

1 **Title: Mechanical matching of implant to host minimises foreign body reaction**

2

3 **One sentence summary:** Foreign body reaction to medical implants can be avoided by  
4 matching the stiffness of the implant surface to that of the host tissue.

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27

28 **Abstract**

29 **Medical implants offer a unique and powerful therapeutic approach in many areas of**  
30 **medicine. However, their lifetime is often limited as they may cause a foreign body**  
31 **reaction (FBR) leading to their encapsulation by scar tissue<sup>1-4</sup>. Despite the importance**  
32 **of this process, how cells recognise implanted materials is still poorly understood<sup>5,6</sup>.**  
33 **Here, we show how the mechanical mismatch between implants and host tissue leads**  
34 **to FBR. Fibroblasts and macrophages, which are both crucially involved in mediating**  
35 **FBR, became activated when cultured on materials just above the stiffness found in**  
36 **healthy tissue. Coating implants with a thin layer of hydrogel or silicone with a tissue-**  
37 **like elastic modulus of ~1 kPa or below led to significantly reduced levels of**  
38 **inflammation and fibrosis after chronic implantation both in peripheral nerves and**  
39 **subcutaneously. This effect was linked to the nuclear localisation of the**  
40 **mechanosensitive transcriptional regulator YAP *in vivo*. Hence, we identify the**  
41 **mechanical mismatch between implant and tissue as a driver of FBR. Soft implant**  
42 **coatings matching the mechanical properties of host tissue minimized FBR and may**  
43 **be used as a novel therapeutic strategy to improve long-term biomedical implant**  
44 **stability without extensive modification of current implant manufacturing techniques,**  
45 **thus facilitating clinical translation.**

46

47

48 Introduction

49 Medical implants have become an indispensable tool for a wide range of applications,  
50 including bladder control<sup>11</sup>, treatment of neurological disorders<sup>12</sup>, drug delivery<sup>13</sup>, and tissue  
51 repair<sup>14</sup>. Recent advances in fabrication techniques have allowed for the development of  
52 increasingly complex implants, capable of better integrating into the host tissue and much  
53 improved functionality. This is particularly visible in the field of neural interfaces, where  
54 implants are capable of establishing electrical connections with individual axons and neurons  
55 without disrupting their connectivity<sup>15-17</sup>. Functionality of such implants actively interacting  
56 with their environment requires the establishment and maintenance of intimate interfaces  
57 between implant and tissue.

58 However, this interface and thus long-term functionality of medical implants is often limited  
59 by foreign body reaction (FBR) – a process by which the body recognises implanted  
60 materials as foreign and attempts to degrade them. FBR is characterised by chronic  
61 inflammation and fibrosis, and with time results in the formation of a scar – a fibrotic capsule  
62 – separating the implant from the host tissue<sup>5</sup>. The breakdown of the tissue-implant interface  
63 is one of the leading causes of implant failure<sup>1-3</sup>, which has led to efforts to develop a  
64 treatment to prevent and manage FBR to implanted materials<sup>18-22</sup>. Impregnation of implants  
65 with anti-inflammatory drugs such as dexamethasone is currently used in clinical practice to  
66 alleviate inflammatory reactions<sup>23</sup>, and suppresses FBR in nerve interfaces<sup>24</sup>. However,  
67 such strategies can have significant side effects<sup>24</sup>. Furthermore, how the body recognises  
68 implanted materials as foreign and what triggers the associated inflammatory response is  
69 still poorly understood<sup>5,6</sup>.

70 Medical implants are usually prepared from materials compatible with miniaturisation,  
71 surgical handling and advanced 3D design. These materials are much stiffer than their  
72 biological host tissues. Shear moduli, a measure of a material's elastic stiffness, of such  
73 implants are usually on the order of hundreds of kPa and above, while most biological

74 tissues have shear moduli of few kPa<sup>25,26</sup>. While recent studies suggested that softer  
75 materials might alleviate inflammation<sup>7-10</sup>, a direct link between cellular  
76 mechanotransduction and FBR has not been shown yet. Furthermore, how soft materials  
77 actually need to be to circumvent FBR is currently not clear, and techniques to manufacture  
78 implants using very soft materials are not readily available.

79

## 80 Results

81 We here investigated whether cell types crucial for FBR in many tissue types, namely  
82 primary macrophages and fibroblasts, indeed respond to differences in the stiffness of their  
83 environment within a physiologically relevant scale. Macrophages are a central component  
84 of innate immunity and are responsible for driving the inflammatory response to implanted  
85 materials<sup>5,6</sup>. Fibroblasts, on the other hand, are the primary mediators of the fibrotic  
86 response, responsible for the formation of the fibrotic capsule around the implant.

87 Most macrophages involved in FBR are derived from blood-circulating monocytes, while  
88 fibroblasts proliferate from tissue-resident populations<sup>5,6</sup>. We cultured primary bone marrow-  
89 derived macrophages and fibroblast populations derived from peripheral nerve tissue on  
90 polyacrylamide substrates of a range of shear moduli. Polyacrylamide substrates are well-  
91 established assays frequently used to assess cellular responses to the mechanical  
92 properties of their environment. We chose nerve fibroblasts because of the high impact of  
93 FBR on peripheral nerve interfaces. While implants with shear moduli of few hundreds of  
94 kPa are commonly considered 'soft'<sup>27-29</sup>, shear moduli of our substrates ranged from 0.1  
95 kPa, mechanically resembling the softest tissues in the body including peripheral nerve  
96 tissue (Supplementary Fig. 1), to 50 kPa, which is already stiffer than most soft tissues and  
97 organs<sup>26</sup> (substrate stiffness measurements are shown in Supplementary Fig. 2).

98 FBR-induced scarring is particularly detrimental to electrical neural interfaces, such as those  
99 implanted in nerves. We therefore initially focused on this tissue. Nerve fibroblasts cultured  
100 on substrates of varying stiffness for 6 days retained a spherical morphology on soft gels  
101 with shear moduli of ~0.1-1 kPa, while cell spreading significantly increased on stiffer  
102 substrates with shear moduli of 10-50 kPa (Fig. 1b) ( $p = 1.2e-7$ , one-way ANOVA). Adhesion  
103 to the substrate via focal adhesions (Fig. 1c) and proliferation (Supplementary Fig. 3)  
104 showed similar significant increases with substrate stiffness (adhesion  $p = 5.2e-9$ ,  
105 proliferation  $p = 0.0023$ ). These observed increases in cell spreading, adhesion, and  
106 proliferation are consistent with a fibroblast FBR-like phenotype<sup>30</sup>.

107 Fibroblasts in FBR and other fibrotic processes typically differentiate into myofibroblasts –  
108 which are characterised by the expression of alpha smooth muscle actin ( $\alpha$ SMA) and high  
109 extracellular matrix production<sup>31</sup>. Exposure to the non-physiological high stiffness of 50 kPa  
110 significantly increased the synthesis of both  $\alpha$ SMA and the extracellular matrix protein  
111 collagen I – a primary component of the FBR capsule (Fig. 1d-f) (collagen  $p = 1.4e-26$ ,  
112  $\alpha$ SMA  $p = 5.3e-26$ , one-way ANOVAs). Fibronectin - a different extracellular matrix protein –  
113 was present in nerve fibroblasts but was largely independent of substrate stiffness  
114 (Supplementary Fig. 3) (no significant differences in multiple comparison analysis between  
115 50 kPa and 10 kPa or 0.1 kPa). Our findings are consistent with previous studies in fibroblast  
116 populations from other tissues reporting that fibroblasts transition into a myofibroblast  
117 phenotype at high substrate stiffnesses<sup>31,32</sup>.

118 Macrophages showed similar functional changes on substrates stiffer than their physiological  
119 host tissue. Similar to nerve fibroblasts, cell spreading, adhesion, and proliferation rates  
120 were significantly increased if compared to softer substrates (Fig 2a-c, Supplementary Fig.  
121 4) (spreading  $p = 9.8e-8$ , adhesion  $p = 0.009$ , proliferation  $p = 0.007$ , one-way ANOVA).  
122 RNA sequencing of macrophages three days post-plating showed significant substrate  
123 stiffness-dependent changes in gene expression, particularly in inflammation-related genes

124 (Fig. 2d), consistent with a switch in phenotype from activated M1 to the anti-inflammatory  
125 M2 on stiffer substrates. On 50 kPa substrates, we found downregulation of proinflammatory  
126 genes such as interleukin 1 beta (*il1b*) ( $p = 1.4e-6$ , FDR adjusted p-value) and  
127 prostaglandin E synthase (*ptgs2*) ( $p = 9.5e-6$ ) in macrophages, while M2-associated genes  
128 such as arginase (*arg1*) ( $p = 0.002$ ) and peroxisome proliferator-activated receptor- $\gamma$  (*ppar $\gamma$* )  
129 ( $p = 0.007$ ) were upregulated if compared to soft substrates. This switch to a macrophage  
130 M2 activation profile is typically associated with tissue regeneration and fibrosis<sup>33,34</sup>.  
131 Together, our *in vitro* experiments indicated that non-physiologically high substrate stiffness  
132 may lead to fibrosis, a hallmark of FBR.

133 Having confirmed that both primary macrophages and nerve fibroblasts assume an FBR-like  
134 phenotype when exposed to materials that are stiffer than their host tissues, we sought to  
135 test if materials mechanically matched to these tissues may alleviate FBR *in vivo* in rat  
136 models. While it is often difficult or impractical to fabricate and use implants made entirely  
137 from extremely soft materials because of fabrication and handling challenges, stiff materials  
138 can be masked from cells underneath a thick enough layer of a softer material<sup>35-37</sup>. Hence,  
139 we designed silicone rubber implants with a shear modulus of ~ 200 kPa which we coated  
140 with a 100  $\mu$ m thick layer of soft material (Fig. 3a) to “stealthen” the underlying stiffer  
141 substrate from cells (see Methods for details)<sup>35-37</sup>. The coatings consisted of either soft 0.2  
142 kPa polyacrylamide (PAA\_0.2kPa), 2kPa silicone (PDMS\_2kPa), or 20 kPa polyacrylamide  
143 (PAA\_20kPa) (shear moduli, Supplementary Fig. 5). One group of implants remained non-  
144 coated (PDMS\_200kPa).

145 These soft-coated devices were implanted first into the subcutaneous space of rats, a  
146 common location for medical implants such as pulse generators<sup>38</sup> and biosensors<sup>17</sup>. Three  
147 months post-implantation, a time point by which acute inflammation due to implantation has  
148 resolved and chronic responses to implanted materials have settled in, FBR was significantly  
149 reduced around the implants with soft coatings if compared to the stiffer materials (Fig. 3b)

150 as revealed by immunohistochemistry. The intensities of markers for myofibroblasts ( $\alpha$ SMA),  
151 collagen I, and macrophages (CD68) were significantly lower in tissues surrounding soft  
152 materials, with greater effects seen at the lowest stiffnesses (Fig. 3c) ( $\alpha$ SMA  $p = 0.005$ ,  
153 collagen  $p = 6.0e-4$ , CD68  $p = 3.4e-5$ , one-way ANOVAs). Similar to our *in vitro* results, the  
154 extracellular matrix protein fibronectin did not vary significantly across groups  
155 (Supplementary Fig. 6) ( $p = 0.14$ , one-way ANOVA). On the other hand, capsule thickness,  
156 a measure of fibroblast proliferation around the implant and of the severity of FBR, was  
157 greatly decreased around implants with soft coatings (Fig. 3d, Supplementary Fig. 7) ( $p =$   
158  $0.006$ , one-way ANOVA), suggesting that FBR can be minimized by matching the  
159 mechanical properties of the implant surface to those of the surrounding tissue.

160 To test if the suppression of FBR by implants with soft coatings is a tissue type-specific  
161 phenomenon or if it is more generally applicable, we then implanted nerve conduits with or  
162 without soft coatings as those described above in a rat model of nerve injury (Fig. 4a) (for  
163 details on sciatic nerve transection see Methods). Similar to the subcutaneous implants,  
164 intensity of markers for myofibroblasts ( $\alpha$ SMA) and macrophages (CD68) were significantly  
165 decreased in tissue exposed to the soft coating compared to the non-coated control implants  
166 (Fig. 4c) ( $\alpha$ SMA  $p = 0.002$ , CD68  $p = 0.007$ , one-way ANOVAs). Collagen I also showed a  
167 similar trend, although differences were not statistically significant (Fig. 4c) ( $p = 0.19$ , one-  
168 way ANOVA), while capsules became significantly thinner around softer materials (Fig. 4d)  
169 ( $p = 0.0008$ , one-way ANOVA), indicating that soft coatings of implants may indeed  
170 represent a general approach to alleviate FBR to biomedical implants irrespective of the type  
171 of host tissue.

172 To benchmark the performance of our soft-coated implants against currently used clinical  
173 strategies exploiting devices impregnated with chemical repressors<sup>23,24</sup> of inflammatory  
174 reactions such as glucocorticoids, we repeated the experiments using implants coated with a  
175  $\sim 100 \mu\text{m}$  thick dexamethasone-impregnated silicone of  $\sim 200 \text{ kPa}$  (Dex)<sup>24</sup>.  $\alpha$ SMA,



176 macrophage, and collagen I levels indicated a similar suppression of FBR in Dex  
177 subcutaneous implants and nerve conduits as seen in implants whose coating was similarly  
178 soft as the host tissue (Fig. 3c, 4c) ( $p > 0.05$  for all stains in both types of implants between  
179 PDMS\_2kPa & Dex and PAA\_0.2kPa & Dex, Bonferroni-corrected Student's t-test).  
180 However, as glucocorticoids such as dexamethasone are not only anti-inflammatory but also  
181 anti-proliferative, neuronal regeneration was significantly reduced in dexamethasone-doped  
182 nerve implants if compared to implants with soft coatings (Fig. 4b, e, Supplementary Fig. 8)  
183 ( $p = 0.003$ , one-way ANOVA). Hence, our data showed that soft coatings of implants not  
184 only suppress inflammation and FBR but also permit regeneration – in contrast to currently  
185 exploited devices relying on the anti-inflammatory properties of glucocorticoids.

186 In FBR and fibrosis, the secretion of dense networks of extracellular matrix rich in  
187 components such as collagen I leads to an increase in the stiffness of the tissue surrounding  
188 the implant<sup>39,40</sup>. To further corroborate the suppression of FBR in tissue surrounding soft  
189 coating implants, we used *ex vivo* atomic force microscopy to measure the apparent elastic  
190 moduli of tissues exposed to the different implants (Fig. 4f). While, with a median apparent  
191 elastic modulus  $K \sim 1200$  Pa, tissue around the stiffest conduits showed a large degree of  
192 stiffening compared to intact nerve epineurium, tissue around all soft-coated implants  
193 showed a low  $K \sim 150$  Pa, indistinguishable from dexamethasone-treated controls ( $p = 0.70$ ;  
194 Bonferroni-adjusted Student's t-test) and similar in value to intact nerve epineurium ( $K = 140$   
195 Pa) (Fig 4g), further indicating that soft implant coatings minimised the development of FBR.

196 Both our *in vivo* and *in vitro* results indicated that FBR can occur as a consequence of a  
197 mechanical mismatch between the native biological tissue and foreign materials. To test if  
198 cells indeed responded to the mechanical properties of their environment, we investigated  
199 the distribution of the transcriptional regulator YAP in tissue surrounding implants after 3  
200 months. In many systems, YAP is majorly involved in mechanotransduction, i.e., the  
201 conversion of mechanical cues into biochemical signals<sup>41</sup>. On soft substrates, YAP is usually  
202 excluded from the nucleus, while on stiffer substrates YAP enters the nucleus, leading to

203 mechanically driven changes in gene expression<sup>42</sup>. This effect has been observed *in vitro* in  
204 several cell types including fibroblasts<sup>43</sup> as well as *in vivo*<sup>44</sup>. Immunohistochemical stains  
205 indeed revealed a significantly lower nuclear localisation of YAP in tissues exposed to softer  
206 coatings if compared to stiffer coatings in both implant types (Fig. 5) (subcutaneous  $p = 0.01$ ,  
207 nerve  $p = 0.001$ , one-way ANOVAs), confirming that cells in the vicinity of medical implants  
208 respond to the stiffness of the implant material.

209

## 210 Discussion

211 Our results are consistent with previous *in vitro* studies showing that cells respond to the  
212 stiffness of their environment<sup>7,45–48</sup>, and they link this cellular mechanosensitivity to a key  
213 problem in medical implants: FBR. Contact with materials stiffer than native tissue leads to  
214 trans-differentiation of fibroblasts into myofibroblasts, and macrophages taking on an  
215 activated M2-like phenotype, driving tissue generation.

216 Collectively, our results show that a mechanical mismatch between implant material and  
217 host tissue stiffness is a primary driver of FBR, and that this principle can be exploited to  
218 minimise FBR by manufacturing mechanically soft implant coatings with a stiffness similar to  
219 the host tissue. Notably, this effect was most pronounced for coatings with shear moduli  
220  $< 10$  kPa, which is orders of magnitude softer than materials commonly used in medical  
221 research that are often referred to as “soft implants”<sup>27–29</sup>. This effect was similar for two  
222 different soft materials: polyacrylamide and PDMS, and it occurred irrespective of implant  
223 design and the location of implantation, indicating a very robust response of host tissues to  
224 implant stiffness.

225 Current medical implants offer a powerful tool for the treatment of a number of clinical  
226 conditions. However, long term stability of implants often remains limited by FBR, particularly  
227 of those actively interacting with the host’s environment such as electrical neural interfaces.

228 Corticosteroid drugs such as dexamethasone offer a viable strategy to control FBR in certain  
229 situations<sup>23</sup>. However, their anti-inflammatory effect also greatly interferes with regeneration  
230 of surrounding tissue<sup>24</sup>, making them ill-suited for use in regenerative implants. As shown  
231 here, soft coatings offer an effective strategy to minimise FBR without impacting surrounding  
232 tissue function or sacrificing the bulk mechanical properties of the implant itself. The coating  
233 dimensions and compatibility with existing microfabrication techniques make this technology  
234 easily applicable to many implant designs, facilitating translation to the clinic. Moreover, the  
235 FBR-reducing effects are linked to the mechanical – and not chemical – properties of the  
236 coating materials, providing great flexibility in material choice for different implant designs  
237 and applications.

238 Materials and Methods

239 **Polyacrylamide cell culture substrates.**

240 Polyacrylamide hydrogels were prepared two days prior to the plating of any cells, using a  
241 protocol previously described<sup>49</sup>. 19 mm diameter glass coverslips were cleaned by alternate  
242 dipping in ddH<sub>2</sub>O and EtOH, and covered in NaOH for 5 min. NaOH was removed, and  
243 coverslips were functionalised with APTMS solution for 2.5 min, followed by thorough rinsing  
244 in water. Coverslips were finally allowed to sit in a glutaraldehyde 0.5% (v/v) solution in  
245 ddH<sub>2</sub>O for 30 min at RT.

246 Polyacrylamide premixes were prepared by mixing of acrylamide (40% w/w; A4058, Sigma),  
247 bis-acrylamide (2%; BP1404-250; Fisher Scientific), and hydroxy-acrylamide (97%; 697931;  
248 Sigma) solutions (177:100:23 ratio). A volume of PBS was added to the premix to achieve  
249 gels of a particular stiffness (Supplementary Table 1). This final gel mix was degassed under  
250 a vacuum for 10 min.

251 To initiate polymerisation, 5 µl of ammonium persulfate solution (0.1 g/ml in ddH<sub>2</sub>O; Sigma,  
252 281778) and 1.5 µl of TEMED (15524-010, Invitrogen) were added to 500 µl of gel mixes. 8  
253 µl drops of mix were placed on the treated surface of 19 mm diameter coverslips, and  
254 covered with a 22 mm diameter glass coverslip (previously treated with a RainX hydrophobic  
255 coating).

256 The gels were allowed to swell in PBS overnight. 22 mm diameter coverslips were then  
257 removed to reveal the hydroxyacrylamide gels bound to 19 mm coverslips. These were  
258 sterilised under UV light for 1 hour, and functionalised with PDL (100 µg/ml in PBS) at room  
259 temperature overnight. Gels used for Schwann cell cultures were further functionalised with  
260 laminin (1 µg/ml in PBS) for 2 hours at room temperature. Prior to cell plating, gels were  
261 placed in culture medium for 30 min to allow medium to fill them.

262

263 ***In vitro* assay. Nerve fibroblasts.** Cultures were prepared from postnatal day 1 to 5 Sprague  
264 Dawley rat sciatic nerves. Using a variation of the procedure previously described<sup>50</sup>. All  
265 animal procedures carried out were in compliance with the United Kingdom Animals  
266 (Scientific Procedure) Act of 1986 and institutional guidelines.

267 Sciatic nerves of 10 - 20 animals were dissected out using sterilised microscissors and fine  
268 forceps, and were kept in chilled HBSS (14170-112, Invitrogen) to stabilise the pH and  
269 osmotic environment. To dissociate the tissue, nerves were transferred to a 2 ml  
270 collagenase solution (2 mg/ml; C9407, Sigma) and incubated for 30 min at 37 °C, after which  
271 2 ml of trypsin (1 mg/ml; T0303, Sigma) was added (20 min, 37 °C incubation). Finally, 2 ml  
272 of deoxyribonuclease (0.1 mg/ml; D5025, Sigma) was added and, after a brief incubation  
273 period (2 min), the cells were centrifuged (4 min, 1000 rpm). The supernatant was removed,  
274 and the cell pellet was re-suspended in 2 ml of triturating solution (containing 10 mg/ml  
275 bovine serum albumin [A7906, Sigma], 0.5 mg/ml trypsin inhibitor [10109886001, Roche],  
276 0.02 mg/ml deoxyribonuclease).

277 To isolate the population of nerve fibroblasts from the dissociated nerves, the cells were  
278 centrifuged and re-suspended in 0.5 ml of DPBS/BSA (Dulbecco's phosphate-buffered  
279 saline supplemented with 5 mg/ml bovine serum albumin) and 50 µl of magnetic-bead  
280 antibodies against rat/mouse CD90.1 (Thy1.1) (Miltenyi Biotec, 120-094-523) – a marker  
281 expressed by nerve fibroblasts<sup>51</sup> - and incubated for 15 min at RT. Cells were centrifuged,  
282 re-suspended in 2 ml of chilled DPBS/BSA, and run through a column equipped with a  
283 magnetic separator (MiniMACS Separator; Miltenyi Biotec, 130-042-102). Once the buffer  
284 had finished running through the column, and the flow-through collected, the column was  
285 removed from the magnetic separator, and the magnetically-labelled cells were flushed out  
286 with chilled DPBS/BSA. Finally, the positive fraction (nerve fibroblasts) were centrifuged and  
287 the cell pellet re-suspended in DMEM (11320-033, Invitrogen) supplemented with a further 4

288 mM of glutamine (25030032, Invitrogen), 100 mg/ml foetal calf serum (FCS, Invitrogen) and  
289 an antibiotic-antimycotic agent (15240-062, Invitrogen). The cells were plated on  
290 polyacrylamide substrates at a density of 10,000 cells/cm<sup>2</sup>.

291 *Bone marrow-derived macrophages.* Macrophages were derived from adult rat bone marrow  
292 hematopoietic stem cells as previously described<sup>52</sup>. Adult Sprague Dawley rats were  
293 sacrificed by exposure to a rising concentration of CO<sub>2</sub>. The femurs were dissected out and  
294 broken open using sterile scissors. Bone marrow contained within the femurs was washed  
295 out with chilled DPBS and collected. The cell suspension was centrifuged for 10 min at 1000  
296 rpm. The supernatant was discarded and the cells were re-suspended in BMDM (the same  
297 supplemented DMEM used for nerve fibroblasts, further supplemented with macrophage  
298 colony stimulating factor [400-28, Peprotech; 50 ng/ml]). Cells were counted in a  
299 hemocytometer and seeded on 100 x 15 mm uncoated petri dishes (Sigma, P5731) at a  
300 density of 5,000 cells/cm<sup>2</sup>. The dishes were supplemented with further medium after 3 days  
301 of culture at 37 °C.

302 Hematopoietic stem cells differentiate into macrophages in the presence of macrophage  
303 colony stimulating factor present in BMDM. After 6 days to allow differentiation of  
304 Hematopoietic stem cells differentiate into macrophages, the medium in the dishes was  
305 removed to dispose of any cells which had not attached to the substrate. The remaining cells  
306 were washed with warm DPBS followed by CellStripper solution (Corning, 25-056-CI). Cells  
307 were incubated in CellStripper for 5 min at 37 °C to detach them from the substrate. The cell  
308 suspension was collected and an equal volume of BMDM was added to inactivate the  
309 CellStripper solution. Cells were then centrifuged at 1000 rpm for 10 min and the  
310 supernatant was removed. Upon re-suspension in BMDM and counting, cells were plated  
311 onto polyacrylamide substrates at a density of 10,000 cells/cm<sup>2</sup>.

312

313 **Immunocytochemistry.**

314 Cell stains were carried out 6 days after plating on polyacrylamide substrates. At this point in  
315 time, warm paraformaldehyde solution (40 mg/ml in PBS) was added to the cells for 15 min  
316 at room temperature. The fixative solution was washed off with PBS (3 washes, 10 min per  
317 wash). To improve antibody specificity cells were incubated for 30 min at room temperature  
318 in a blocking solution consisting of 0.03% v/v Triton X-100 (Sigma, T8787) and 3% v/v  
319 bovine serum albumin (Sigma, A9418) in PBS. Primary antibodies (in blocking buffer) were  
320 then added to cells and incubated overnight at 4 °C. Further details regarding antibody  
321 concentrations can be found in Supplementary Table 2.

322 Excess primary antibodies were washed off using PBS (3 washes, 10 min). Secondary  
323 antibodies in blocking buffer were incubated on the cells for 2 hr at room temperature.  
324 Following a wash with non-saline Tris-buffered solution, Fluorsave mounting agent (Millipore,  
325 345789) was added to sections to preserve fluorescence before gels were placed onto glass  
326 slides and stored at 4 °C prior to imaging.

327 Imaging of stained cells on polyacrylamide substrates was carried out using a confocal  
328 microscope (Leica TCS SP5). For every condition 3 images were taken at random sites  
329 within each gel. The nuclear stain was used as a guide to ensure cells were present in the  
330 field of view when pictures were taken. Gain and exposure settings for each channel used  
331 were maintained constant between imaging sessions and across different stains.

332 Cell counts were performed by hand in the Image-J software package (v1.48, National  
333 Institutes of Health, USA). Contrast of images was modified prior to analysis. For  
334 morphological stains, one image of high stiffness (50 kPa) and one of low stiffness (0.1 kPa)  
335 were opened and their contrast modified to an equal degree until a satisfactory pattern of  
336 stain was achieved in both. This contrast modification was then applied to all images of the  
337 same batch of stained gels. For cell-type specific stains such as alpha-smooth muscle actin  
338 this same contrast modification was carried out using negative and positive control stains.

339 Cells were counted, or the area stained was designated, by hand. Statistical analysis and  
340 data plotting was carried out using MATLAB (Mathworks, R2016b).

341

#### 342 **RNA sequencing.**

343 RNA sequencing was performed in parallel on n = 4 biological replicates. RNA extraction  
344 was carried out at day 3 of culture on polyacrylamide substrates using an RNeasy Plus  
345 Micro Kit (Qiagen, 74034). Prior to and at regular intervals during the extraction procedure,  
346 work surfaces and pipettes were cleaned with RNase Zap decontamination solution  
347 (ThermoFisher, AM9780) to inactivate RNases and prevent sample RNA degradation. To  
348 collect the cells, each coverslip was briefly washed in PBS and lifted out of the solution using  
349 forceps. The gels onto which the cells were attached were gently scraped off from the  
350 coverslips using a sterile steel blade and placed in RLT lysis buffer plus (Qiagen). The  
351 samples in buffer were then moved to QIAshredder tubes (Qiagen, 79654) and centrifuged  
352 in a microcentrifuge (MSE, mistral 1000) for 2 min at 8,000g. Instructions provided by the kit  
353 manufacturer were then followed to extract cellular RNA, which was collected and stored at -  
354 80°C.

355 RNA quantification and integrity analysis were carried out on all samples prior to library  
356 preparation. Using an RNA 6000 Pico Kit (Agilent, 5067-1513), samples concentration and  
357 integrity was analysed using an Agilent 2100 Bioanalyzer. No samples with an RNA integrity  
358 number <7 were used. Library preparation was thereafter carried out using an Ovation RNA-  
359 Seq System V2 kit (NuGen, 7102-32), following manufacturer instructions. Samples were  
360 finally submitted for sequencing in an Illumina HiSeq 2500 system.

361 Illumina read data files were run through a bioinformatics pipeline and aligned with the  
362 *Rattus norvegicus* genome (Ensembl Rnor\_6.0). Fold changes and p-values were calculated  
363 for each gene between each experimental group and a control group (consisting of the



364 combined 0.1 kPa and 1 kPa conditions). Genes expressed were filtered to produce lists of  
365 differentially expressed genes (DEGs). Defined by a minimum of 2-fold change in  
366 expression, a base expression above 3 normalised counts, and an adjusted p-value below  
367 0.05. Principal component analysis was carried out on data from these experiments  
368 (Supplementary Fig. 9). RNAseq data have been deposited in the ArrayExpress database at  
369 EMBL-EBI (<https://www.ebi.ac.uk/arrayexpress/experiments/E-MTAB-7900>) under accession  
370 number E-MTAB-7900. Code to create the figures displaying RNAseq results is available in  
371 the following GitHub repository: <https://github.com/CTR-BFX/2019-Carnicer-Lombarte>.

372

### 373 **Implant fabrication.**

374 *Nerve conduits.* Moulds of the cuff implants were designed in 3D-CAD software (AutoCAD,  
375 Autodesk Inc) and 3D printed in PLA (polylactic acid) plastic using a MakerGear M2 3D  
376 printer (MakerGear). The moulds were covered in Sylgard 184 PDMS, and were placed in an  
377 oven at 65°C overnight. The 3D printed mould was removed from the cured PDMS,  
378 producing a negative pattern of the cuff implants. The surface of the PDMS negative moulds  
379 was activated using oxygen plasma (Diener plasma etcher) for 25 seconds at 30 W, and 0.8  
380 mbar chamber pressure. PDMS moulds were then functionalised using  
381 Trichloro(1H,1H,2H,2H-perfluorooctyl)silane (Sigma, 448931). A few drops of silane were  
382 placed on a glass petri dish and into a desiccator together with the PDMS moulds. The  
383 desiccator was pumped down into a vacuum, and functionalisation was allowed to take place  
384 overnight. The resulting layer of silane prevented any new PDMS cured on these moulds  
385 from binding to them, allowing for the casting of the PDMS cuffs from these negative moulds.

386 To cast the cuffs, flat petri dishes with raised edges were prepared. These edges were  
387 produced through layering multiple layers of insulation PVC tape, until a thickness of 0.6 mm  
388 was achieved. The functionalised PDMS negative moulds were coated with a thick layer of  
389 Sylgard 184 PDMS and placed on top of these dishes. The raised edges of the dishes

390 created a 0.6 mm thick layer of Sylgard 184 PDMS below the moulds, which would become  
391 part of the cuffs after curing. The moulds and dishes were placed in an oven at 65°C  
392 overnight.

393 The freshly-cured layer of PDMS was carefully peeled from the moulds and trimmed to the  
394 appropriate dimensions with a steel blade. The resulting PDMS implants were functionalised  
395 with an additional layer of silicone/polyacrylamide before rolling into cuffs. To roll into cuffs,  
396 the edges of the implants were brought together and carefully secured with insulation PVC  
397 tape. The edges were covered with RTV PDMS (SA03073, Farnell), which was allowed to  
398 cure overnight. An additional layer of RTV was then added and allowed to cure before the  
399 cuffs were stored in PBS and sterilised under UV prior to implantation. The resulting conduit  
400 had a length of 7 mm, an internal diameter of 1.5 mm, and a wall thickness of 0.6 mm.

401 *Subcutaneous implants.* A 3 mm thick layer of Sylgard 184 PDMS was cast and cured  
402 overnight at 65 °C. This was then trimmed using a steel blade into 5 x 5 mm blocks. Each  
403 block then was functionalised with a coating. Four blocks – one for each of the 4 stiffness-  
404 controlled conditions – were combined into one 10 x 10 mm implant and stuck together using  
405 RTV silicone. The sides of each of the four component blocks were notched to later be able  
406 to identify them. Dexamethasone-doped implants remained as 5 x 5 mm blocks, and were  
407 not combined with other implants. Implants were stored in PBS and sterilised under UV prior  
408 to implantation.

409 *Coatings.* To produce dexamethasone-doped silicone implants (Dex), Sylgard 184 PDMS  
410 was doped with 10 mg/ml of dexamethasone and spin-coated into 100 µm-thick films. The  
411 dexamethasone-doped films were cut into appropriately-sized rectangles. A small amount of  
412 RTV PDMS was spread over an implant and a dexamethasone-doped PDMS rectangle was  
413 placed on top. After allowing the RTV to cure overnight, the dexamethasone-doped film was  
414 further trimmed of any overhangs and the PDMS cuff was rolled as described above.

415 Soft silicone coatings (PDMS\_2kPa) were prepared from a mix of NuSil 8100 and Sylgard  
416 184 (99% to 1% w/w, respectively). An implant was thoroughly cleaned with ethanol and  
417 ddH<sub>2</sub>O and dried with nitrogen gas, followed by the application of a 9 µl drop of the soft  
418 silicone mix to its surface. This drop was spread out to ensure that the entire inner surface of  
419 the implant was completely covered. The implant was then transferred to an oven and baked  
420 at 65 °C for one week. This long curing time was a necessary step to remove traces of non-  
421 cured PDMS.

422 Polyacrylamide coatings (PAA\_0.2kPa and PAA\_20kPa) were grafted onto Sylgard 184  
423 PDMS implants following a published protocol<sup>53</sup>. Glass coverslips were cleaned by alternate  
424 dipping in ddH<sub>2</sub>O and EtOH. PDMS implants were thoroughly cleaned with methanol, dried  
425 with nitrogen gas, and covered with a benzophenone solution 10% w/w in EtOH (Sigma,  
426 B9300) for 2 min at room temperature. Benzophenone was removed and cuffs cleaned with  
427 methanol and dried with nitrogen gas. Polyacrylamide hydrogel mixes were prepared by  
428 combining acrylamide and bisacrylamide solutions at a 2:1 ratio. The mixes were combined  
429 with PBS to achieve the desired stiffness, as described in Supplementary Table 1. To initiate  
430 the polymerisation of the gel mix, 5 µl of APS solution (0.1 g/ml in ddH<sub>2</sub>O) and 1.5 µl of  
431 TEMED were added to 500 µl of gel mixes. A 9 µl drop of the gel mix was then transferred to  
432 a PDMS implant, which had been previously soaked in 10% (w/w) benzophenone solution in  
433 ethanol. The drops were spread out by covering with a clean glass coverslip. Implants were  
434 then quickly transferred under a 12 J/cm<sup>2</sup> UV lamp for 10 min (SUSS MicroTec MJB4). After  
435 the 10 min of UV exposure, glass coverslips were removed and polyacrylamide-PDMS  
436 composites were placed in PBS for a further 30 min. The polyacrylamide coating was kept  
437 hydrated with PBS at all times until implantation.

438 Stiff silicone implants (PDMS\_200kPa) were not coated with anything, leaving the surface of  
439 the Sylgard 184 implant exposed to the tissue.

440

441 ***In vivo* implantation.**

442 All experimental procedures were performed in accordance with the UK Animals (Scientific  
443 Procedures) Act 1986. Surgical procedures were carried out under aseptic conditions. ~250  
444 g Lewis rats (Charles River UK) were housed in groups of 5 and provided *ad libitum* access  
445 to food and water for a minimum of 7 days prior to surgical procedures. Immediately prior to  
446 all surgical procedures, animals received an injectable dose of the non-steroidal anti-  
447 inflammatory drug meloxicam (1.5 mg/ml, subcutaneous). Anaesthesia was induced and  
448 maintained with isoflurane delivered via a facemask. Body temperature was monitored via a  
449 rectal probe and maintained at 37 °C using a thermal blanket.

450 *Nerve conduits.* Biceps femoris and vastus lateralis muscles of the right leg of the animals  
451 were approached dorsally and separated to expose the septum through which the sciatic  
452 nerve travels. The sciatic nerve trifurcation point was located and followed 2 mm proximal.  
453 This site was used as a landmark to achieve consistent location of injury or implantation. The  
454 nerve was cleanly transected at this location using scissors and the conduit was positioned  
455 between the two resulting nerve stumps, leaving a 5 mm long empty gap within the conduit  
456 between the stumps. The epineurium of each nerve stump was sutured to the silicone tube  
457 using 9/0 nylon sutures (Ethicon). Each animal received only one conduit, with a single type  
458 of coating.

459 *Subcutaneous implants.* An incision was done dorsally over the right leg of an animal  
460 (approximately above the femur). The skin was separated from the underlying muscle using  
461 blunt forceps to create a tunnel from the site of incision towards the midline of the animal.  
462 The implant was fed through this tunnel and placed ~1 cm away from the midline, with the  
463 coating facing the layer of muscle. Each animal received one subcutaneous implant; either a  
464 composite stiffness or a dexamethasone-doped implant.

465 All animals were allowed to recover following implantation. A further dose of meloxicam was  
466 given orally the day after surgery. 3 months post-implantation, animals were sacrificed by  
467 exposure to a rising concentration of CO<sub>2</sub> and the tissue collected.

468

#### 469 **Immunohistochemistry.**

470 All tissue was fixed prior to processing and staining by immersion in paraformaldehyde  
471 solution (40 mg/ml in PBS) overnight at 4 °C. Samples which required sectioning were then  
472 transferred to a sucrose solution (30% w/w in PBS) for cryoprotection. They were kept in this  
473 solution for a minimum of 16 hr at 4 °C, and then stored until further processing.

474 Cryopreserved samples were embedded in optimal cutting temperature compound (Tissue-  
475 Tek, 4583), which was frozen and mounted on a cryostat (CM3050 S, Leica). 12 µm - thick  
476 sections were cut from the samples at a cutting temperature of -20 °C. Sections were placed  
477 on glass slides and allowed to dry at room temperature overnight before storage at -20 °C  
478 until stained.

479 Sections ready to be stained were washed in a Triton X-100 0.1% v/v solution in PBS to or  
480 permeabilisation. These and all further washes were performed three times for 10 min. To  
481 minimise non-specific antibody binding, sections were incubated in a blocking buffer,  
482 consisting of tris-buffered saline containing 0.03% v/v Triton X-100 and 10% v/v donkey  
483 serum (Millipore, s30-100ml). After blocking for 1 hr at room temperature, primary antibodies  
484 were added to sections (further details in Supplementary Table 2). Sections were covered  
485 with paraffin film to prevent drying and were incubated in primary antibodies overnight at 4  
486 °C.

487 Sections were washed in PBS-Triton solution to remove excess primary antibodies, and then  
488 incubated in secondary antibodies in blocking buffer for 2 hr at room temperature.

489 Secondary antibodies were finally washed off with a non-saline Tris-buffered solution.

490 Fluorsave mounting agent (Millipore, 345789) was added to sections to preserve  
491 fluorescence before encasing with a glass coverslip and storing at 4 °C prior to imaging.

492 Imaging of stained nerve tissue was carried out using a confocal microscope (Leica TCS  
493 SP5). Image files were exported and processed for analysis in Image-J software package  
494 (v1.48, National Institutes of Health, USA). Stain intensity profiles of FBR capsules was  
495 carried out through a combination of custom Matlab and Fiji scripts. The edge of the nerve  
496 capsule was delineated by the user and aligned by the scripts. An intensity profile (intensity  
497 vs. depth into the nerve) of the each stain was obtained. The average intensity from the  
498 edge of the nerve to a depth of 25  $\mu\text{m}$  was calculated and provided as a ratio to the same  
499 intensity of the PDMS\_200kPa group. The only exception were CD68 stains, were a depth of  
500 50  $\mu\text{m}$  was instead chosen as macrophages were found to mostly locate deeper into the  
501 tissue than other markers. Capsule thickness was using a Matlab script, after its edge was  
502 marked by hand based on the  $\alpha\text{SMA}$  stain. Axon density was analysed in an automated  
503 fashion using a Fiji script over 3 randomly chosen 100 x 100  $\mu\text{m}$  boxes for every image.  
504 Statistical analysis and data plotting was carried out using MATLAB (Mathworks, R2016b).

505

#### 506 **Atomic force microscopy.**

507 Sample elasticity was determined via atomic force microscopy (AFM) as previously  
508 described<sup>54</sup>. Indentation measurements using a cantilever probe were taken on samples  
509 placed on an inverted optical microscope (Axio Observer.A1, Carl Zeiss Ltd.) using a JPK  
510 Nanowizard Cellhesion 200 AFM (JPK Instruments AG). Tipless silicon cantilevers (Arrow-  
511 TL1; NanoSensors) with a spring constant of  $\sim 0.02$  N/m were used in experiments were  
512 tissue stiffness was measured. Material characterisation made use of either these or stiffer  
513 cantilevers (SICON-TL-20, spring constant  $\sim 0.29$  N/m, AppNano; TL-FM-10, spring constant  
514  $\sim 2.8$  N/m; Nanosensors; spring constant value calculated via the thermal noise method<sup>55</sup>).

515 Each cantilever had a polystyrene bead (~37  $\mu\text{m}$  diameter for tissue, ~20 $\mu\text{m}$  for stiffer  
516 materials; Microparticles GmbH) glued (ultraviolet curing, Loctite) prior to all measurements.

517 *Tissue preparation.* Lewis rats (Charles River UK) were sacrificed by overdose of euthatal  
518 (pentobarbitone) administered intraperitoneally, followed by neck dislocation. Euthatal was  
519 combined at a 1:1 v/v ratio with lidocaine anaesthetic and delivered at a total dose of 3 ml/kg  
520 of bodyweight. Further processing of tissue was carried out within 1 - 2 hr of animal sacrifice,  
521 and all AFM measurements were completed within 5 hr to minimise tissue degradation.

522 To dissect out the sciatic nerve, the dorsal side of the hindlegs was exposed and skin  
523 removed. Biceps femoris and vastus lateralis muscles were separated to expose the septum  
524 through which the sciatic nerve travels. The nerve was dissected out and transected just  
525 below the trifurcation point, and 1 cm above it. The nerve was then transferred to a dish  
526 containing mammalian physiological saline previously described by others<sup>56</sup> (121 mM NaCl,  
527 5 mM KCl, 1 mM  $\text{MgCl}_2$ , 1 mM  $\text{CaCl}_2$ , 0.4 mM  $\text{NaH}_2\text{PO}_4$ , 23.8 mM  $\text{NaHCO}_3$ , 5.6 mM  
528 glucose). Mammalian physiological saline was prepared freshly prior to the experiment. To  
529 establish a pH of 7.3, a gas mixture of 95%  $\text{CO}_2$  and 5%  $\text{O}_2$  was bubbled through the  
530 solution.

531 For naïve nerve measurements, under a dissection microscope, blood vessels and excess  
532 tissue surrounding the nerve were removed, and the nerve stumps were trimmed off with a  
533 steel blade. The remaining nerve was then cut into several fragments for mounting and  
534 sectioning. Nerve fragments were embedded in warm 4% w/w low melting point agarose  
535 (Sigma) in PBS. Agarose was allowed to cool and harden for a few minutes before trimming  
536 into blocks containing the nerve fragments. Blocks were stuck to a steel stage with  
537 cyanoacrylate glue and transferred to a chamber filled with chilled mammalian physiological  
538 saline. The blocks were cut into 500  $\mu\text{m}$  thick sections in a vibrating microtome. Nerve  
539 sections were transferred to mammalian physiological saline solution containing the live  
540 stain fluoromyelin (1:250 v/v in mammalian physiological saline; Invitrogen, F34651) and

541 incubated for 1 hr at room temperature to stain the myelin surrounding axons. This allowed  
542 the endoneurial compartment to be identified and later probed via AFM. Sections were  
543 mounted onto 35 mm plastic dishes (Z707651, Sigma). The sections were gently deposited  
544 onto two strips of cyanoacrylate glue which adhered to the agarose on which the tissue was  
545 embedded. Dishes were filled with room temperature mammalian phosphate buffer and  
546 transferred to the inverted microscope to perform the measurements.

547 For nerve FBR capsules and epineurium measurements, implanted cuffs were extracted  
548 from rats 3 months post-implantation with regenerated sciatic nerves still within them. Under  
549 a dissection microscope, fibrotic tissue covering the outside of cuffs was removed. A cut was  
550 done along the length of the cuff, and the regenerated nerve fragment within was carefully  
551 removed. The nerve fragment was embedded on its side on a shallow bed of 4% w/w low  
552 melting point agarose (Sigma), and submerged in mammalian physiological saline. Finally,  
553 AFM measurements of the side of the nerve were carried out.

554 *Material preparation.* The stiffness of polyacrylamide hydrogel and silicone rubber implants  
555 and substrates was checked for every manufactured batch by AFM. Implants and substrates  
556 were all cleaned by immersion in PBS overnight. These were then transferred to 35 mm  
557 plastic dishes and fixed in place using a small amount of vaseline petroleum jelly. Dishes  
558 were transferred to the inverted microscope to perform AFM measurements.

559 *Indentation experiments.* Petri dishes containing the samples to be analysed were placed on  
560 a motorised xy stage, which allowed movement of the sample relative to the AFM cantilever.  
561 A CCD camera (The Imaging Source GmbH) was used to image and track the position of the  
562 cantilever above the sample. This setup was used to locate and define an area of interest on  
563 the sample on which AFM measurements were taken. A custom python script broke down  
564 this area into 20 x 20  $\mu\text{m}$  squares, inside which a single measurement was taken. The  
565 motorised stage was moved as measurements were taken to perform a raster scan of the  
566 area of interest.



567 For each elasticity measurement, the cantilever probe was lowered onto the surface of the  
568 sample at a speed of 10  $\mu\text{m/s}$ . Upon contact and indentation of the sample, the probe  
569 continued to be lowered until a force of 10 nN was reached (usually equivalent to an  
570 indentation depth  $\delta$  of 1 to 5  $\mu\text{m}$ ). The probe was then retracted, the sample moved, and a  
571 measurement repeated at a different location.

572 The force-distance measurements taken by the AFM were translated into elasticity values  
573 using the Hertz model<sup>57</sup> using a previously described custom<sup>58</sup> MATLAB (Mathworks,  
574 R2008a) script. for every indentation of the sample. The cantilever and polystyrene bead  
575 probe on the sample was modelled as a sphere and a half space, and used to calculate the  
576 apparent reduced elastic modulus  $K$ .

$$F = \frac{4}{3}KR^{\frac{1}{2}}\delta^{\frac{3}{2}}$$

577 With  $F$  being the force applied and  $R$  the radius of the polystyrene bead. Elasticity was  
578 calculated at an indentation depth  $\delta$  of 2 mm. The reduced elastic modulus may be further  
579 transformed into other elastic moduli, including Young's modulus ( $E$ )<sup>58</sup> and Shear modulus  
580 ( $G$ ). A Poisson ratio  $\nu$  of polyacrylamide was set to 0.48<sup>59</sup>, while for PDMS a value of 0.499  
581 was used<sup>60</sup>.

$$K = \frac{E}{1 - \nu^2}$$

$$G = \frac{E}{2(1 + \nu)}$$

582

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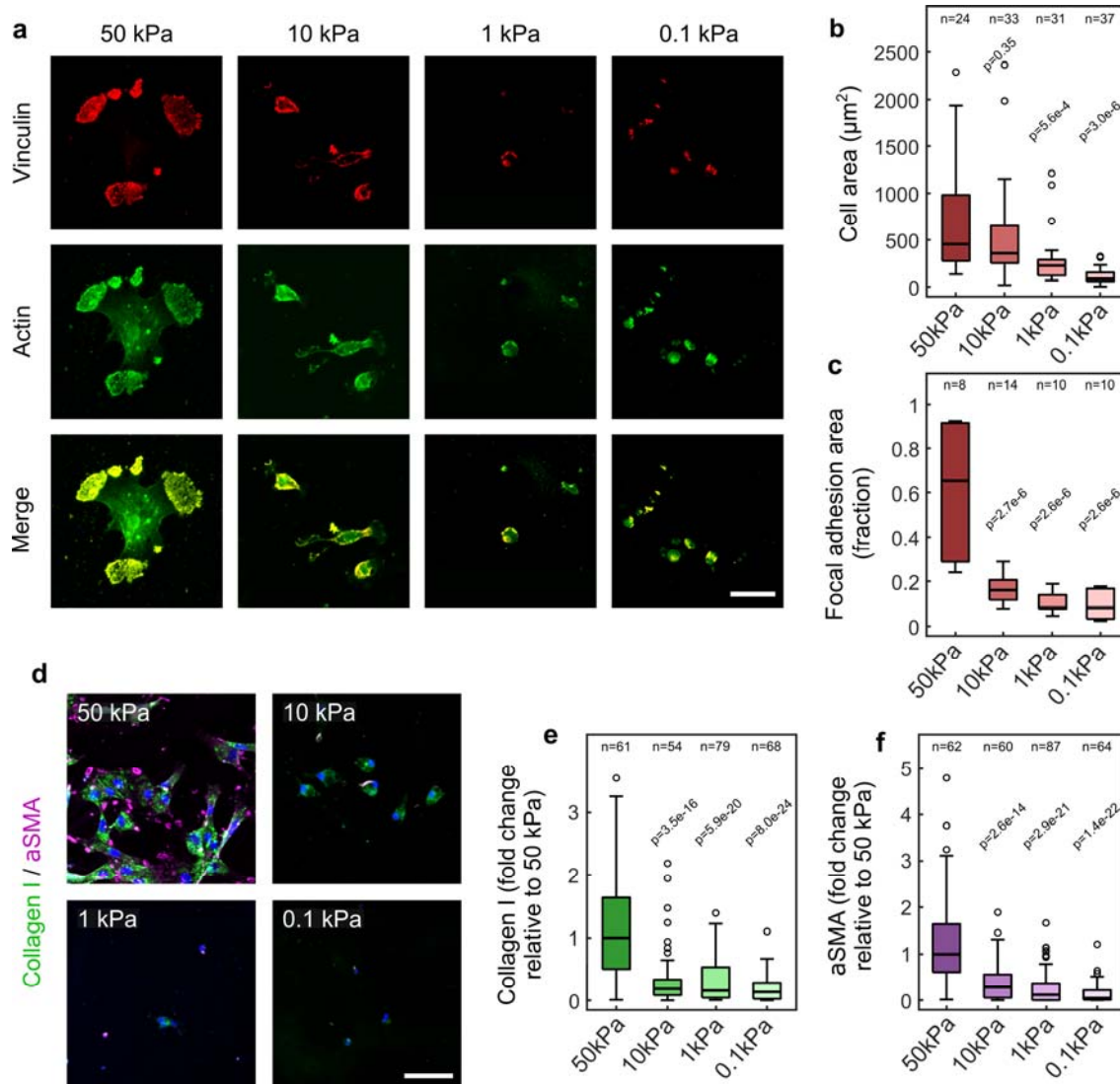
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731



732 **Figures**

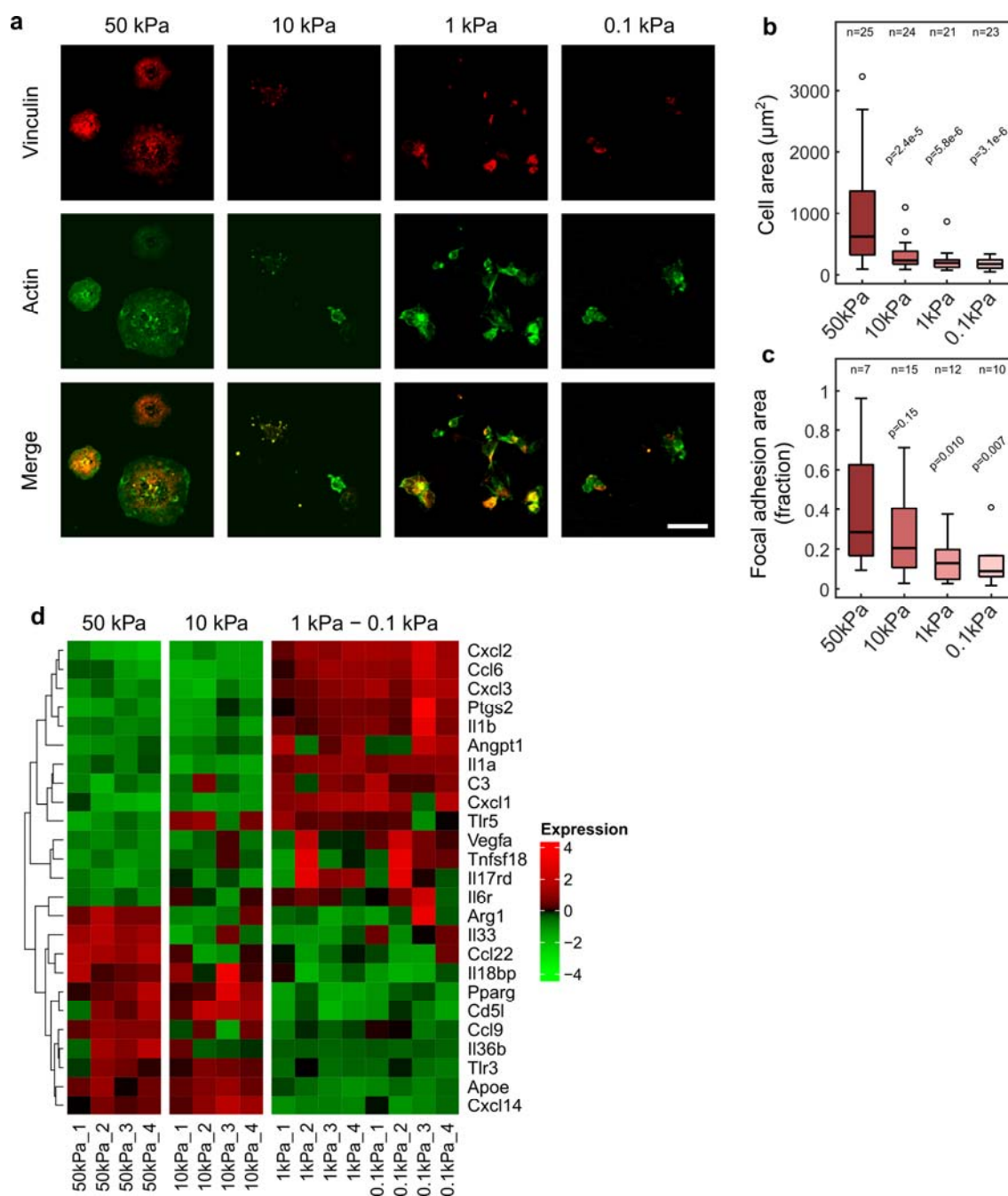


733

734 **Fig 1 | Substrates with a stiffness above that of the native tissue trigger fibrosis *in***  
 735 ***vitro*.** **a**, Maximum intensity projections of z-stack confocal images of nerve fibroblasts at 6  
 736 DIV cultured on polyacrylamide substrates of various stiffness (50, 10, 1 and 0.1 kPa shear  
 737 modulus), stained for cytoskeleton (actin) and focal adhesion (vinculin) markers. Cell  
 738 morphologies significantly changed on non-physiologically stiff (shear modulus of 50 kPa)  
 739 substrates. Scale bar: 30  $\mu\text{m}$ . **b,c**, Box plots of fibroblast cell area (**b**) and focal adhesion  
 740 area (**c**).  $n$  = number of cells. **d**, Images of fibroblasts stained for myofibroblast markers  
 741  $\alpha$ SMA (magenta) and collagen I (green) show an increase in fibrotic phenotype on 50 kPa

742 gels. Cell nuclei stained with DAPI (blue) Scale bar: 60  $\mu\text{m}$ . **e,f**, Box plots of relative stain  
743 intensities for collagen I (**e**) and  $\alpha\text{SMA}$  (**f**).  $n$  = number of cells. All statistical comparisons  
744 done via one-way ANOVA followed by Dunnett's multiple comparisons test comparing to the  
745 50 kPa condition. All experiments performed 3 - 4 times.

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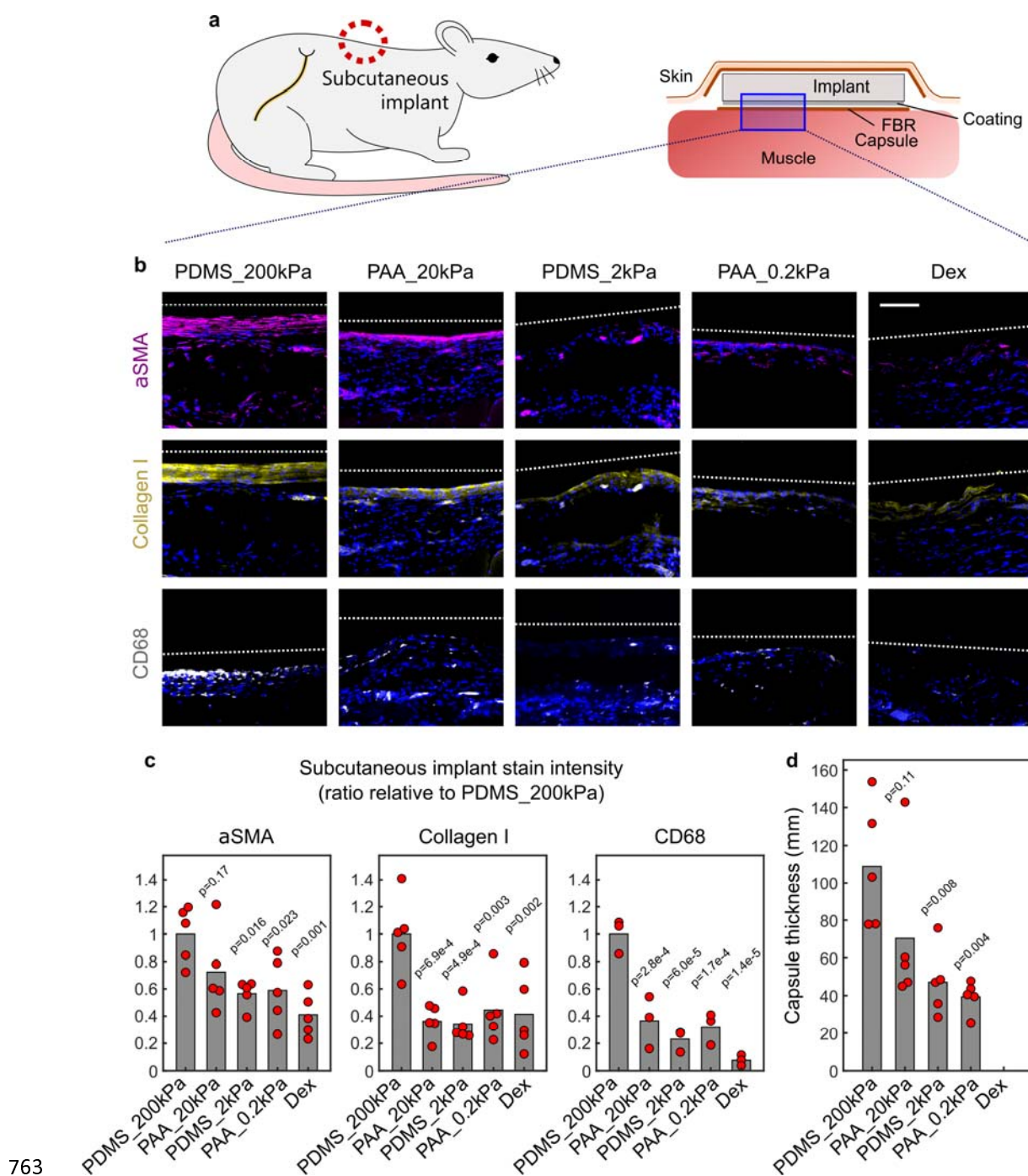


747

748 **Fig 2 | Substrates with a stiffness above that of the native tissue trigger macrophage**  
 749 **activation and changes in inflammatory profile *in vitro*.** **a**, Maximum intensity projections  
 750 of z-stack confocal images of bone marrow-derived macrophages at 6 DIV cultured on  
 751 polyacrylamide substrates of various stiffness (50, 10, 1 and 0.1 kPa shear modulus),  
 752 stained for cytoskeleton (actin) and focal adhesion (vinculin) markers. Cell morphologies  
 753 significantly changed on non-physiologically stiff (50 kPa) substrates. Scale bar: 30  $\mu$ m. **b,c**,

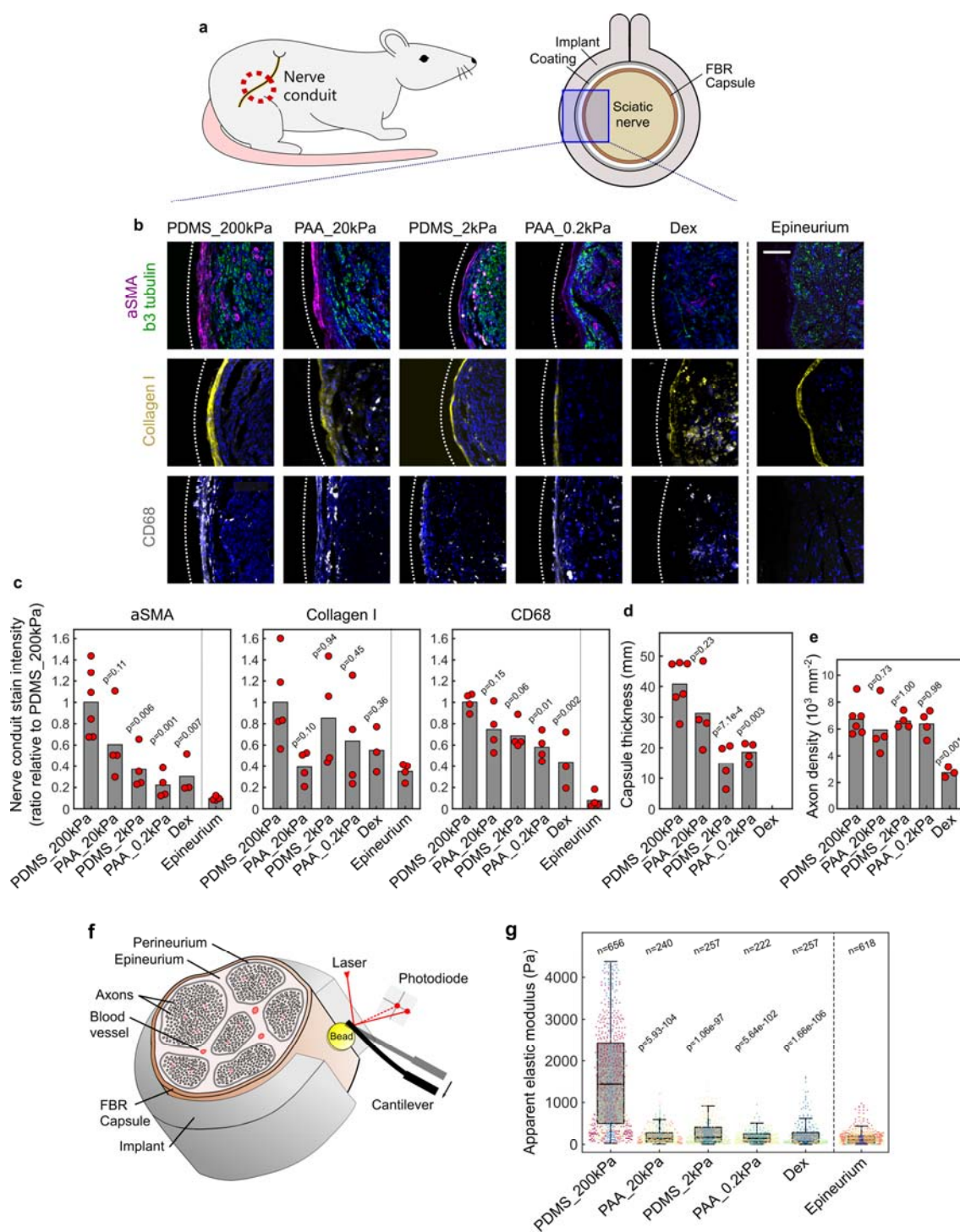
754 Boxplots of macrophage cell area (**b**) and focal adhesion area (**c**). *n* = number of cells. All  
755 statistical comparisons done via one-way ANOVA followed by Dunnett's multiple  
756 comparisons test comparing to the 50 kPa condition. **d**, Heatmap of changes in inflammatory  
757 differentially expressed gene (DEG) expression profile of macrophages at 3 DIV detected in  
758 RNAseq (expression values represented as mean centred rlog-transformed counts).  
759 Changes in markers such as *arg1*, *pparg*, *il1b*, and *ptgs2* indicate a switch to an M2-like  
760 phenotype of macrophages grown on high stiffness substrates. Four samples analysed for  
761 each substrate stiffness condition.

762



769 2 kPa PDMS (PDMS\_2kPa), 0.2 kPa polyacrylamide (PAA\_0.2kPa), or PDMS impregnated  
770 with 10 mg/ml of the anti-inflammatory drug dexamethasone (Dex). Tissue is fluorescently  
771 labelled for myofibroblasts ( $\alpha$ SMA), extracellular matrix components (collagen I), and  
772 macrophages (CD68). Approximate edges of implant are indicated by dashed white lines. All  
773 images show DAPI stains of nuclei (blue). FBR was significantly alleviated when the implant  
774 was coated with a soft material of  $G \leq 2$  kPa. Scale bar: 100  $\mu$ m. **c**, Plot of stain intensities.  
775 **d**, Plot of fibrotic capsule thickness. Quantification of thickness in Dex group absent as  
776 dexamethasone impeded formation of a structured boundary between tissue and implant.  
777 For all plots: bars represent mean, dots represent individual animals.  $N = 5$  rats. Statistical  
778 comparisons carried out via one-way ANOVA followed by Dunnett's multiple comparisons  
779 test comparing groups to the PDMS\_200kPa control condition.

780



781

782 **Fig 4 | Soft coatings significantly reduce foreign body reaction to nerve conduits 3**

783 **months post-implantation *in vivo*.** **a**, Schematic of implant location and structure. Implants

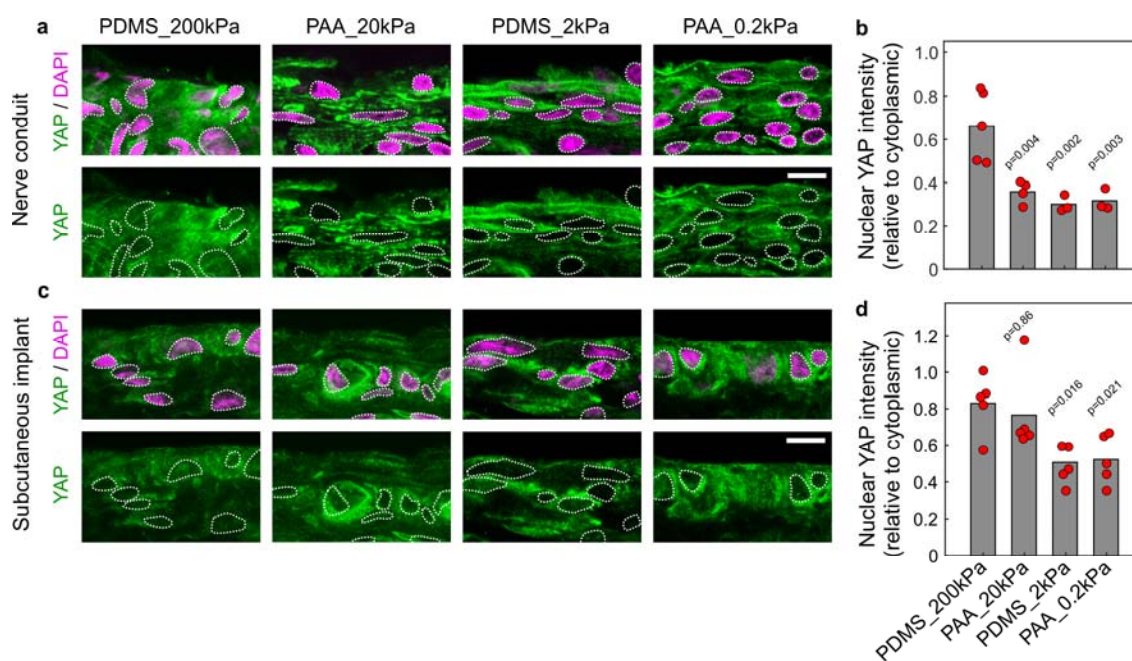
784 had an internal diameter of 1.5 mm. Blue square represents area shown in images in **b**. **b**, Z-

785 stack confocal images of sciatic nerve tissue regenerated through conduits of 200 kPa shear

786 modulus silicone PDMS (PDMS\_200kPa), or conduits coated with 20 kPa polyacrylamide  
787 hydrogel (PAA\_20kPa), 2 kPa PDMS (PDMS\_2kPa), 0.2kPa polyacrylamide (PAA\_0.2kPa),  
788 or PDMS impregnated with 10 mg/ml dexamethasone (Dex). Approximate edges of implant  
789 are indicated by dashed white lines. Tissue is fluorescently labelled for axons ( $\beta$ 3 tubulin),  
790 myofibroblasts ( $\alpha$ SMA), extracellular matrix components (collagen I), and macrophages  
791 (CD68). All images are stained for nuclei (DAPI, blue). Images of non-operated naïve nerves  
792 are also included for comparison (epineurium). Softer coatings reduced cell activation and  
793 fibrotic capsule thickness while permitting nerve regeneration. Dex-treatment, however, not  
794 only abolished the fibrotic capsule but also nerve regeneration. Scale bar: 100  $\mu$ m. **c**, Plot of  
795 relative stain intensities. **d**, Plot of axon density in nerves 5 mm downstream of implantation  
796 site. **e**, Plot of fibrotic capsule thickness. Quantification of thickness in Dex group absent as  
797 dexamethasone impeded formation of a structured boundary between tissue and implant.  
798 For all plots: bars represent mean, dots represent individual animals.  $N = 3$  to 6 rats. **f**,  
799 Diagram of *ex vivo* AFM setup for nerve tissue stiffness measurements. **g**, Box and scatter  
800 plot of tissue stiffness values for tissue in proximity to the implants with various coatings,  
801 showing a significant stiffening of tissue around implants with a stiff surface. Tissue stiffness  
802 was similar around PDMS\_2kPa, PAA\_0.2kPa, and Dex-treated implants ( $p > 0.05$ ).  $n =$   
803 number of measurements. Measurements taken on  $N = 3$  to 6 rat nerves; measurements  
804 taken from the same animal are depicted with the same colour. All statistical comparisons  
805 carried out via one-way ANOVA followed by Dunnett's multiple comparisons test comparing  
806 groups to the PDMS\_200kPa control condition. Epineurium condition not included in  
807 statistical analysis. Bonferroni-corrected Student's t-test used for comparisons to Dex group.  
808



809



810

811 **Fig 5 | Reduction in foreign body reaction *in vivo* correlates with exclusion of the**  
812 **mechanosensitive transcriptional regulator YAP from nuclei in tissues surrounding**  
813 **soft-coated implants. a,c,** Confocal images of nerve (a) and subcutaneous tissue (c)  
814 stained for the transcriptional regulator YAP and nuclei (DAPI). Tissue was exposed to  
815 implants of 200 kPa shear modulus silicone PDMS (PDMS\_200kPa), or to implants coated  
816 with 20 kPa polyacrylamide hydrogel (PAA\_20kPa), 2 kPa PDMS (PDMS\_2kPa), or 0.2 kPa  
817 polyacrylamide (PAA\_0.2kPa), for 3 months. YAP preferentially localised into nuclei of cells  
818 in contact with stiff but not with soft-coated implants, indicating a mechanosensitive process.  
819 Nuclei delineated by dashed white lines. Scale bars: 10  $\mu$ m. **b,d,** Plots of relative nuclear  
820 YAP intensities. Bars represent mean, dots represent individual animals.  $N = 3$  to 5 rats (b)  
821 and  $N = 5$  rats (d). All statistical comparisons carried out via one-way ANOVA followed by  
822 Dunnett's multiple comparisons test comparing groups to the PDMS\_200kPa condition.

823