculate for argon and xenon shows that closer to the glancing angle (smaller milling angles)  
214 the sputter rates increase greatly. However, more surprisingly the sputter yield for argon  
215 becomes a much greater relative yield with respect to xenon, with the relative sputter rates  
216 changing from 4.5  $\mu\text{m}^3/\text{nC}$  to 59.3  $\mu\text{m}^3/\text{nC}$  from 90° incidence to 10° incidence, suggesting a  
217 high angular dependence on sputter rates. Interestingly, this effect is greater for argon than  
218 for xenon (13-fold increase for argon vs 9.4-fold increase for xenon). These observations are  
219 consistent with published simulations<sup>35</sup> but mean that at very shallow angles the choice of  
220 gas becomes less critical (between Xe and Ar). Other contributory factors, such as curtaining  
221 propensity (either based on probe characteristic or defocus variation) should therefore be  
222 considered to a greater extent at these angles.

223  
224 The mean free path of an electron at 300 kV in water is approximately 300 nm<sup>36,37</sup>. This limits  
225 the thickness of samples that can be imaged by electrons. Traditionally lamella below 300 nm  
226 are prepared to achieve sufficient contrast for cryoET; a simple rule of thumb is that the  
227 thinner the lamella the better the contrast. Although higher in contrast, such thin lamellae (<  
228 150 nm) suffer from having a greater proportion of the volume subject to milling induced  
229 effects and images a smaller volume of biological space reducing contextual information. An  
230 open question is whether plasma FIB can be used autonomously to fabricate lamella suitable  
231 for *in situ* structural biology. We report a protocol that produces lamella with a mean  
232 thickness of  $250 \pm 70$  nm that runs unsupervised. We show that the lamella produced in this  
233 way yield information rich tomograms. From these tomograms it was possible to use sub  
234 tomographic averaging to obtain a structure of the human ribosome to 4.9 Å within lamellae.  
235 This compares extremely well with the current contemporary studies in cryo-tomography;  
236 ribosomes in isolated bacteria to 3.5 Å<sup>33</sup>, ribosomes from lamella of *S. Cerevisiae* (Yeast) to a  
237 resolution of 5.1 Å<sup>38</sup> and recently lamella of isolated vertebrate myofibrils where the thin  
238 filament was determined to 4.5 Å resolution<sup>8</sup>. The milling process was complete in 2 days and  
239 the TEM data acquisition in 4 days. Thus, at a practical level, the protocol opens a route to  
240 rapid *in situ* structural biology with resolutions at or exceeding the highest currently available.

241  
242 We can only speculate why such high-quality data were obtained from lamella that would be  
243 regarded as “thick” (i.e. in comparison to a gallium milled example of ~185 nm<sup>32</sup>). Any  
244 implantation of the milling ion into lamellae during fabrication will increase the proportion of  
245 inelastic vs elastic scattering reducing image contrast. Argon has over 1/3 less electrons than  
246 gallium thus for an equivalent level of implantation argon would be less distorting due to  
247 lower local change in potential derived from the implanted ion species. Accurate  
248 measurement of gallium and argon implantation upon vitrified samples requires scanning  
249 electron methods and/or electron energy loss spectrometry that are beyond the scope of this  
250 work. Notably both these approaches require very large electron fluences (2000-3000 e<sup>-</sup>/Å<sup>2</sup>)



251 which would severely damage the sample. Secondly, the very high vacuum in the both the  
252 FIB/SEM and the TEM instrument used here have practically eliminated redeposited ice  
253 contamination in the chamber. Thus, the milled lamella thickness that we fabricate is virtually  
254 the same thickness of the lamella that is measured during the TEM experiment. This is in  
255 contrast to other approaches where the lamella that is measured by TEM is often the milled  
256 thickness plus a layer of (contrast reducing) redeposited material<sup>11</sup>. The transfer between the  
257 autoloading devices on the machines employs a portable cassette filled with liquid nitrogen  
258 at atmospheric pressure, some ice contamination was seen on some of our early lamella (Fig.  
259 1). Subsequently using careful precautions, (low humidity, heating and drying) of all  
260 equipment and freshly (< 2 hrs) dispensed liquid nitrogen we have been able greatly reduce  
261 ice contamination. Vacuum transfer between microscopes would significantly simplify the  
262 process, eliminate ice contamination, and allow a more flexible workflow.

263  
264 A robust determination on the effect of FIB milling on the damage caused to the surface of  
265 the lamella (structural changes (amorphisation if lattice) rather than vitrification) is yet to be  
266 reported. Ribosomes populate much of the cell making them an attractive model to assess  
267 the effect of milling. We were able to show via sub-volume averaging, that particles near the  
268 lamella edge contain less high-spatial frequency information compared to particles further  
269 away from the edge. In the 30 nm closest to the top or bottom surface of the lamella,  
270 averaging the ribosomes yields the resolution of 9.2 Å vs 8.0 Å for those particles found in the  
271 middle of the lamella. In addition, the particles producing the higher resolution structure were  
272 taken from tomograms with a greater mean thickness (190 nm vs 216 nm); this suggest  
273 particles in the central region of lamellae are more useful for sub-volume averaging despite  
274 the reduction in signal-to-noise due to this greater thickness. This was also evident from the  
275 calculated B-factors. Taken together, these data suggest that argon plasma, even at low  
276 currents, leads to some structural change within the first 30 nm of the lamella edge. Since the  
277 diameter of the ribosome is 25 Å, a portion of the damage could result from partial ablation  
278 of the ribosome or ribosomes immediately at the surface. Nevertheless, this is the first  
279 empirical measure of (plasma) beam damage for biological samples, and this analysis  
280 establishes a standard that can be used by us and others to robustly evaluate other milling  
281 protocols and plasma chemistries.

## 282 283 **Conclusions**

284 Electron cryo-tomography is the frontier of structural biology but there is a pressing need to  
285 streamline workflows to enable its widespread adoption. Here we present a robust method  
286 for lamella fabrication using plasma FIB. This is the first demonstration of plasma for usage in  
287 lamella fabrication for cryoET. Our results show that a plasma cryo FIB/SEM with low  
288 contamination rates and with a robotic multi-specimen loading device can streamline the  
289 fabrication of lamella for pseudo-atomic structure determination. In our first attempt using  
290 argon plasma, we produced around 20 high quality lamella per day. We have accurately  
291 quantified the depth of damage from plasma milling with argon but show this does not  
292 prevent pseudo atomic resolution structure determination. We conclude that plasma FIB  
293 lamella fabrication is a suitable route for high throughput *in situ* structural biology.

## 294 **Materials and Methods**

295

### 296 **Dual Beam FIB/SEM Descriptions**

297 Milled samples and plasma characteristics data were acquired on either i.) a dual-beam  
298 focused ion beam scanning electron (FIB/SEM) “Helios Hydra” microscope (ThermoFisher  
299 Scientific, Oregon, USA) equipped with a cryogenic stage and plasma multi-ion source (argon,  
300 nitrogen, xenon, and oxygen) or ii.) a dual-beam focused ion beam scanning electron  
301 microscope with redesigned sample chamber and loading mechanism (ThermoFisher  
302 Scientific, Oregon, USA). Briefly, this microscope was equipped with several modifications to  
303 enable plasma FIB milling within a small, enclosed, (ultra) high vacuum chamber.

304

305 *Chamber/Autoloader/Stage:* The chamber volume of the system is much smaller than the  
306 Helios Hydra, with a reduction in volume from approximately  $\sim 0.04 \text{ m}^3$  to  $< 0.01 \text{ m}^3$  (less than  
307 a 5th of the volume of a conventional Helios-type chamber). This enables a vacuum on the  
308 order of  $1 \times 10^{-7}$  mbar to be achieved. It includes ports for the plasma FIB and SEM columns,  
309 cryogenic stage, robotic multi-specimen entry/exit (Autoloader), charge dissipation via a  
310 platinum metal sputter target and organometallic platinum protective coating chemistry via  
311 a gas injection system (GIS). Its volume reduces the contamination rates of a conventional  
312 chamber to  $< 2 \text{ nm/hr}$ . The chamber also comprises of a stage which allows loading of grids  
313 assembled into compatible mounts (autogrids) via the Autoloader. The autogrids are also  
314 compatible with the Titan Krios for imaging via TEM and can be shuttled between the  
315 instruments via a cassette without need for manual handling. Up to 12 samples may be stored  
316 and recalled from the autoloader to the main stage and then subsequently transferred to the  
317 TEM. The stage is enclosed with shielding which enables a clean working environment within  
318 the instrument needed to support long automation runs. The system is cooled via an Autofill  
319 system enabling long run times and unattended operation over days.

320

321 *Electron Column:* The electron optical column is a NiCol column (ThermoFisher). The SEM  
322 comprises of a field emission gun assembly with Schottky-emitter, with a dual objective with  
323 both field-free magnetic and electrostatic lenses. The electron column provides detection in-  
324 lens using back scatter and secondary electrons.

325

326 *Plasma ion column:* The PFIB column can switch between three ion species (xenon, argon, and  
327 oxygen) switchable plasma ion source. These species have both chemical and physical  
328 differences which allow for a range of milling effects.

329

330 *Sputter Target:* An ion sputtering device comprising a platinum mass is used to create thin  
331 layers of conductive platinum. Here, the primary ion source is used at high current to remove  
332 atoms of metal from a target allowing a dense vapor to flow over the sample. This renders  
333 samples conductive for imaging via SEM.

334

335 *Gas injection system:* Organoplatinum is an organometallic compound which is used as a  
336 protective cap to protect the leading edge of the sample during milling. It enables lamella of  
337 sensitive samples to be protected by masking stray ions from affecting the delicate samples.  
338 This is typically condensed on the surface of the sample around a few micrometres thick as  
339 part of the preparation flow. The compound is evolved from a crucible contained within a  
340 delivery mechanism which generates vapor and delivers it via a needle to the chamber in

341 small aliquots. The low temperature of the sample traps the vapour on contact and this  
342 aggregates over time to form the layer.

343

### 344 **Sputter rate measurements**

345 The sputter rate was determined for each aperture commonly used during lamella  
346 preparation and on each beam by milling cross-sections with a controlled dose. Rates were  
347 determined at 30 kV.

348

349 The sample was plunge frozen yeast on a grid. To give sufficient sample thickness to mill  
350 ramps, cross-sections were generally made on the thickest regions of yeast clumps.

351

352 A schematic of the method for sputter rate determination is shown in Supplementary Fig. 3a.  
353 It is well established that FIB milling is sensitive to surface topography<sup>19</sup>. To reduce these  
354 effects the sample was first coated in GIS and platinum, as is the case for lamella preparation.  
355 A ramp was then milled into this to produce a smoother sample surface. The stage was then  
356 rotated and tilted so that now smooth sample surface of the ramp is presented perpendicular  
357 to the FIB column for cross-sectional milling. Each condition was repeated three times and an  
358 average taken.

359

360 For the angular dependence study, plunge frozen HeLa cells were used. In this case the stage  
361 was tilted after milling the ramps to introduce the required milling angle.

362

363 The milled depth was measured using the SEM. For perpendicular milling, the SEM imaging  
364 occurs at an angle relative to the trench (52°). The perpendicular depth was therefore  
365 calculated using the angle of imaging (i.e., tilt correction). The milled cuboid volume was  
366 calculated using this perpendicular depth and the known milling area, as shown in Equation  
367 1.

368

369 For milling at non-normal angles, the milled volume was assumed to be a parallelepiped (side  
370 profile of this is shown in Supplementary Fig. 3e) with two rectangular and two square faces;  
371 the volume in this case is given in Equation 2. The volume was calculated by first calculating  
372 the area of the parallelogram (shown in yellow in Supplementary Fig. 3e) milled into the  
373 sample. The equations governing the relationship between the distances measured using the  
374 SEM and the milled volume dimensions are shown in Supplementary Fig. 3e.

375

376 Theta ( $\vartheta$ ) is the angle between the SEM and FIB column, in our case 52°, and alpha ( $\alpha$ ) is the  
377 milling angle relative to the milled ramp surface. Distances  $b_m$  and  $z_m$  were measured with the  
378 SEM without stage movement and trigonometry applied to work out the distances  $b$  and  $z$ .  
379 From  $z$  the distance  $h$  can be calculated and therefore area of the parallelogram subsequently.  
380 The volume was obtained by multiplying this calculated area by the width,  $w$ , of the initial  
381 milling square pattern.

382

383

384 **Equation 1**

$$volume = depth \times area$$

385

$$volume = \frac{wb_m z_m \sin(\alpha)}{\sin(\alpha + \theta) \sin(\theta)}$$

386 **Equation 2**

387

388 The sputter rate in  $\mu\text{m}^3/\text{s}$  was calculated for each mill by dividing the calculated volume by  
389 the milling time. Sputter rates in  $\mu\text{m}^3/\text{nC}$  were calculated by also dividing by the beam current  
390 that was measured for the chosen aperture.

391

392 Each mill was a standard square pattern with the Thermo Fisher Scientific default Silicon  
393 milling settings, which by default will set the pitch size and the dwell time. The pitch size was  
394 set as suggested as it is sample independent but milling time was changed.

395

396 Error bars shown on the plots in Supplementary Fig. 3 were calculated by taking the standard  
397 deviation of the 18 repeats (three repeats for six different times) at each beam current. Errors  
398 shown in Table 2 are the standard error derived using the least squares method from a linear  
399 line of best fit on the sputter rate ( $\mu\text{m}^3/\text{s}$ ) against beam current (nA) plots. The linear line of  
400 best fit was forced to pass through the origin. Error bars shown in Table 3 are the same as  
401 those on plots shown in Supplementary Fig. 3g and h and are the standard deviation.

402

### 403 **Cell culture**

404 UltraAuFoil on gold 200 mesh R2/2 grids (Quantifoil) were subjected to micropatterning using  
405 the Primo module from Alveole mounted on a Leica DMi8 microscope following the  
406 manufacturer procedure. Briefly, the grids were coated with polylysine (100  $\mu\text{g}/\text{ml}$ , 30 min)  
407 followed by mPEG-SVA (100 mg/ml, 1 h) and PLPP gel (1 h) prior to exposure to UV (50  
408  $\text{mJ}/\text{mm}^2$ ) to create circular patterns of 40  $\mu\text{m}$  diameter. Then the grids were profusely rinsed  
409 with PBS before incubation with fibrinogen couple to Alexa 633 (Thermo Fisher).  
410 Micropatterned grids were then stored in Hank's Balanced Salt Solution (HBSS) (Gibco).

411

412 HeLa cells were grown in Dulbecco's Minimal Essential Media (DMEM) (Gibco) with high  
413 glucose and non-essential amino acids complemented with 10% FBS, 1% glutamine and  
414 gentamycin (25  $\mu\text{m}/\text{ml}$ ). Cells were seeded on micropattern grids for 2 h before washing.  
415 Infection with *C.trachomatis* LGV02 were performed as previously described<sup>39</sup>. 24 h post  
416 infection cells were PF using the Vitrobot (Thermo Fisher) offsetting the blotting pad to favour  
417 back blotting. Just prior the PF, the cells media has been replaced with the complete media  
418 with 10% glycerol (v/v). Vitrified grids were then clipped (Thermo Fisher) into autogrids  
419 (Thermo Fisher) and subsequently stored under liquid nitrogen.

420

421 Plunge frozen *S.cerevisiae* (Yeast) samples were prepared as previously described<sup>38</sup>.

422

### 423 **Automated lamella fabrication**

424 Autogrid clipped TEM grids were loaded into a cassette and loaded into the PFIB's  
425 multispecimen cassette which was then subsequently loaded onto the stage using the robotic  
426 sample delivery device (termed Autoloader) (Thermo Fisher Scientific). The SEM was used to  
427 screen the grids to ensure suitability for lamella preparation (Fig. 1a). Once selected for  
428 milling, grids were coated with trimethyl(methylcyclopentadienyl)platinum(IV) using the gas  
429 injection system (GIS) and then sputter coated with platinum metal using an inbuilt platinum  
430 mass which was targeted with the FIB beam at 16 kV and 1.4  $\mu\text{A}$  to liberate clusters for  
431 adjacent surface metal coating.

432

433 Lamella sites were then identified and loaded into the AutoTEM cryo Software (Thermo Fisher  
434 Scientific) with the stage at eucentric height. The milling was then carried out using 30 kV  
435 argon automatically by the software. The milling procedure was as follows: eucentric height  
436 was refined at each site before milling stress relief cuts 5  $\mu\text{m}$  each side of the intended lamella  
437 using a 2.0 nA ion beam. Three rough milling steps were then used to remove material above  
438 and below the intended lamella position: i.) at 2.0 nA, 0.74 nA and 0.2 nA. This left a lamella  
439  $\sim 700$  nm thick. This thicker lamella was then “polished” using 60 pA and 20 pA down to a  
440 nominal software target thickness of 70 to 90 nm. As the software is intended originally to  
441 operate with gallium, a considerable offset exists between the target thickness and the final  
442 lamella thickness for plasma ion sources. Rough milling was carried out first on all lamella  
443 sites before proceeding onto the final polishing step. Drift corrected milling was used to  
444 ensure accurate milling. Each pattern was the default Thermo Fisher Scientific ‘Rectangle’  
445 with Silicon milling settings. All milling was done at 30 kV.

446

#### 447 **Cryo-electron tomography**

448 Multi-specimen cassettes with grids were directly transferred from the FIB/SEM to the  
449 transmission electron microscope via an Autoloader (Thermo Fisher Scientific). Tomography  
450 data were acquired with a Titan Krios (Thermo Fisher Scientific) transmission electron  
451 equipped with a Falcon 4 camera and a Selectris energy filter. Dose-symmetric tilt-series were  
452 collected using Tomo 5 (Thermo Fisher scientific) software at a nominal magnification of  
453 64000 x magnification (corresponding to a calculated magnification of 81081 x) at a 1.85  $\text{\AA}$   
454 pixel size in electron counting mode from  $51^\circ$  to  $-51^\circ$ , corrected for the pre-tilt of the lamellae,  
455 with  $3^\circ$  increments, with a total dose of  $175 \text{ e}^-/\text{\AA}^2$ .

456

#### 457 **Sub volume averaging**

458 Warp version 1.0.9<sup>40</sup> was used for gain- and motion correction using 10 frame groups per tilt,  
459 contrast transfer function (CTF) correction and creating tilt-series stacks. Tilt-series were  
460 aligned and reconstructed at a binning factor 8 using AreTomo 1.1.1<sup>41</sup> and tomograms were  
461 flipped using IMOD 4.11<sup>42</sup>, followed by bandpass filtering using EMAN 2.91<sup>43</sup>. Ribosomes were  
462 automatically picked using crYOLO 1.8.3<sup>44</sup> by creating a training dataset from 5 representative  
463 tomograms, followed by prediction on 180 tomograms, identifying 70436 putative ribosomes.  
464 Particles were extracted at a binning factor 4 using Warp with a box size of  $64 \times 64 \times 64$  pixels  
465 and imported into RELION 3.1<sup>45</sup>. An initial reference was generated by performing 3D  
466 refinement on a random subset of 1000 particles, followed by 3D refinement with all the  
467 particles. 3D classification was performed with a circular mask, identifying 18119 particles as  
468 ribosomes. Another round of 3D classification with a tighter ribosome-shaped mask was  
469 performed, yielding 8 classes with 80S ribosomes totalling 16204 particles, one class of the  
470 60S ribosome with 1896 particles and one junk-class with 19 particles. The 16204 particles  
471 from 80S ribosome classes were extracted with warp at binning 2 with a box size of  
472  $128 \times 128 \times 128$  pixels. Particle poses were subjected to 3D refinement in RELION using a  
473 thresholded ribosome mask. Particles poses for 15628 particles (six tomograms could not be  
474 refined post-RELION in M due to program errors) were imported into M version 1.0.9<sup>33</sup> for 5  
475 rounds of refinement where sequentially the following additional parameters were solved  
476 for: 1) 3 x 3 image warp grid and particle poses 2) 4 x 4 image warp grid 3) 3 x 3 x 2 x 10 volume  
477 warp grid 4) same settings as round 3 5) stage angles and per-particle defocus estimation.  
478 Structures determined were fitted using a human ribosome model (PDB: 4UG0<sup>46</sup>), and the

479 pixel size adjusted empirically to fit the PDB model. The determined pixel size for the map  
480 was 1.9 Å. This value was used for all FSC calculations (0.143) and B-factor calculations.

481

482 To produce structures of ribosomes at different distances from the lamella edge, a custom  
483 Python script was written that was able to calculate the distance of particles from XYZ  
484 coordinates within the output from M/RELION. The software is freely available on GitHub:  
485 <https://github.com/rosalindfranklininstitute/RiboDist>. Briefly, boundary models were  
486 created manually for each tomogram every ~100 YZ slices using IMOD<sup>42</sup> and converted into a  
487 text files for use in the script. The script interpolates the manually created boundary models,  
488 determines the nearest distance for each ribosome coordinate to the edge model, determines  
489 the distance of the top and bottom edge model in the centre of the tomogram and generates  
490 STAR format files with additional columns for the distance for every particle and the thickness  
491 of the lamella for each tilt-series. The column of values can then be used as threshold for  
492 generating STAR files in RELION3.1<sup>45</sup>. All particles with distances less than 30 nm were then  
493 parsed into a new STAR file and the number of particles used to generate a second  
494 randomised STAR file from the remaining particle list. The two STAR files were then used for  
495 new rounds of sub-volume averaging in RELION3.1. The local resolution was then determined  
496 after import into M, without any further refinement.

497

498 B-factors were calculated using RELION based on Rosenthal and Henderson<sup>34</sup>. Briefly,  
499 reconstructions were determined for subset of the particles from within the 30 nm layer and  
500 beyond this distance. The determined resolutions were then plotted as a function of the  
501 particle number and the 2 over slope of the linear fit calculated (Fig. 4).

502

### 503 **Data analysis**

504 To determine the distance where tilt-series were recorded to the platinum layer in the front  
505 of the lamellae, search images saved by Tomo 5 were stitched in FIJI<sup>47</sup> and the distance from  
506 each position to the front of the platinum layer along the direction of the FIB beam was  
507 measured with the line measurement tool.

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- 613

614 **Acknowledgements**

615 The authors would like to thank Lu Gan, Alex de Marco, and Sebastian Tacke for robust  
616 discussions around sputtering rates. This work was supported by the Wellcome Trust through  
617 the Electrifying Life Science project (220526/Z/20/Z to J.H.N). The Rosalind Franklin Institute  
618 is funded by UK Research and Innovation through the Engineering and Physical Sciences  
619 Research Council (EPSRC).

620

621 **Author Contributions**

622 M.D. prepared samples for cryo-FIB. C.B., M.D., and T.G. performed cryo-FIB, and collected  
623 cryoET data. C.B. and M.G. performed sub-volume averaging. N.B.y.Y. implemented new  
624 computation tools to help with data analysis. J.M.M., and Z.P., managed the development of  
625 the prototype cryo-FIB/SEM instrument, with input from M.D., J.H.N. and M.G. C.B., T.G., and  
626 M.G. prepared figures. J.H.N. and M.G. supervised the project. C.B. and M.G. wrote the  
627 manuscript. All authors reviewed the data and commented on the manuscript.

628

629 **Competing Interests**

630 J.M.M. and Z.P. are employees of Thermo Fisher Scientific. All other authors declare no  
631 competing interests.

632

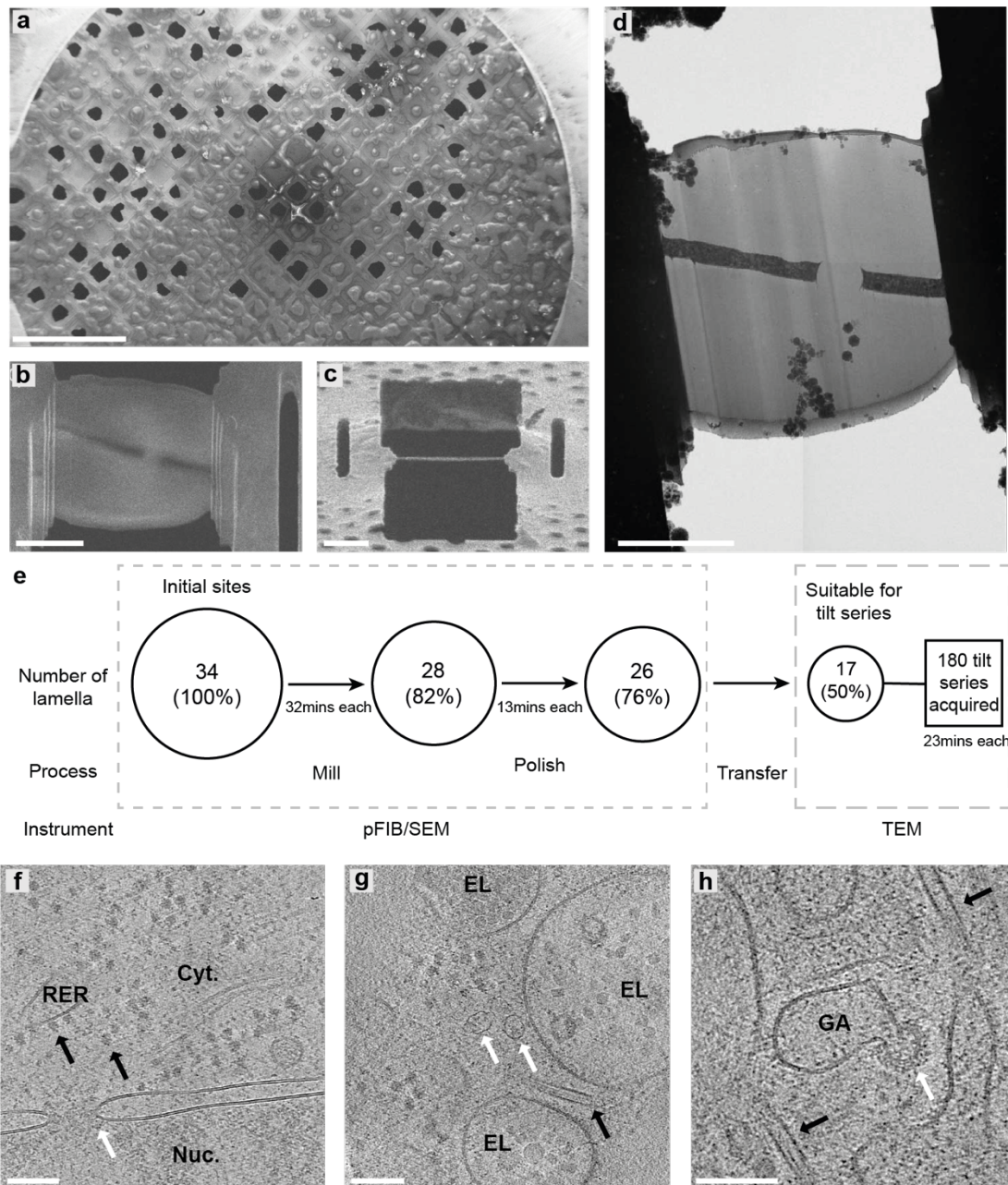
633 **Supplementary Information**

634 The online version contains supplementary material available at XX

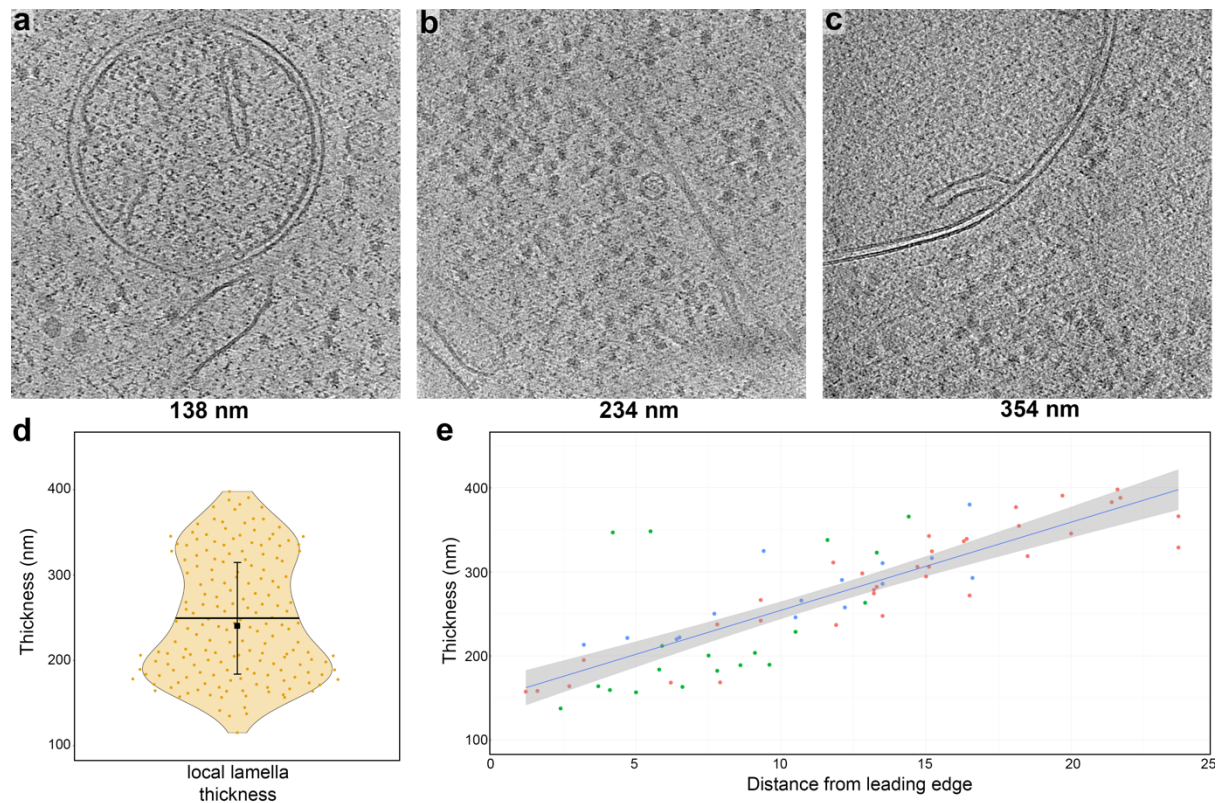
635

636 **Data Availability**

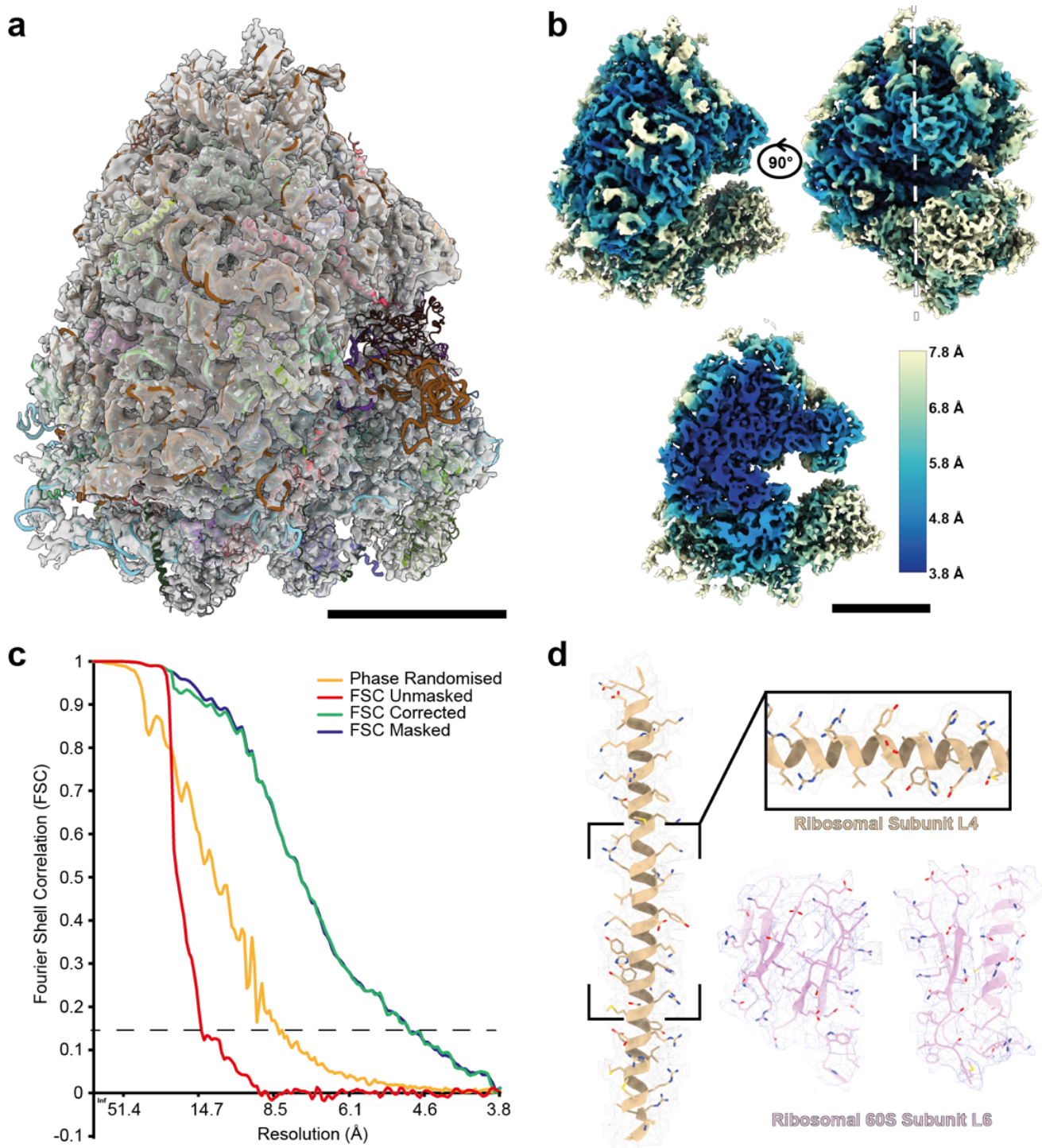
637 Sub volume averages are deposited in the Electron Microscopy Data Bank (EMDB) under the  
638 following codes: EMD-XXXX (full reconstruction), EMD-XXXX (lamella edge proximal) and  
639 EMD-XXXX (away from lamella edge). The frames and associated metadata needed to  
640 reconstruct the data (180 tomograms) have been deposited on the EMPIAR data server  
641 (EMPIAR-XXXXX).



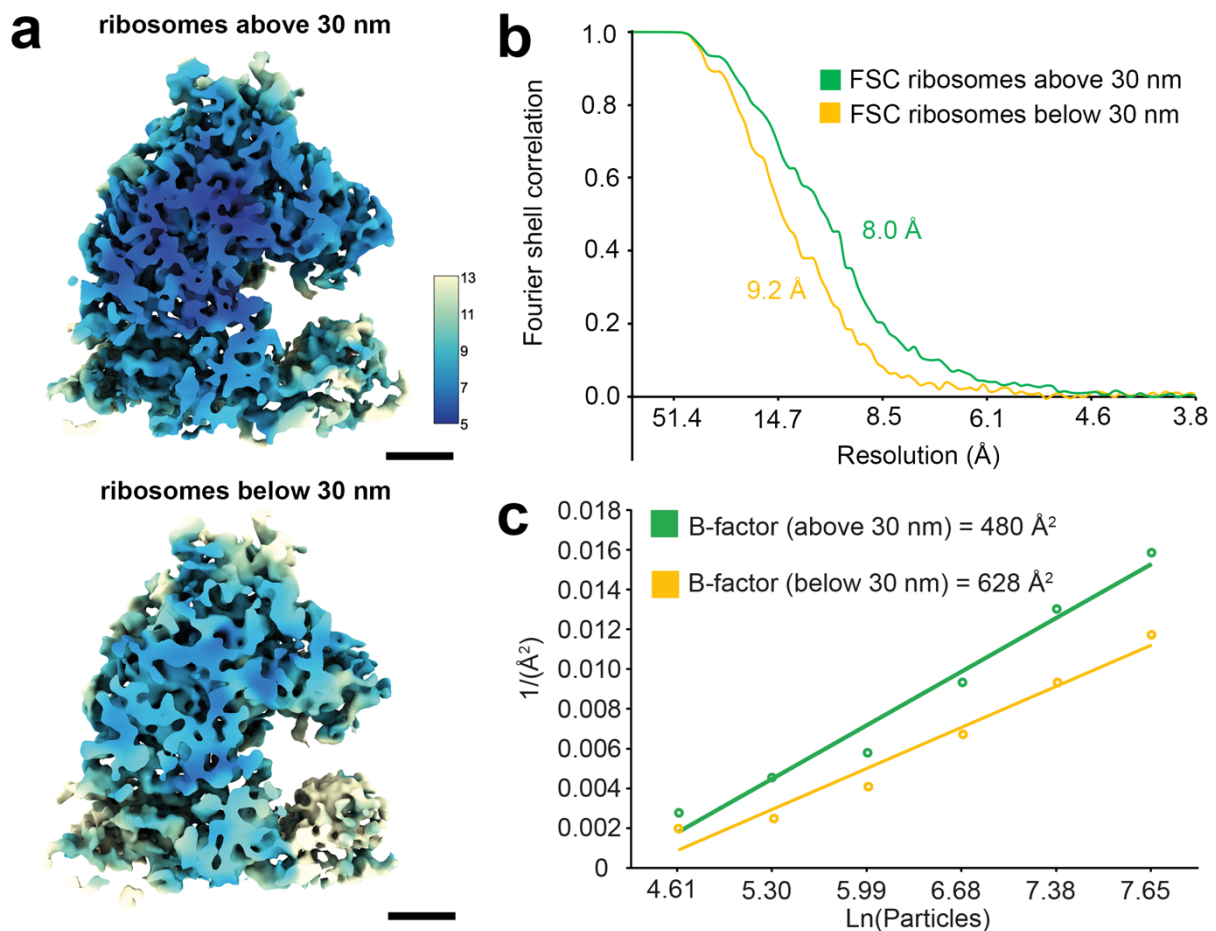
**Figure 1 – Overview of lamella production using plasma FIB milling.** (a) SEM overview of the uncoated grid. (b) SEM image of a finished lamella and (c) its respective FIB image. (d) Low magnification TEM image of the same lamella. Scale bar are 500  $\mu\text{m}$  in (a) and 5  $\mu\text{m}$  in (b), (c) and (d). The chart in (e) shows the number of successful lamella sites at each stage with the time taken between coarse and finer milling steps, starting with 34 initial sites down to 17 on which tilt series were acquired. (f-h) Representative tomographic slices containing several sub-cellular features, including (f) nucleus (Nuc.) and cytosol (Cyt.) with a nuclear pore (white arrow) in the nuclear envelope. Ribosomes (black arrows) are visible both free within the cytosol and tethered to the rough-endoplasmic reticulum (RER). scalebar: 100 nm; (g) endolysosomal vesicles (EL), a microtubule (black arrow) and vault complexes (white arrows). Scale bar: 100 nm; (h) microtubules (black arrows) and vesicle budding from the Golgi (GA) via a retromer coat (white arrow). The full tomographic volume is shown in Supplementary Video 1. Scale bar: 100 nm.



**Figure 2 - Plasma-milling produces lamellae with a range of thicknesses that exhibit usable sub-cellular tomographic data.** (a-c) representative tomographic slices of tomograms recorded on lamellae with different thicknesses (indicated below each panel). Scale bars: 100 nm. (d) Violin plot showing the distribution of tomogram thickness as determined in the reconstructed tomograms in all 180 tilt-series (orange, mean SD; 250 nm  $\pm$  70). Horizontal black bars indicate the mean thickness, error bars the standard deviation and the black square the median (241 nm). (e) Scatter plot for tomogram thickness and the distance to the front of the lamellae for all tomograms recorded on three different lamellae (datapoints coloured in green, red and blue for each lamella). A linear trend line (blue) is shown, with the 0.95 confidence interval shown in grey. Plots for 8 individual tomograms are also shown in Fig. S5.



**Figure 3 - Structure of the human 80S ribosome obtained from cellular tomograms. (a)** Density map of the 80S ribosome with fitted single-particle determined model of the human ribosome obtained from isolated ribosomes (PDB: 4UG0<sup>46</sup>) scale bar: 10 nm. **(b)** The same density map colour-coded for the local resolution from the outside (top) and a central slice (bottom, white line) (scale bar: 10 nm). **(c)** Fourier shell correlation for the 80S ribosome gives a masked global resolution of 4.9 Å (dotted line, FSC 0.143). **(d)** Representative fits into the EM density for different ribosomal regions (60S subunits L4 (gold) and L6 (magenta)), suggesting quality of the map.



**Figure 4 – Effect of distance from the lamella edge on the ability to determine ribosome structures.** (a) Ribosome structures were determined from acquired tomograms, where 2099 particles taken from outside (**top**) and within (**bottom**) the 30 nm of the edge of the lamella. Ribosomes are shown colour by the local resolution of the maps (key in Å). Scale bars are 5 nm and 1 nm (inset). (b) FSC curves are shown for the two maps, with a global resolution of 9.2 Å for the ribosomes within 30 nm of the edge (yellow) and 8 Å for ribosomes outside of this distance (green). (c) B-factors determined for particles above (green) and below (yellow) the 30 nm distance from the lamella edge after sub-volume averaging. These are calculated as 2 over the slope (Å<sup>2</sup>).

**Table 1 – Measured sputter rates taken from triplicate measurements on plunge-frozen yeast samples.** Sputter rates for Xenon, Nitrogen, Oxygen and Argon are shown. Sputter rates were measured as described in Supplementary Fig. 3. Errors shown are the standard error derived using the least squares method from a linear line of best fit that passes through the origin.

| <b>Plasma</b>  | <b>Xenon<br/>(MW 131.29)</b> | <b>Nitrogen (MW<br/>14)</b> | <b>Oxygen (MW<br/>15.99)</b> | <b>Argon<br/>(MW 39.95)</b> |
|--|------------------------------|-----------------------------|------------------------------|-----------------------------|
| <b>Sputter rate<br/>(<math>\mu\text{m}^3/\text{nC}</math>)</b> | $16.7 \pm 0.2$               | $10.6 \pm 0.2$              | $10.0 \pm 0.4$               | $4.3 \pm 0.1$               |

**Table 2 – Measured sputter rates taken from triplicate measurements on plunge-frozen HeLa samples with varying milling angles.** Sputter rates for Xenon, and Argon are shown. Sputter rates were measured as described in Supplementary Fig. 3. Errors shown are the standard error derived using the least squares method from a linear line of best fit that passes through the origin.

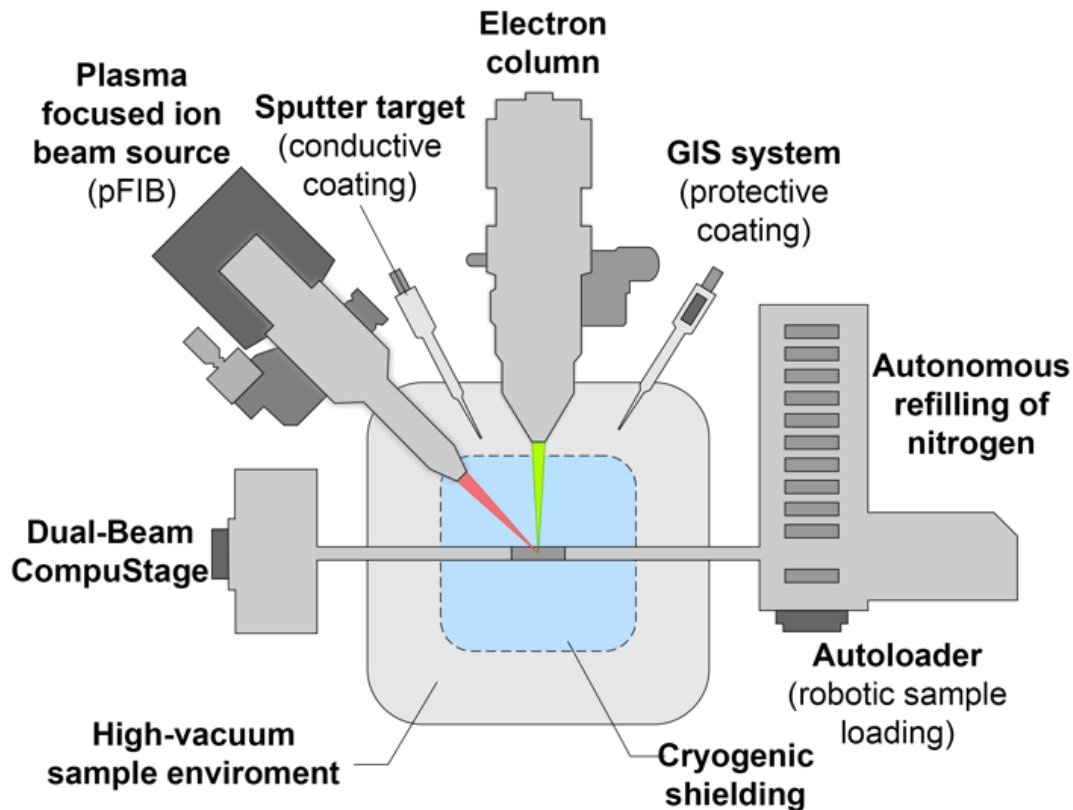
|                         | Xenon<br>(MW<br>131.29)                          | Argon<br>(MW<br>39.95)                           |
|-------------------------|--|--|
| Milling<br>angle<br>(°) | Sputter<br>rate<br>( $\mu\text{m}^3/\text{nC}$ ) | Sputter<br>rate<br>( $\mu\text{m}^3/\text{nC}$ ) |
| 90                      | 16.7 ±<br>1.3                                    | 4.5 ± 0.3  |
| 40                      | 34.1 ± 3.0                                       | 14.4 ±<br>1.4                                    |
| 30                      | 52.4 ±<br>7.9                                    | 23.1 ±<br>0.7                                    |
| 20                      | 80.1 ±<br>10.4                                   | 33.3 ±<br>2.7                                    |
| 10                      | 157.6 ±<br>16.1                                  | 59.3 ±<br>12.1                                   |

**Table 3 – Milling parameters used for lamella preparation.**

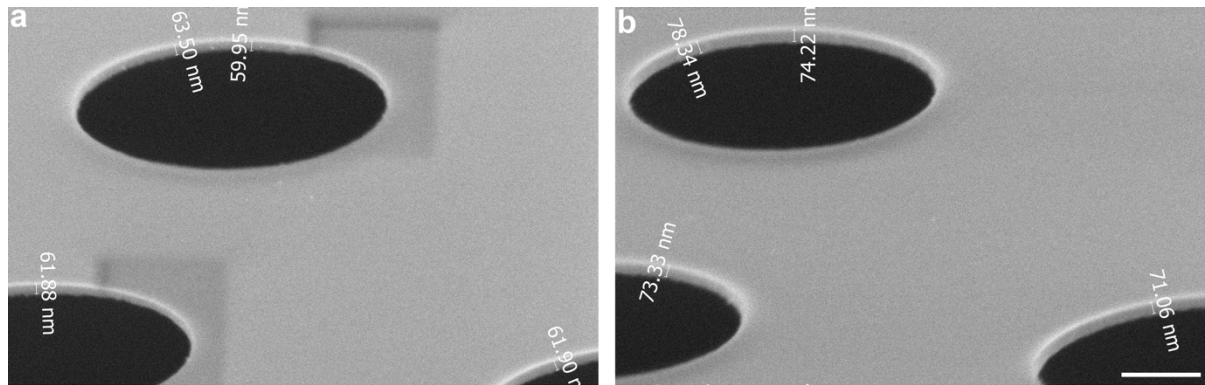
| Step              | Milling<br>current<br>(pA) | Width<br>( $\mu\text{m}$ )        | Height<br>( $\mu\text{m}$ ) /<br>Overlap<br>(%) | Offset<br>from<br>lamella<br>(nm) | Milling<br>depth<br>( $\mu\text{m}$ ) | Drift <sup>642</sup><br>correction<br>interval<br>(s) |
|-------------------|----------------------------|-----------------------------------|---|-----------------------------------|---------------------------------------|---|
| Rough<br>milling  | 2000                       | 15 (top)<br>14<br>(bottom)        | 7 (top)<br>7<br>(bottom)                        | 1500                              | 1.875                                 | 500   |
| Medium<br>milling | 200                        | 13.3<br>(top)<br>13<br>(bottom)   | 200%  | 600                               | 1.41                                  | 500   |
| Fine<br>milling   | 60                         | 12.7<br>(top)<br>12.2<br>(bottom) | 200%  | 300                               | 1.125                                 | 60  |
| Polish 1          | 60                         | 12                                | 200%  | 150                               | 0.5625                                | 30  |
| Polish 2          | 20                         | 12                                | 200%  | 0                                 | 0.5625                                | 10  |



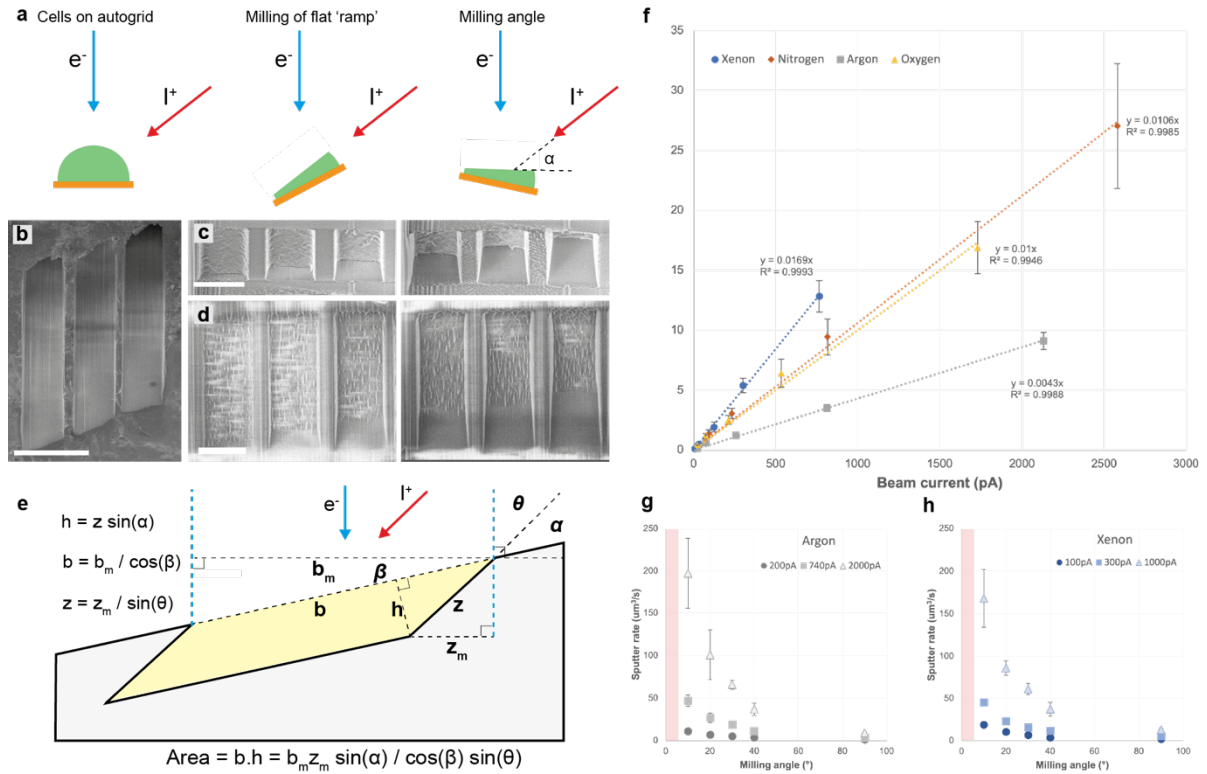
643 **Supplementary Figures**



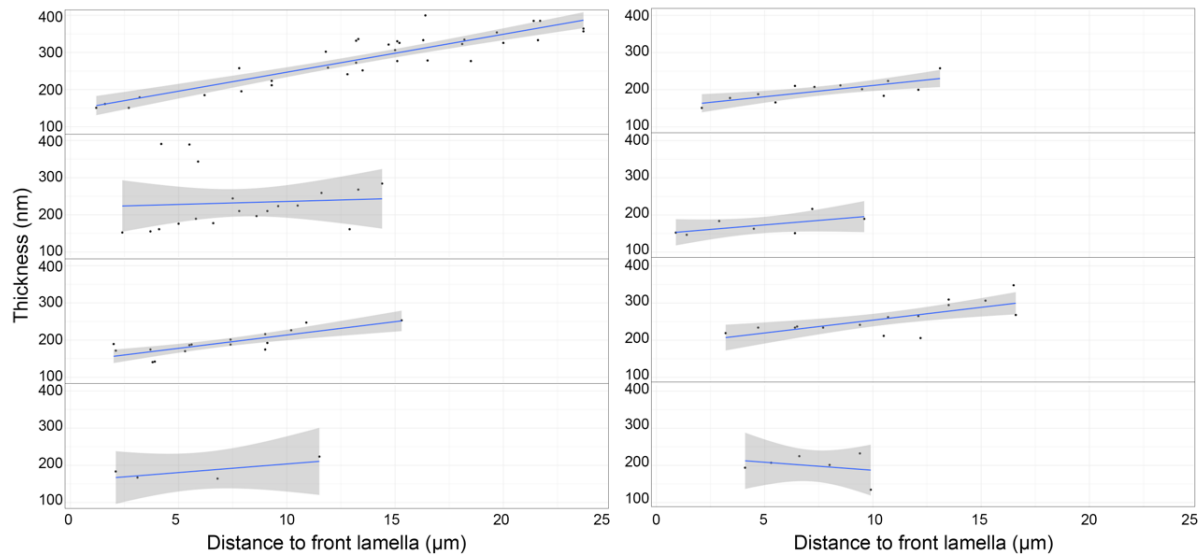
**Supplementary Figure 1 - Schematic overview of the FIB/SEM microscope used in this study.** The microscope has several elements that make it amenable to lamella production on long time scales in an automated fashion. This includes the high-vacuum chamber ( $10^{-7}$  mbar), multi-specimen loading and automatic refilling with liquid nitrogen. The instrument is equipped with a Hydra-type focused ion beam column, which utilises xenon, argon, or oxygen gas sources for plasma generation. The CompuStage enables robust position of the sample within the co-incidence point with  $\sim 500$  nm accuracy. Gas injection system (GIS) and sputter target enable protective layer and charge mitigation functionality to enable scanning electron imaging using the NiCol SEM column.



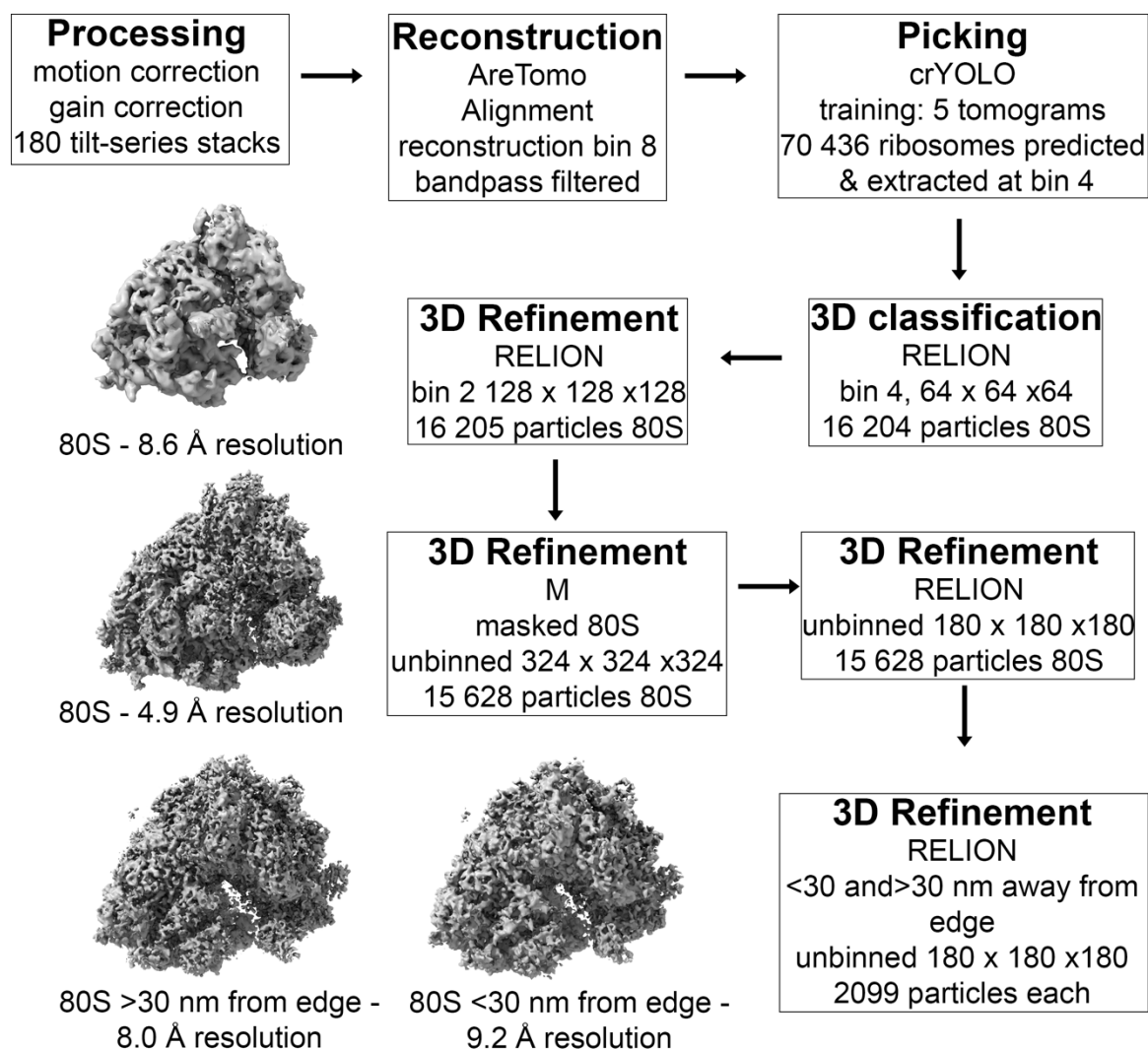
**Supplementary Figure 2 - Ice growth over time in in the column of the FIB/SEM microscope.** SEM images of the same location of a TEM grid were acquired at 0 h (a) and 8 h (b) and the EM grid foil thickness measured before and after, corrected for the stage tilt. This gave an average ice growth rate in the column of 1.6 nm/h. Scale bar: 500 nm. The pixel size for these images was 1.3 nm.



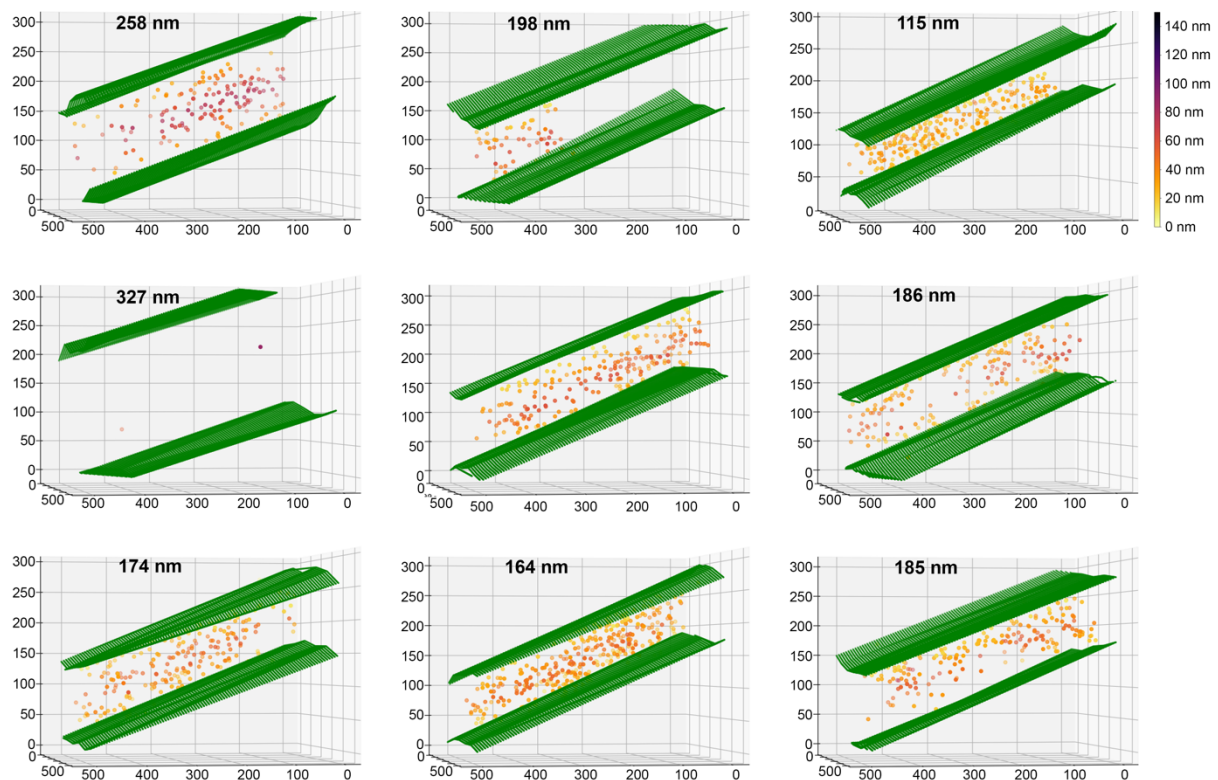
**Supplementary Figure 3 - Determination of PFIB sputter rates at 30 kV on vitrified yeast samples.** **(a)** Cartoon showing sputter rate measurement process, which involves milling a flat ramp **(b)** before tilting to present this fresh surface to the FIB at a milling angle of alpha. Six rectangles of varying exposure time were then milled and repeated three times, giving 18 measurements for each condition. Representative images of milling angles of 90° and 20° are shown in **(c)** and **(d)** respectively. The depth of the milled trench increases in **(c)** and **(d)** from left to right, as a function of milling time. The volume was then calculated from measurement of the milled depth using the SEM, as shown in **(e)** from which the sputter rate could be calculated (see materials and methods). **(f)** Sputter rate vs current for xenon, nitrogen, argon, and oxygen at a range of currents with a milling angle of 90°. Error bars are the standard deviation. **(g)** and **(h)** show the sputter rate as a function of milling angle for three different beam currents of argon and xenon respectively. The red area shows where the trend is expected to reduce back to zero but was not included as accurate measurement of such low angle mills was experimentally challenging. Error bars are the standard deviation.



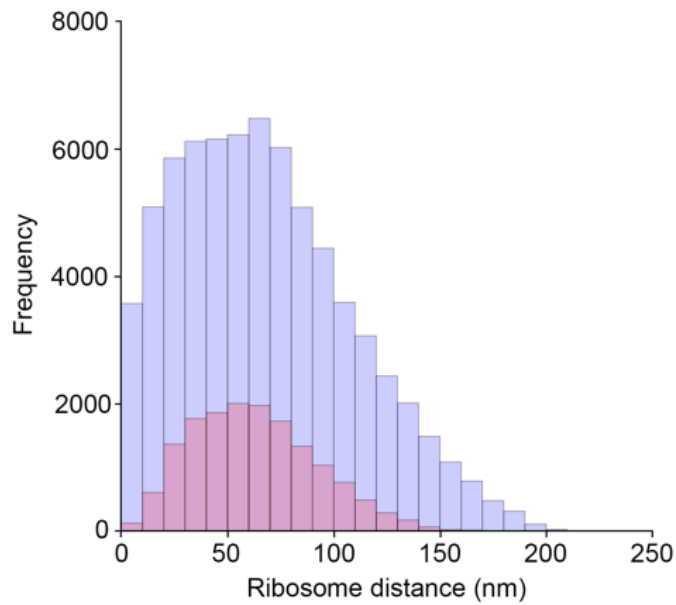
**Supplementary Figure 4 – Distances from the edge of the front of the lamella for tomograms acquired for 8 lamellae vs the measured thickness for a given tomogram.** Plots are shown for 8 lamellae of the distribution of thickness against the distance form the leading edge of the lamella. The plots show the approximate thickness profile of the lamella and indicate their flatness.



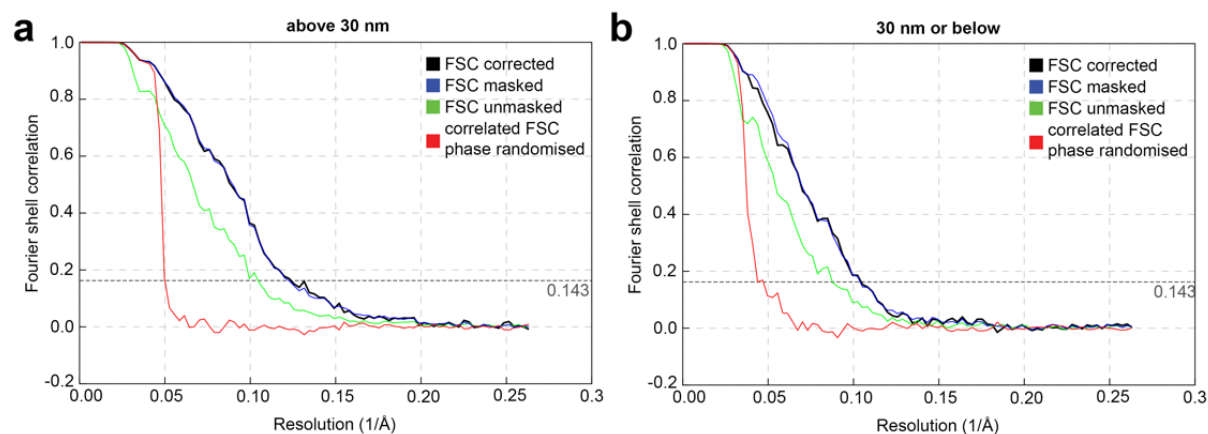
**Supplementary Figure 5. Schematic overview of sub volume averaging workflow.** All 180 tilt-series were processed, reconstructed and putative ribosomes were identified with crYOLO. Further 3D classification in RELION and subsequent 3D refinement in RELION and M yielded a 4.9 Å reconstruction of the 80S ribosome, and two separate reconstructions with a subset of the particles based on their distance from the lamella milling edge.



**Supplementary Figure 6 – Distribution of ribosome location within 9 representative tomograms.** Graphical representation of the annotated lamella edges (green) of 9 different tomograms. Ribosomes used for sub-volume averaging (see Fig. 3) are shown, where the colour of the ribosome points are indicative of their distance from the edge (in nanometres). XYZ coordinates are indicated for 8 x down sampled tomograms (pixel size: 1.48 nm).



**Supplementary Figure 7 – Distribution of ribosome distances to the milling edge before and after 3D classification.** The frequency putative-ribosomes identified with crYOLO (blue) and 80S ribosomes after 3D classification (red) based on the ribosome distance in nanometers (bin size: 10 nm).



**Supplementary Figure 8 – FSC curves for ribosome structures from within 30 nm distance from lamella edge and in the centre.** FSC curves are shown for ribosomes structures determined from: (a) 2099 ribosomes greater than 30 nm from the lamella edge and (b) 2099 ribosomes within 30 nm of the lamella edge. Resolutions are 8.0 Å and 9.2 Å, respectively.



## Supplementary Videos

**Supplementary Video 1.** Tomographic volume of the slice shown in Fig. 1h. Scalebar: 100 nm.

**Supplementary Video 2.** Video of the 3-dimensional volume local resolution map shown in Fig. 3b. Scale bar is 10 nm

**Supplementary Video 3.** Sub volume average density map of the ribosomal L4 subunit obtained in this study overlaid with the previously obtained structure of isolated human ribosomes (PDB: 4UG0<sup>46</sup>). Scale bar is 1 nm.