

1 **Magneto-active substrates for local mechanical stimulation of**
2 **living cells**

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4 Cécile M. Bidan^{1,2}, Mario Fratzl^{3,4,5}, Alexis Coullomb^{1,2}, Philippe Moreau^{1,2}, Alain H. Lombard^{1,2},
5 Irène Wang^{1,2}, Martial Balland^{1,2}, Thomas Boudou^{6,7}, Nora M. Dempsey^{3,4}, Thibaut Devillers^{3,4*} and
6 Aurélie Dupont^{1,2*}

7

8 1 Univ. Grenoble Alpes, LIPHY, F-38000 Grenoble, France

9 2 CNRS, LIPHY, F-38000 Grenoble, France

10 3 Univ. Grenoble Alpes, Inst. NEEL, F-38042 Grenoble, France

11 4 CNRS, Inst. NEEL, F-38042 Grenoble, France

12 5 Univ. Grenoble Alpes, CNRS, Grenoble INP, G2Elab, F-38000 Grenoble, France

13 6 CNRS, LMGP, F-38000 Grenoble, France

14 7 Univ. Grenoble Alpes, LMGP, F-38000 Grenoble, France

15

16 * Corresponding author

17 E-mails: aurelie.dupont@univ-grenoble-alpes.fr ; thibaut.devillers@neel.cnrs.fr

18 **Abstract**

19 Cells are able to sense and react to their physical environment by translating a mechanical cue
20 into an intracellular biochemical signal that triggers biological and mechanical responses.
21 This process, called mechanotransduction, controls essential cellular functions such as
22 proliferation and migration. The cellular response to an external mechanical stimulation has
23 been investigated with various static and dynamic systems, so far limited to global
24 deformations or to local stimulation through discrete substrates. To apply local and dynamic
25 mechanical constraints at the single cell scale through a continuous surface, we have
26 developed and modelled magneto-active substrates made of magnetic micro-pillars embedded
27 in an elastomer. Constrained and unconstrained substrates are analysed to map surface stress
28 resulting from the magnetic actuation of the micro-pillars and the adherent cells. These
29 substrates have a rigidity in the range of cell matrices, and the magnetic micro-pillars generate
30 local forces in the range of cellular forces, both in traction and compression. As an
31 application, we followed the protrusive activity of cells subjected to dynamic stimulations.
32 Our magneto-active substrates thus represent a new tool to study mechanotransduction in
33 single cells, and complement existing techniques by exerting a local and dynamic stimulation,
34 traction and compression, through a continuous soft substrate.

35

36 **Introduction**

37 Living cells have a sense of touch, which means that they are able to feel, respond and adapt
38 to the mechanical properties of their environment. The process by which cells convert
39 mechanical signals into biochemical signals is called mechanotransduction. Defects in the
40 mechanotransduction pathways are implicated in numerous diseases ranging from
41 atherosclerosis and osteoporosis to cancer progression and developmental disorders ^{1,2}. Since
42 the 1990s, different static studies focused on mechanosensing have shown that cells can
43 migrate along the rigidity gradient direction ³ and that stem cells can differentiate in vitro
44 according to their substrate's stiffness ⁴ and geometry ⁵. The interplay between a mechanical
45 force and the reinforcement of cell adhesion has also been documented ^{6,7}. In their natural
46 environment, cells face a complex and dynamic mechanical environment. Cyclic strain can
47 induce reorientation of adherent cells and affect cell growth depending on the temporal and
48 spatial properties of the mechanical stimulation ⁸⁻¹¹. The relevant timescales span from the
49 milli-second for the stretching of mechanosensitive proteins, minutes for

50 mechanotransduction signalling to hours for global morphological changes and even longer
51 for adapting cell functions¹². Taken together, previous works have shown that cells are
52 sensitive to both the spatial and temporal signatures of mechanical stimuli. In order to study
53 mechanotransduction, it is thus essential to stimulate cells with mechanical cues controlled
54 both spatially and temporally.

55 To address this topic, various methods have been proposed to exert experimentally controlled
56 mechanical stimuli on adherent cells¹³. For instance, local stimuli were applied by direct
57 contact with an AFM tip¹⁴, or with microbeads adhering on the cell membrane and actuated
58 by magnetic¹⁵ or optical tweezers¹⁶. Although local enough to address the subcellular
59 mechanisms of mechanotransduction, these methods involve intrinsic perturbations of the cell
60 structure through mechanical interactions with a stiff object of fixed geometry. Cell stretchers
61 were developed to induce mechanical stimulation via substrates of tunable substrate rigidity
62^{8,17}. Despite being more physiological and less invasive, such approaches only enable global
63 deformation at the cellular scale. To get around this limitation, different geometries of vertical
64 indenters were used to impose various deformation patterns on soft continuous cell substrates
65¹⁸. Surfaces made of micropillars that can be actuated with a magnetic field were proposed to
66 apply local and dynamic mechanical stimuli^{19,20}, but such discrete surfaces can affect the
67 cellular behavior^{21,22}.

68 Interestingly, none of these systems were used to apply compression on single cells. Yet,
69 compressive stress is present in healthy tissue such as cartilage^{23,24} and is crucial during
70 embryonic development²⁵. A compressive stress has also been shown to alter tumour growth
71 and shape in vitro²⁶⁻²⁸ which seems relevant in vivo where tumours have to grow against
72 surrounding tissue. Most of the studies on compressive stress have been carried out at the
73 tissue or multicellular level. There is currently a lack of studies at the single cell scale,
74 required to understand the possible differences in the mechanotransduction response between
75 traction and compression stresses.

76 In this article, we propose a new method to produce deformable substrates that enable local
77 and dynamic mechanical stimulation of cells plated on a continuous surface. These substrates
78 consist of iron micro-pillars spatially arranged in a soft elastomer and locally actuated using a
79 magnetic field generated by two electromagnets. Localized deformation of the substrate is
80 controlled through the current input to the coils of the electromagnet and is quantified by
81 tracking fluorescent markers incrustated under the surface of the elastomer. Traction force
82 microscopy (TFM) is used to estimate the magnitude of stress generated by the pillar on the

83 surface, which is in the range of typical stress applied by contractile cells. Stress variation
84 graphs demonstrate that cells spread on the magneto-active substrates can be mechanically
85 stimulated both in tension and in compression. Finally, a proof of principle experiment on
86 living cells is presented, showing increased protrusive activity of fibroblasts after a period of
87 mechanical stimulation.

88 The present approach allows the stimulation of living cells with deformation patterns
89 controllable in space and time. The magneto-active substrates can be deformed continuously
90 at the single cell scale both in traction and compression while the inherent coupling with TFM
91 allows to map the corresponding stresses. Thanks to their compatibility with standard
92 fluorescence techniques, these magneto-active substrates pave the way to quantitative studies
93 of intracellular biochemical responses resulting from controlled mechanical stimulations.

94 **Methods**

95 The active substrates developed in this work consist of an array of magnetic micro-pillars
96 embedded in a continuous layer of soft polydimethylsiloxane (PDMS). Upon application of a
97 magnetic field generated by two electromagnets, the micro-pillars are slightly tilted, resulting
98 in the local deformation of the surface of the PDMS. This section describes the methods used
99 to produce and characterize the micro-pillars and the substrates as well as the protocols to
100 actuate the magneto-active substrates for live-cell experiments (Fig.1).

101 **Microfabrication of magnetic micro-pillars and magnetic templates**

102 The micro-pillars consist of cylindrical silicon cores coated with a shell of soft magnetic iron
103 (Fe). The silicon pillars were produced by etching the surface of a silicon wafer according to
104 the following procedure. A network of 10 μ m diameter disks of S1818 resin was produced on
105 a silicon wafer by standard optical lithography. The silicon wafer was then etched over 30 μ m
106 using deep reactive ion etching (DRIE) following the Bosch process. The resin acts as a mask,
107 as the much higher rate of etching of the silicon compared to the resin results in the formation
108 of a dense array of silicon pillars which are 30 μ m in height and 10 μ m in diameter (Fig.1A).
109 The remaining resin is then dissolved in acetone, and the substrate cleaned with an O₂ plasma
110 to remove residual organic traces.

111 The entire wafer surface was covered with a trilayer of Ta(100nm)/Fe(15 μ m)/Ta(100nm)
112 deposited by triode sputtering at room temperature under a base pressure of 10⁻⁶mbar

113 according to a procedure described elsewhere²⁹. The role of the tantalum (Ta) is to protect the
114 iron from oxidation. The deposition by triode sputtering is relatively directive, so that the
115 layer deposited on the top of the pillar and on the substrate between pillars is thicker than the
116 layer deposited on the sidewalls of the pillars, as shown in the focused ion beam cut of a pillar
117 in Fig.1A. The pillars were mechanically detached from the substrate by swiping the surface
118 of the patterned substrate with a glass coverslip within a bath of absolute ethanol. The micro-
119 pillars from a known surface area were then air dried and stored in 5mL tubes for later use.

120 Using the magnetic field gradient force, it is possible to organize the magnetic micro-pillars
121 inside PDMS, to favour a particular spatial arrangement³⁰. For this purpose, we produced a
122 magnetic template using the approach described above (patterning of a Si substrate using
123 DRIE following by film deposition). As a template, 50 μm wide Si stripes separated by 500
124 μm were produced at the surface of a Si substrate and Ta(100nm)/Fe(15 μm)/Ta(100nm) was
125 deposited on the substrate using triode sputtering. The application of an external magnetic
126 field, produced by a macroscopic permanent magnet positioned below the template, served to
127 magnetize the Fe micro-stripes. The strong magnetic field gradients produced by the micro-
128 stripes of the template attract the magnetic micro-pillars, inducing an organization of the
129 pillars along parallel lines in the PDMS. The stray field of the bulk magnet serves to align the
130 long axis of the micro-pillars out of the plane of the substrate.

131 **Magneto-active substrate fabrication**

132 PDMS base and crosslinker from a standard kit (Sylgard 184, Dow Corning) were mixed in
133 the respective volume proportions 40:1. To reach a softer but non-sticky matrix, 8 volumes of
134 silicone oil (50cSt, 378356, Sigma) were added. After stirring thoroughly, the mixture was
135 degassed for 20min and then 1mL of the mixture was added into each tube containing about
136 $8 \cdot 10^4$ pillars. After stirring thoroughly in each tube, the PDMS/pillar mixtures were degassed
137 for 1h. Carboxylate-modified fluorescent beads (dark red FluoSpheres® 660/680, 0.2 μm , Life
138 Technologies) were diluted 1:500 in isopropanol and sonicated for 3 minutes. 50 μL of bead
139 solution were spread with a pipette tip on 35 x 35 mm squares of 50 μm thick polypropylene
140 sheet (PP301351/1, Goodfellow), the excess was removed and the isopropanol was then air
141 dried.

142 To assemble the magneto-active substrate, 120 μL of PDMS/pillar mixture was poured on
143 each polypropylene square and carefully covered with a 32mm coverslip to avoid the

144 inclusion of bubbles. To orient and organize the micro-pillars in the substrate, the stack was
145 positioned over the magnetic template on a large permanent magnet (60mm diameter,
146 Supermagnete) as shown in Fig.1B. The ensemble was kept at 65°C overnight to cure the
147 PDMS and then the magneto-active substrates were stored at room temperature in the dark.
148 Before use for cell culture, the polypropylene sheet was carefully removed and a 26mm
149 silicon ring activated with O₂ plasma was stuck on the PDMS surface to confine cells and
150 culture medium.

151 **Magnetic field source for actuation on the microscope stage**

152 A pair of electromagnets were made by winding 1mm diameter copper wire around 10cm
153 long aluminum tubes, i.e. 250 turns over 2.5 layers. Magnetically soft iron cores were used to
154 focus the magnetic field 2cm away from the coil and their tips were shaped as a bevel (30° on
155 one side, 13° on the other side). The coils were plugged in series on a current generator ($I_{\max} =$
156 9A) . The pair of electromagnets was mounted on the stage of an epifluorescence microscope
157 equipped with a chamber maintained at 37°C to enable live cell imaging. Before cell
158 experiments, the tips of the electromagnet cores were wrapped in a thin film of Teflon to
159 prevent sticking on the surface of the substrate and rusting of the core material (soft iron).

160 **Characterization methods**

161 The magnetic field generated in between the electromagnets was estimated macroscopically
162 with a Hall probe (Allegro A1302, Microsystem Inc.). The magnetic moment of individual
163 pillars was calculated from the magnetic moment of a population of about $8 \cdot 10^4$ pillars
164 measured using an extraction magnetometer.

165 The average dimensions of the magnetic pillars after collection were estimated from the
166 measurement of 39 pillars with a bright field microscope.

167 The average thickness of the substrate was measured optically on samples containing
168 fluorescent beads at both interfaces of the PDMS. To assess the local mechanical properties of
169 the magneto-active substrates, force-indentation profiles were performed in 1% Pluronic
170 diluted in PBS at a frequency of 1Hz using an atomic force microscope BioCatalyst (Bruker)
171 equipped with borosilicate sphere-tipped cantilevers of radius $R = 2.5\mu\text{m}$ (Novascan
172 Technologies) and a spring constant of 0.4 N/m. Young's moduli were calculated by least-
173 square fitting of the experimental force indentation curves using NanoScope Analysis

174 (Bruker). Soft PDMS patches of 20mm diameter and 2mm thickness were also produced to
175 measure the global viscoelastic properties (shear storage modulus G' and loss modulus G'') of
176 the substrate with a rheometer (Bohlin) used in parallel plane geometry at deformation
177 amplitudes γ varied between 0.01% and 20% of shear and frequencies varied between 0.01Hz
178 and 10Hz.

179 **Numerical modelling**

180 In a simple approximation, the present system can be considered as a network of elongated
181 magnets which experience a torque due to the application of a transverse (in-plane) magnetic
182 field. Numerical modelling was performed using COMSOL Multiphysics 5.0 (COMSOL
183 Group, Stockholm, Sweden) in order to define the limits of this simplistic approach and
184 understand the parameters which are relevant to the dimensioning of the system. All
185 simulations were performed on a Dell OptiPlex 9020 (Dell Inc., Round Rock, TX.) powered
186 with an Intel Core i5 4th generation / 3.3 GHz (Intel Corporation, Santa Clara, Ca.).

187 In a first simulation, the magnetic field distribution produced by the electromagnets was
188 modelled in 2D (i.e. in the x - z plane bisecting the electromagnets' cores, where x is in the
189 plane of the substrate and z is out of the substrate plane – see Fig.1C). As the magnetic field
190 source, we considered the magnetic cores of the coils, and used as input their real geometry
191 with a distance of 6mm between the cores' apex. To reduce the computational load, the
192 electromagnets were modelled as permanent magnets, homogeneously magnetized along their
193 axis. Their magnetization was adjusted in such a way that the theoretical value of the
194 generated field's projection along the horizontal direction (B_x) matches the experimental
195 values measured in the centre of the system for a current of 5A.

196 A second model was developed to analyse quantitatively the magnetic and mechanical
197 response of a single pillar in PDMS when exposed to an external magnetic field. In this 3D
198 magneto-mechanical model, the pillar was represented as a cylinder of silicon, with a
199 diameter of 10 μ m and a height of 25 μ m. The Fe shell is 2 μ m thick on the sidewalls and 10 μ m
200 thick on the top of the pillar. The relative permeability of iron was taken as $\mu_r = 5000$. The
201 PDMS film is 115 μ m thick, with the top of the pillar being 1 μ m below the PDMS surface.
202 The top surface of PDMS is described as a free surface while the bottom one is mechanically
203 attached to the glass substrate. The PDMS was approximated as a linear elastic material, with
204 a Young's modulus $E = 20.3$ kPa and a shear modulus $G = 7.1$ kPa as estimated
205 experimentally.

206 **Cells**

207 The magneto-active substrates were sterilized in 70% ethanol for at least 20min and rinsed
208 with PBS before use. The surface was then incubated for 1h in 20 μ g/mL fibronectin (Sigma)
209 solution diluted in PBS, a disk of Teflon was used to spread the fibronectin drop on the
210 hydrophobic surface. The substrate was rinsed with PBS and conditioned with culture
211 medium at 37°C for at least 30min.

212 These substrates were tested with wild type NIH3T3 fibroblasts as well as NIH3T3 cells
213 expressing vinculin-eGFP (kindly provided by K. Miroshnikova and C. Albiges-Rizo, Institut
214 Albert Bonniot INSERM U823/ERL CNRS 3148, Grenoble) previously cultured at 37 °C in a
215 humidified 5% CO₂ incubator with Dulbecco Modified Essential Medium D-GlutaMAX
216 (Gibco) supplemented with 1% Penicillin Streptomycin (Sigma) and 10% fetal bovine serum
217 (Gibco). The cells were seeded at low density on the magneto-active substrates coated with
218 fibronectin (around 3·10³ cells/cm²). The magneto-active substrates and the fluorescent cells
219 were imaged on an epifluorescence microscope (Olympus IX83) equipped with a white laser
220 (Fianium) to excite the fluorescent beads and the eGFP.

221 Cell response was assessed by imaging the eGFP fluorescence every 4s, during a 30min
222 experimental procedure, which consists of a 10min rest, followed by a 5-minute dynamic
223 stimulation with a 0.25Hz rectangular signal of 5A current input in the electromagnets, and a
224 final 15min rest. Cell culture and experiments were carried out in accordance with the
225 relevant guidelines and regulations.

226 **Image analysis**

227 **Deformation of the substrate.** To quantify the deformation induced by the actuation of the
228 pillars, surface displacements were determined from images of the fluorescent beads with and
229 without application of the magnetic field, by using an algorithm originally developed for cell
230 traction force microscopy (TFM) ³¹. After correction for experimental drift, the images were
231 divided into 256x256 pixel square sub-images. First, cross-correlation was used to yield the
232 average displacement on each pair of sub-images, which were shifted accordingly. The
233 fluorescent beads were then tracked with high accuracy (20 nm) to obtain a displacement map
234 with high spatial resolution (<5 μ m) ³². The final displacement was obtained on a square grid
235 with 1 μ m spacing using linear interpolation. Force reconstruction was carried out under the

236 assumption that the substrate is a linear elastic half space considering in-plane stress only,
237 using Fourier Transform Traction Cytometry with zeroth-order regularization³³. The problem
238 of calculating the stress field (T_x and T_y) from the displacement was solved in Fourier space,
239 then inverted back to real space. The final stress magnitude $\sqrt{T_x^2 + T_y^2}$ was obtained on a
240 grid with 1 μ m spacing. To distinguish clearly the regions undergoing traction and those under
241 compression, the derivative of the stress was calculated with respect to each direction using
242 the central difference method, and the sign of $\frac{dT_x}{dx} + \frac{dT_y}{dy}$ gives the type of stress applied on the
243 PDMS surface. Positive stress variations correspond to tensile stress whereas negative ones
244 correspond to compressive stress. All calculations and image processing were performed with
245 Matlab R2015b and figure generation was done with Python.

246 The same methods were used to derive the distribution of stress magnitude and spatial stress
247 variations induced by the cells. In that case, the reference image was taken after detachment
248 of the cell with 0.2% SDS, without any magnetic field.

249 **Protrusion activity of the cells.** Images were first Gaussian filtered to reduce noise and
250 thresholded to obtain a mask of the cell. The contour velocity was evaluated using a method
251 previously described^{34,35}. The normal velocity $v(x,y,t)$ is estimated as

$$252 \quad v(x, y, t) = \frac{[I(x, y, t + \Delta t) - I_0] - [I(x, y, t - \Delta t) - I_0]}{|\nabla I(x, y, t)|}$$

253 where $I(x, y, t)$ is the gray level at position (x, y) and time t , I_0 is the same threshold as the
254 one used to obtain the masks, $|\nabla I(x, y, t)|$ is the local gradient magnitude, and Δt is a number
255 of frames.

256 For each cell, a velocity map was derived by i) averaging velocity values of each boundary
257 pixel of the cell and its 12 nearest neighbours and ii) sorting these resulting values as a
258 function of the normalized position along the perimeter of the cell at a given time. The color-
259 coded smoothed velocities along the normalized perimeter were then displayed as a function
260 of time throughout the experiment. To obtain a value for each cell, we extracted the velocity
261 of the most active region as a function of time by averaging the 5% highest velocity values.
262 The ratio between the average maximum velocity of a 5-minute period after and that of the 5-
263 minute period before mechanical stimulation was then calculated.

264 **Data Availability**

265 The datasets generated during and/or analyzed during the current study are available from the
266 corresponding authors on reasonable request.

267 **Results and Discussion**

268 Magneto-active substrates were fabricated by incorporation and organization of magnetic
269 micro-pillars in a continuous layer of soft elastomer (Fig.1). The elements involved in the
270 fabrication and actuation of the magneto-active substrates were characterized, before plating
271 cells on their surface and measuring their protrusive activity after stimulation, to demonstrate
272 the potential of this technology for mechanobiology studies.

273 **Physical properties of the pillars**

274 Since the aim is to produce a torque on the soft magnetic element embedded in the elastomer,
275 the element needs to be anisotropic in shape (if it were magnetically soft but isotropic in shape,
276 the magnetic moments would simply rotate to align with the applied field, resulting in no
277 tilting of the object itself). The considered micro-pillars consist of a core shell structure, based
278 on a silicon cylindrical core coated with an iron shell, produced by lithography, DRIE and Fe
279 deposition (Fig.1A). After collection, the dimensions of 39 pillars were measured to be $33.5 \pm$
280 $2.5 \mu\text{m}$ in height and $15.7 \pm 1.2 \mu\text{m}$ large. This indicates that the micro-pillars are broken
281 roughly $10 \mu\text{m}$ above their base, which corresponds to the thickness of Fe deposited between
282 the pillars. The saturation magnetic moment of about $8 \cdot 10^4$ pillars was estimated to be $3.3 \cdot 10^{-2}$
283 $\text{A}\cdot\text{m}^2$, i.e., $4.1 \cdot 10^{-9} \text{A}\cdot\text{m}^2$ per pillar. Considering that the saturation magnetization of iron is
284 $1.7 \cdot 10^6 \text{A}\cdot\text{m}^{-1}$, the volume of iron deposited on a given pillar is estimated to be around 2400
285 μm^3 , which matches the order of magnitude obtained from a geometrical estimation.

286 The protocol of assembly established in the methods section produces magneto-active
287 substrates containing magnetic micro pillars arranged in a layer of soft PDMS according to
288 the chosen magnetic template (see Fig.1B). To better control the position of the pillars, we
289 tried using template arrays of magnetic islands that could trap individual magnetic micro-
290 pillars at regular distances. However, trapping one pillar per island proved difficult. It is
291 important to mention that organizing the pillars using a stripe-template does not affect their
292 ability to locally deform the substrate, provided that the pillar density is kept low enough to
293 prevent magnetic and mechanical interactions between pillars.

294 **Mechanical properties of the substrate**

295 PDMS is a biocompatible elastomer widely used as a cell substrate. We first observed that
296 standard Sylgard 184 used in a ratio base to crosslinker of 40:1, leads to a substrate with a
297 Young's modulus of 40kPa and a sticky surface. We found that adding 8 parts of silicon oil to
298 the Sylgard mixture reduced both the rigidity, to a suitable range to be deformed by cells so as
299 to probe their contractile forces, and the stickiness of the surface, to facilitate handling of the
300 samples. The mechanical properties measured on 3 samples of soft PDMS with a rheometer
301 indicate that both the shear storage modulus G' and the shear loss modulus G'' are
302 independent of the amplitude of deformation γ between 0.01% and 20% of shear when
303 deformed at 1Hz (see Supplementary Fig.S1), with $G'=6756$ Pa and $G''=1117$ Pa. However,
304 varying the frequency between 0.01Hz and 10Hz of a 10% shear reveals that below 0.2Hz, G''
305 becomes an order of magnitude inferior to G' which is about 6000Pa. The material's
306 behaviour is thus dominated by elasticity as indicated by the phase angle $\delta = \text{Arctan}\left(\frac{G''}{G'}\right)$,
307 which remains close to 0 at the frequencies of interest (see Supplementary Fig.S1). For
308 homogeneous isotropic linear elastic materials, Young's modulus can be derived as $E =$
309 $2 \cdot (1+\nu) \cdot G'$. Since Poisson's modulus of our PDMS is $\nu = 0.418$ ³⁶, we estimate $E = 19.2$ kPa
310 when deformed at 1Hz. This global value matches the results of 5 mechanical profiles
311 obtained by atomic force microscopy, as the local Young's modulus obtained away from a
312 pillar is $E = 20.3 \pm 2$ kPa (Fig.2A). The PDMS mixture used in this study leads to a soft
313 substrate with negligible viscosity at the frequencies of interest, and thereby allows us to use
314 the TFM algorithm and derive force fields from the displacement field of the fluorescent
315 beads.

316 As expected, the rigidity sharply increases by an order of magnitude when indenting above
317 the pillars (Fig.2A), which leads us to avoid analysing cells lying above pillars.

318 The layer of soft PDMS is 115.5 ± 12.5 μm thick, as measured in 9 points of 3 different
319 samples. As shown in Fig.1B, this layer is thick enough to neglect the effect of the glass
320 coverslip on the mechanical properties of the substrate and on the pillar displacement close to
321 the surface, while thin enough to have optical properties compatible with fluorescence
322 imaging. Of note, the fluorescent beads embedded under the surface of the soft PDMS are
323 homogeneously dispersed in a single plane with an average density of 0.2 beads/ μm^2 ,
324 including above the pillars, as verified from images taken on a substrate positioned upside-
325 down (see Supplementary Fig.S2). This density of beads is low enough to neglect their

326 influence on the mechanical behaviour of the soft substrate^{37,38} and allows automated tracking
327 of the displacement with high accuracy (20 nm) and spatial resolution ($<5 \mu\text{m}$). This
328 reproducible method of bead incorporation is thus particularly suited to improve TFM on a
329 soft elastomer³⁹ and widens the range of soft substrates that can be used to study cell
330 contractility.

331 **Generation of the magnetic field**

332 To actuate the magnetic micro-pillars, two electromagnets have been designed to generate a
333 predominantly in-plane magnetic field (Fig.1C). While the iron cores have been designed to
334 focus the field at the surface of the magneto-active substrate, the dimensions and the position
335 of the electromagnets were determined by the geometrical constraints related to the
336 microscope and the cell culture dishes.

337 The B_x component of this field was measured as a function of current input into the coils with
338 a Hall probe positioned at the mid-point between the two cores (Fig.2B). We restricted the
339 current input to a maximum intensity of 5A for the rest of the experiments, which corresponds
340 to a magnetic field of 100mT.

341 The magnetic field distribution calculated within the actuation zone is represented in Fig.2C.
342 This simulation indicates that the B_x component of the magnetic field is relatively
343 homogeneous in-between the electromagnets, while the vertical component (B_z) varies
344 symmetrically with respect to the centre, where it vanishes. In the plane of the pillars and
345 roughly $70\mu\text{m}$ below the electromagnets, if B_x is set to $\sim 100\text{mT}$ in the middle, as generated
346 with 5A in the coils, then B_z varies from 0mT (exactly mid-way between the electromagnets)
347 to $\pm 70\text{mT}$ (near the cores' apex). The important role of the out-of-plane component is
348 described below.

349 **Actuation of the substrate**

350 *Influence of B_z*

351 The B_z component of the applied field serves to induce a vertical component of the
352 magnetization of the micro-pillars. This in turn allows the B_x component of the applied field
353 to produce a torque on the micro-pillar. The role of B_z in inducing a mechanical response of
354 the substrate was assessed within the framework of the magneto-mechanical model, by
355 varying B_z between 0 and 70mT while fixing the horizontal field B_x at 100mT and neglecting

356 the B_y component. Fig.3A shows the vertical cut of a pillar experiencing a purely horizontal
357 magnetic field ($B_z=0$). The magnetization of the pillar lies practically in the horizontal plane
358 and there is no significant torque on the pillar. However, as soon as the pillar experiences a
359 vertical field component B_z , even small compared to B_x , the magnetization in the sidewalls
360 tends to align along the long axis, forming a non-zero angle with the applied field, and
361 resulting in an effective torque on the pillar. Pillars experiencing a vertical magnetic field of
362 20mT are predicted to produce displacements of around $3.1\mu\text{m}$ at the surface of the PDMS
363 (Fig.3B). These simulations show that a non-zero vertical component of the magnetic field B_z
364 is essential to take advantage of the shape anisotropy of the iron shell to align the
365 magnetization of the pillar along its long axis, and thereby generate a torque able to deform
366 the surface of the substrate. Fig.3C indicates that this effect is modulated by the value of the
367 vertical magnetic field, since for $B_x=100\text{mT}$ and $B_z=50\text{mT}$, the surface above the pillar is
368 expected to experience in-plane displacements of over $7.5\mu\text{m}$. Note that the slight vertical
369 distortion ($<1\mu\text{m}$) expected close to the tilted pillar at high deformation was observed
370 experimentally through a local defocusing of the fluorescent beads.

371 The magneto-mechanical model indicates that the displacement generated by a pillar
372 decreases with increasing distance from the closest core, which was observed experimentally
373 (Fig.S3). Thus in a given experiment with cells spread across the substrate, the influence of
374 different values of displacement on a given set of cells can be studied.

375 *Experimental deformation.*

376 The magneto-mechanical simulation of a pillar placed at 1mm from the magnetic core was
377 performed with $B_x = 119\text{mT}$ and $B_z = 27\text{mT}$, as estimated from the magnetic field distribution
378 simulation of the experimental actuation system (Fig.2C). A top view of the displacement
379 field in the xy plane was used to estimate the stress magnitude of the surface using the traction
380 force microscopy algorithm, and the stress variations. These numerical estimations agree with
381 experimental measurements performed on 5 pillars significantly actuated by electromagnets
382 powered with 5A ($B_x\sim 100\text{mT}$, $B_z\sim 70\text{mT}$). Indeed, such actuation generates a displacement
383 field that decays sharply by 50% within $20\mu\text{m}$ (Fig.4A), which corresponds to the cellular
384 length scale. The resulting deformation of the surface is qualitatively symmetric with respect
385 to the pillar.

386 The displacements induced by a series of 15 pillars aligned about 1.5mm away from the tip
387 ($B_z\sim 20\text{mT}$) and experiencing an incremental actuation (from 0 to 5A in the coils) was

388 systematically measured to estimate the variability of the actuation between pillars (see
389 Supplementary Fig.S4). The broad distribution of maximum displacements, which increases
390 with the current input, can be explained by the influence of pillar geometry on the resulting
391 actuation. 3D magneto-mechanical simulation was performed on core-shell pillars with
392 different shapes (cylinder and cone of different heights), and positioned with the iron cap
393 either towards the surface (up) or towards the coverslip (down) (see Supplementary Fig.S5).
394 We found that cylinders are insensitive to the up/down orientation whereas conical pillars are
395 sensitive to it, showing a 50% displacement increase when placed upside down. Hence, a
396 conical pillar with iron cap up induces 25% less deformation than a cylindrical one, but when
397 placed upside down the obtained deformation becomes larger than that of a cylindrical pillar.
398 Regarding the influence of the pillar length, which can be affected during mechanical
399 collection from the wafer, we found that a cylindrical pillar is expected to deform the surface
400 about 22% less if reduced by 5 μm from its basis, and up to 45% less if reduced by 10 μm .

401 Coupling the active substrates with TFM allows to measure the actual displacements and
402 stresses induced by each pillar after removal of the cells. Hence, the variability of actuation
403 amplitudes from pillar to pillar can be used to our advantage: a range of stimuli can be
404 explored in one experiment, where the precise stress applied on each cell is known.

405 Deriving the stress magnitudes at the surface of the 20kPa PDMS reveals that the magneto-
406 active substrates allow generating a stress within 30 μm around the pillars, up to 2.4kPa of
407 stress amplitude in the close vicinity of the pillars (Fig.4B). Such a value corresponds to the
408 range of stress that cells are able to generate on their substrate^{40,41} and therefore supports the
409 relevance of the present system in mimicking the mechanical coupling of neighbouring cells
410 through their matrix⁴². In terms of force, magneto-active substrates can locally transmit nN-
411 forces to cell adhesions, if a typical adhesion surface of 1 μm^2 is considered, which also
412 compares to forces generated by single adhesions^{43,44}. Moreover, the maps of stress variation
413 show a clear localization of the mechanical stimulation in a 30 μm -radius around the pillars.
414 This subcellular length scale confirms that the present magneto-active substrates are
415 appropriate tools to investigate the spatiotemporal evolution of intracellular signals triggered
416 by a local extracellular mechanical cue sensed at the focal adhesions.

417 Displaying the stress variation map also highlights the different modes of stimulation
418 available with the magneto-active substrates. Indeed, the torque applied by the magnetic field
419 on a pillar stresses the surface in tension on one side (positive stress variation) and in
420 compression on the other side (negative stress variation). This feature offers the possibility to

421 perform both stretching and compression experiments on the same setup and thus to propose
422 rigorous comparisons of the differential response of stimulated cells. This is particularly
423 relevant to investigate muscle cells, cell types sitting in weight bearing tissues^{24,45} but also
424 stem cells, the differentiation of which is already known to be tuned by static mechanical cues
425 of their environment such as stiffness¹² and geometry⁵ via the mechanotransduction
426 processes.

427 *Control through the current input.*

428 Electromagnets rather than permanent magnets were chosen to facilitate dynamic control of
429 the system. As expected from the evolution of the magnetic field measured experimentally
430 (Fig.2B), the deformation, stress and stress variation profiles progressively spread along the x
431 direction up to 5A and stabilize thereafter (Fig.S6). This observation supports our choice to
432 limit the stimulations to 5A for subsequent experiments. Cyclic stimulations manually
433 performed between 0 and 5A on 4 different pillars show that the temporal pattern of
434 deformation is reproducible over several cycles of actuation (Fig.4D). The residual
435 deformation appearing after the first cycle can be explained by a slight defocusing of the
436 surface around the pillar and/or a local micro-delamination at the interface between the soft
437 PDMS and the pillar. Besides tuning the stress applied to the cell in a physiological range by
438 varying the amplitude of the current, it is also possible to tune the temporal pattern of the
439 stimulation with a function generator. Micrometric displacements of the pillars were
440 detectable up to 10Hz with our optical setup, which is comparable to the current cell stretcher
441 technologies⁸.

442 **Application to cells**

443 To test our system in relevant conditions for biological studies, a low density of NIH3T3
444 fibroblasts was plated on the magneto-active substrates after functionalization of the surface
445 by adsorption of fibronectin. The cells adhered normally and homogeneously within 3 to 4
446 hours after seeding (Fig.5A).

447 First, we investigated whether the stresses induced by a pillar in the vicinity of a cell can
448 mimic the action of neighbouring cells, both in magnitude and spatial patterns. Fig.5B shows
449 that a pillar positioned ~1.5mm away from the tip of an electromagnet powered by 5A,
450 generates up to 1.5kPa stress in absence of a neighbouring cell, and that NIH3T3 cells also
451 generate up to 1.5kPa when lying close to the same pillar at rest. These values indicate that

452 the forces imposed by the actuation of the magnetic pillar are comparable to the traction
453 forces generated by the cells on the same substrate, which supports the relevance of the
454 method. Furthermore, mapping the stress variations (Fig.5C) both confirms the symmetric
455 pattern of stretching and compression generated by the pillar and recalls that cells compress
456 the surface they adhere on. Stress variations induced by the actuation of a pillar also appear
457 very similar to those produced by a neighbouring cell, with the juxtaposition of regions that
458 are highly compressed (blue) and regions that are highly stretched (yellow).

459 We show that the magneto-active substrates are fully compatible with fluorescence imaging
460 techniques (Fig.6A). As such, the precise location of a cell with respect to a pillar is obtained
461 by combining the fluorescence images of far red fluorescent beads with those of fibroblasts
462 expressing eGFP vinculin. Fig.S7 and Movie S8 reveal that increasing gradually the current in
463 the electromagnets not only deforms the surface but also the adhering cell, and thus
464 demonstrate the possibility to tune the amplitude of mechanical stimulation induced by the
465 active substrate.

466 Other magnetically actuated systems proposed previously^{19,20} were designed to stimulate cells
467 specifically at the focal adhesion level with discrete surfaces. The present method rather
468 addresses larger length scales (from the subcellular to the cellular scales) with displacements
469 spanning continuously over larger distances (see Fig.5B). The actuation is also particularly
470 efficient allowing displacements 5 fold larger than those reported by Sniadecki et al.¹⁹ with 3
471 times less magnetic field. A specificity of this setup is to provide a continuous adhesive
472 surface thereby not restricting the adhesion dynamics and distribution and allowing the cells
473 to freely respond to the stimulation.

474 Finally, we investigated the cellular response to a 5-minute dynamic stimulation at 0.25Hz
475 (Movie S9). Fig.6 shows the fluorescent image of a cell (A) and the outline of the same cell
476 (B) before (cyan), during (yellow) and three minutes after (magenta) the mechanical
477 stimulation. When the pillar is actuated, the cell is deformed with a local maximal
478 displacement of 3.8 μ m in this case, and after stimulation, a new protrusion appears on the
479 opposite side. To quantify the overall increase of protrusive activity observed on several cells,
480 we estimated the normal velocity of the cell boundaries using a method introduced by
481 Döbereiner et. al.³⁴. Velocity profiles along the normalized perimeter of the cell were
482 represented as a function of time as in a kymograph (Fig.6C). In this example, the cell rapidly
483 develops and retracts a protrusion with velocities of +5 μ m/min and -5 μ m/min, respectively,
484 as shown by the red and blue streaks, shortly after mechanical stimulation. The same protocol

485 of mechanical stimulation and data analysis was conducted over 11 different cells, and 4
486 control cells with no mechanical stimulation. We compared the velocity of the most active
487 region along the edge before and after the mechanical stimulation (see Methods). The velocity
488 ratios plotted in Fig.6D show that approximately half of the cells increased significantly their
489 protrusive activity after dynamic mechanical stimulation, while the rest of the population
490 rather compares to the control cells. The random positioning of the cells with respect to the
491 magnetic pillars, which leads to different types of stimulation for each cell, is a current
492 limitation of the system and may explain the heterogeneous responses observed. Indeed, we
493 can speculate that a polarized cell may react differently if pulled at the front or at the back, or
494 along different orientations. Patterning adhesive islands on the substrate appears to be a
495 relevant solution to normalize the cell experiments.

496

497 **Conclusion and Outlook**

498 The magneto-active substrates developed here complement the current cell stretching
499 technologies used to establish that cells change contractility, spreading phenotype, fibre
500 formation and proliferation differently upon various mechanical stimulation patterns^{8,17-19}.
501 Indeed, the present work demonstrates that magneto-active substrates made of soft magnetic
502 micro-pillars embedded in a soft elastomer uniquely combine advantages that were not
503 associated so far. The magneto-active substrates i) have been designed to meet criteria relative
504 to biocompatible materials and optical compatibility with high-resolution fluorescence
505 microscopy, ii) enable to apply a local and controlled mechanical stimulation on single cells
506 spread on a continuous surface, and iii) have an additional potential to quantify cellular
507 mechanical responses via traction force microscopy. Live-cell experiments including a period
508 of dynamic deformation via the magneto-active substrates further support the relevance of this
509 new tool in the study of cell response to mechanical stimulation.

510 Alternative approaches to micro-pillar fabrication (e.g. electro-deposition in patterned
511 moulds) and template preparation (e.g. electro-deposition in patterned moulds or chemical
512 etching of foils) can be explored to improve the reproducibility of actuation from pillar to
513 pillar. The use of hard magnetic micro-pillars, which would be permanently magnetized in a
514 given direction, could also be studied.

515 Patterning adhesive islands on the surface, as done in ^{31,46}, would overcome the limitations
516 related to the positioning and orientation of the cells with respect to the pillars. A strategic
517 functionalization of the magneto-active substrate would then guarantee the repeatability of
518 cell experiments by controlling the degree of spreading and the geometry of the cell. Most
519 importantly, patterning would enable to choose the mode of stimulation (stretching,
520 compression or shear) and the distance to the pillar. This improvement would also constitute a
521 first step toward high throughput experiments. In the long term, we believe that magneto-
522 active substrates have potential to become a standard tool to investigate cell response to
523 dynamic mechanical stimulation and thus improve our quantitative understanding of
524 mechanotransduction.

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635

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648 FIB cutting, respectively.

649

650 **Author contributions**

651 MB and AD contributed to the concept of the magneto-active substrates. CB, PM, ND, TD,
652 and AD designed the experimental setup. MF, ND, TD implemented the magneto-mechanical
653 models and generated the numerical data. TD produced and characterized the magnetic
654 templates and pillars. CB established the protocols and performed the experiments. CB and
655 TB performed the mechanical characterisations. CB, AC, AL, IW and AD contributed to the
656 data analysis. CB, MF, ND, TD and AD drafted the work. All authors approved the final
657 version of the manuscript.

658

659 **Competing financial interests**

660 The authors declare that they have no competing interests.

661

662

663 **Figure legends**

664 Fig.1: Experimental workflow. (A) Magnetic micro-pillars and template are made by optical
665 lithography followed by deep reactive ion etching (DRIE) and iron deposition by sputtering.
666 The magnetic pillars are mechanically detached from the wafer and mixed with soft PDMS
667 prior to casting between a sheet of polypropylene coated with fluorescent beads and a
668 coverslip. (B) The resulting sandwich structure is positioned on a magnetic template laid on a
669 large permanent magnet, so as to align the pillars vertically (side view) and organize them
670 according to the pattern of the template (top view). (C) After peeling off the polypropylene
671 sheet, coating the surface with proteins and plating cells on the top, the substrates are placed
672 in the magnetic field generated by two electromagnets so as to actuate the pillars via a
673 magnetic torque. The actuation setup is mounted on a microscope to quantify the local
674 deformations of the surface by tracking the fluorescent beads and to follow the response of the
675 cells to the mechanical stimulation.

676 Fig.2: Characterization. (A) Mechanical profiles of the magneto-active substrate measured
677 around 5 pillars by atomic force microscopy reveal homogeneous PDMS substrates with local
678 increases of the Young modulus above the pillars. (B) The horizontal component of the
679 magnetic field B_x increases with the current input to the electromagnets and reaches 100mT
680 for 5A (*) at the electromagnet tip. (C) A numerical model in 2D evaluates the distribution of
681 the horizontal and vertical components of the magnetic field (B_x and B_z respectively) between
682 the bevel shaped cores of the electromagnets.

683 Fig.3: Contribution of B_z . The magnetization of the iron pillar induced by a purely horizontal
684 magnetic field (A), a magnetic field with a slight (B) or large (C) vertical component B_z and
685 the subsequent magnitude of displacement in the substrate have been predicted with the
686 magneto-mechanical model.

687 Fig.4: Actuation of the substrate. (A) The magnitude of displacement induced by a pillar
688 positioned 1mm away from the electromagnet and experiencing $B_x = 119\text{mT}$ and $B_z = 27\text{mT}$
689 was estimated with a 3D model and compared to the magnitude of displacement measured
690 experimentally around 5 pillars stimulated by an electromagnet supplied with 5A. (B) Maps
691 of stress magnitude were derived from the displacement fields by Fourier Transform Traction
692 Cytometry, and (C) maps of stress variation were calculated to distinguish the regions under
693 traction and compression. Scale bar: $30\mu\text{m}$. Profiles corresponding to the dashed region of
694 each map are also displayed. (D) Maximum magnitude of displacement measured on 4

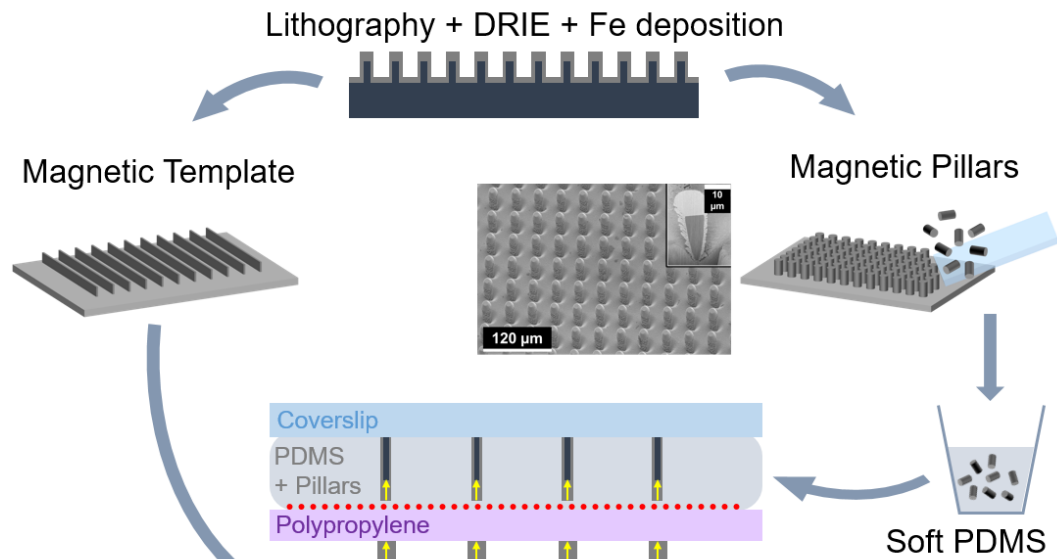
695 different pillars undergoing cyclic stimulations applied by manually adjusting the intensity of
696 the current input (red curve).

697 Fig.5: Cells on magneto-active substrates. (A) Bright field images of NIH3T3 fibroblast cells
698 spread on magneto-active substrate. Stress magnitude (B) and variation (C) experienced by
699 the surface were measured under the action of a pillar without cell (5A) and in the presence of
700 contractile cells in the vicinity of a pillar at rest. Scale bar: 50 μ m.

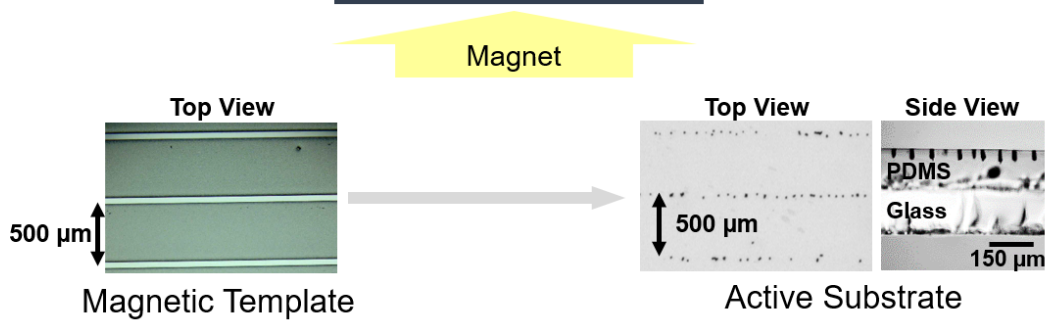
701 Fig.6: Cells response to stimulation. (A) Fluorescent images of a NIH3T3 vinculin-eGFP
702 fibroblast (green) and the beads spread under the surface (red) were used to draw the contours
703 of the cell and the pillar respectively (B). Scale bar: 10 μ m. The cell is deformed (yellow)
704 when the pillar is displaced, and a new protrusion appears (magenta) a few minutes later.
705 Corresponding movie as supplementary material (C) Velocity profiles of the cell boundary
706 along a normalized perimeter as a function of time. Positive values (shown in red) represent
707 protrusions whereas negative values represent retractions. The protrusion is visible (magenta
708 arrow) shortly after the end of the stimulation and followed by a retraction. (D) Quantification
709 of the cellular response for 11 stimulated cells and 4 control cells. Velocity ratio between,
710 after and before the stimulation were calculated and plotted for each cell. Approximately half
711 of the cell population shows an increased protrusion activity after the mechanical stimulation.

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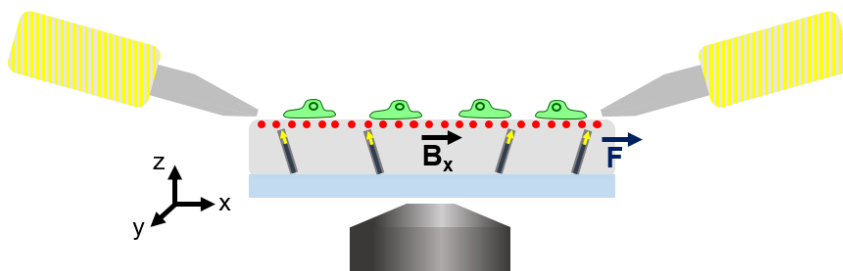
A. Microfabrication



B. Assembly



C. Actuation

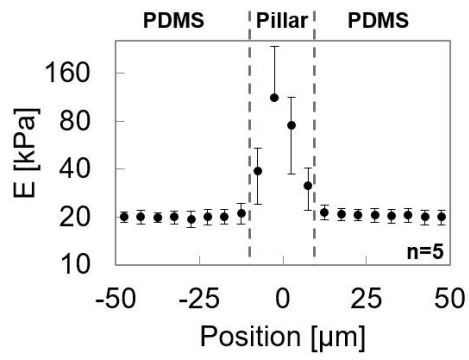


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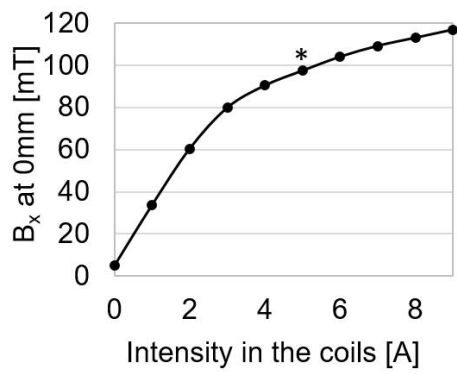
714 Figure 1

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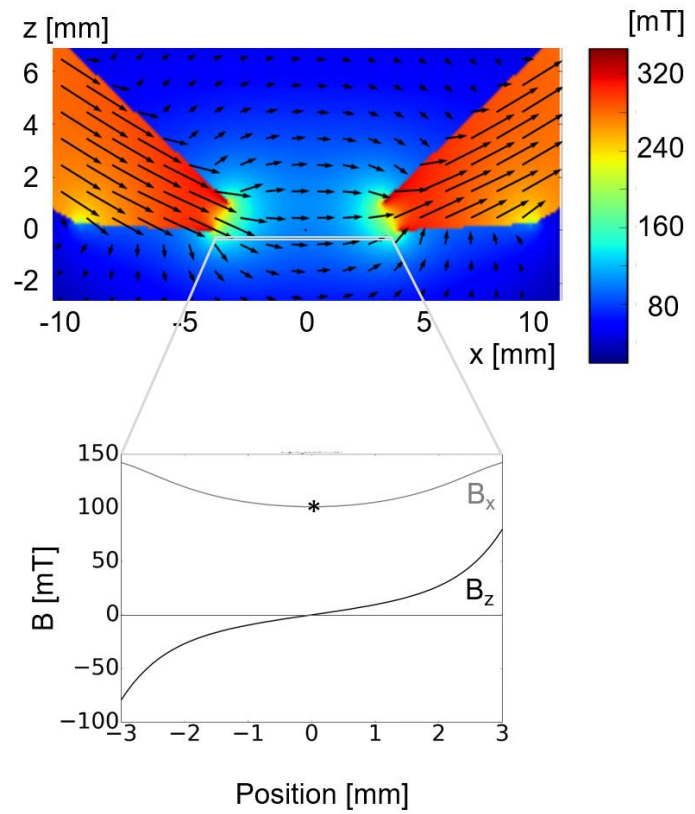
A. Mechanical profile (Experiment)



B. Magnetic Field (Experiment)



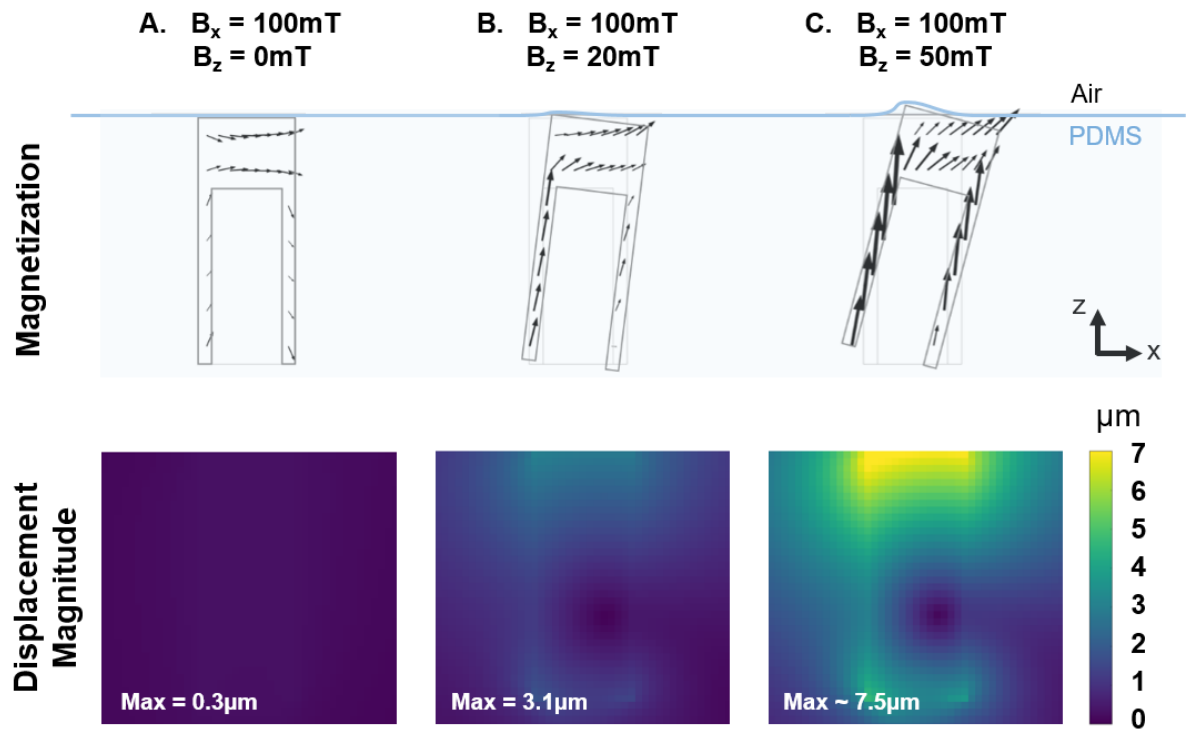
C. Profile of the magnetic field (2D simulation)



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717 Figure 2

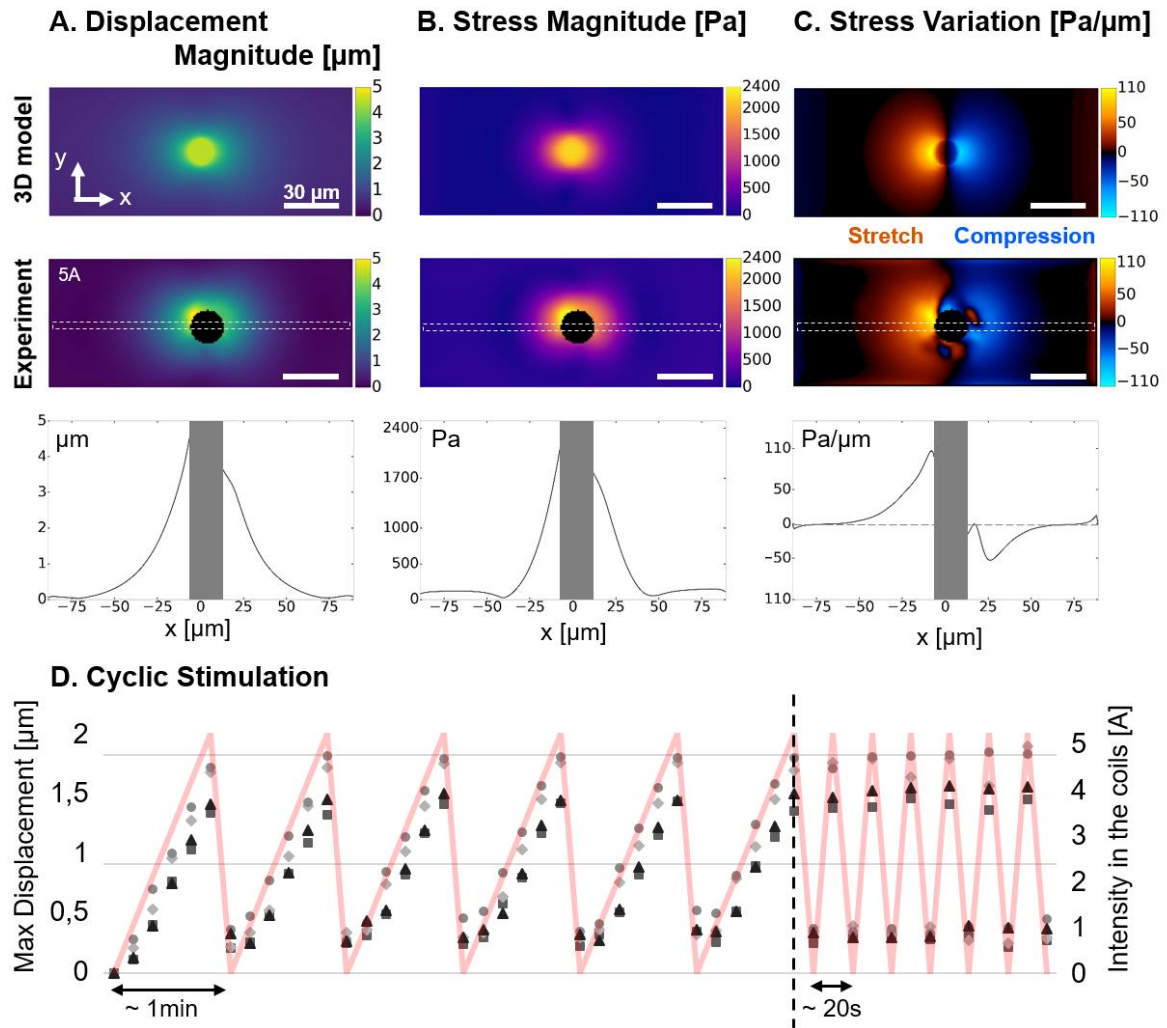
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720 Figure 3

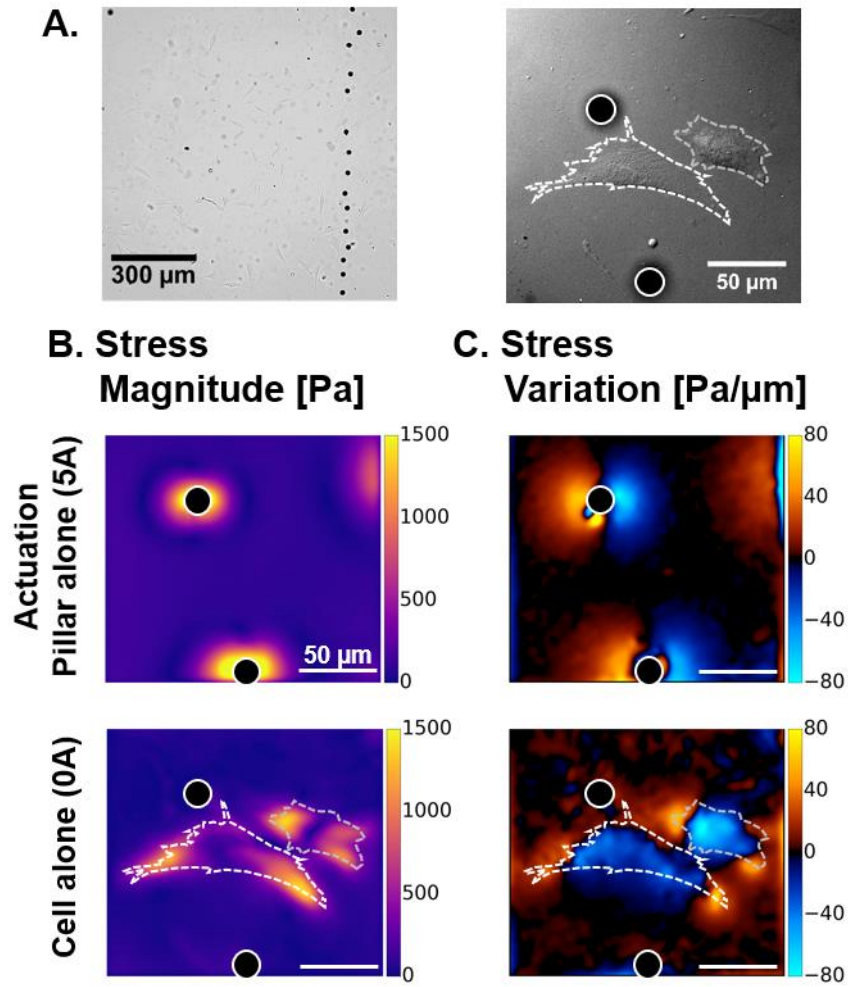
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723 Figure 4

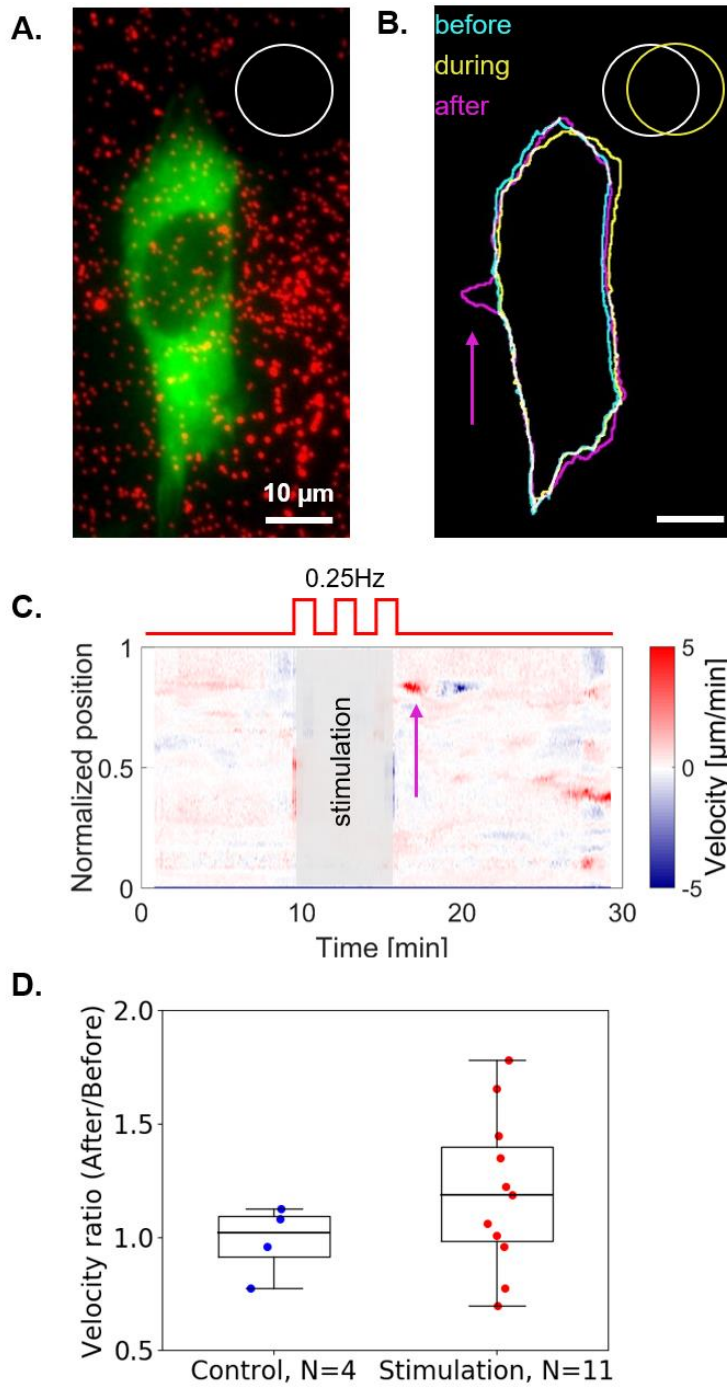
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726 Figure 5

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728

729 Figure 6